

**POME-derived Ag/TiO<sub>2</sub>-based ternary nanocomposites for  
oxytetracycline degradation under visible light and membrane  
integration**

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## DECLARATION

I hereby declare that the work presented in this thesis was conducted in full compliance with the regulations of Universiti Malaysia Sarawak (UNIMAS). Except where proper acknowledgment is given, this work is solely the effort of the author. This thesis has not been accepted for the award of any other degree and is not being **concurrently** submitted for any other academic qualification.

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## ***POME-derived Ag/TiO<sub>2</sub>-based photocatalyst for oxytetracycline degradation under visible light and membrane integration***

### **ABSTRACT**

The rise of emerging pollutants demands efficient and sustainable water treatment technologies. Ag/TiO<sub>2</sub>-driven photocatalysis is a viable solution due to its durable visible-light activity and antimicrobial properties, yet stability, recovery, and overall synthesis sustainability remain major challenges that hinder its practical application. Therefore, the need to develop a green, durable, and easily recoverable Ag/TiO<sub>2</sub>-based photocatalyst is crucial to advance photocatalysis as an eco-friendly solution for the treatment of complex pollutants. To overcome these drawbacks, this study explored the green synthesis of Ag/TiO<sub>2</sub>-based ternary nanocomposites using palm oil mill effluent (POME) as a natural reducing and stabilizing agent. Two ternary photocatalysts, Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> (ABMOT) and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> (AGCNT), were synthesized via a microwave-assisted method and extensively characterized by Fourier Transform Infrared Spectroscopy (FTIR), X-ray Diffractometry (XRD), Scanning Electron Microscopy with Energy-Dispersive X-ray (SEM-EDX), and UV-Visible Diffuse Reflectance Spectroscopy (UV-Vis DRS). Their photocatalytic activities were evaluated against oxytetracycline hydrochloride (OTC). Under visible-light irradiation, ABMOT achieved 97.2% OTC degradation at an optimal dosage of 0.6 g/L, while AGCNT reached 81.3% under the same conditions. ABMOT demonstrated stronger dark-adsorption capacity and greater tolerance to increasing pollutant concentrations than AGCNT, indicating better surface affinity and active-site accessibility. Regardless, both nanocomposites showed good stability over four consecutive recycling cycles, indicating superior stability and reusability potential. Additionally, both nanocomposites further demonstrated antibacterial activity against *Escherichia coli* and *Staphylococcus aureus*, confirming their multifunctional potential. To address the limitation of powder recovery in slurry photocatalysis, both nanocomposites were immobilized in polyacrylonitrile (PAN) membranes via the phase-inversion method. Structural and surface analyses confirmed successful integration, with nanocomposite-loaded membranes exhibiting enhanced hydrophilicity and porosity. Permeation studies revealed that AGCNT@PAN (2 wt%) achieved the highest pure water flux (170.94 L·m<sup>-2</sup>·h<sup>-1</sup>) and rejection rate (90.34%) compared to bare PAN membranes with 149.51 L·m<sup>-2</sup>·h<sup>-1</sup> pure water flux and 44.88 % rejection rate. Consequently, ABMOT@PAN also showed improvements. Overall, this study demonstrates a novel, eco-friendly approach for synthesizing multifunctional photocatalysts by valorizing agro-industrial waste. These findings provide valuable insights into advancing green nanotechnology for sustainable water treatment applications.

**Keywords:** Palm oil mill effluent, green synthesis, titanium dioxide, silver, photocatalysis, antibacterial, membrane.

## ***Fotopemangkin berasaskan Ag/TiO<sub>2</sub> daripada POME untuk degradasi oksitetrasiklin di bawah cahaya nampak dan integrasi dalam membran***

### **ABSTRAK**

Peningkatan bahan cemar muncul yang semakin membimbangkan memerlukan teknologi rawatan air yang cekap dan lestari. Fotopemangkinan yang berasaskan Ag/TiO<sub>2</sub> berpotensi menjadi salah satu kaedah untuk mengatasi isu ini kerana keupayaannya untuk diaktifkan di bawah cahaya nampak serta sifat antibakterianya. Namun demikian, aplikasi fotopemangkin berasaskan-Ag/TiO<sub>2</sub> masih terhad disebabkan oleh isu kestabilan, kesukaran untuk perolehan semula dan cabaran dalam kaedah sintesis yang lestari. Oleh itu, terdapat keperluan untuk membangunkan fotopemangkin berasaskan-Ag/TiO<sub>2</sub> yang hijau, lestari dan mudah untuk diperolehi semula agar teknologi fotopemangkinan dapat dikembangkan sebagai solusi mampan untuk rawatan air serta degradasi bahan cemar muncul. Kajian ini meneroka sintesis hijau nanokomposit pertigaan berasaskan Ag/TiO<sub>2</sub> dengan menggunakan efluen kilang kelapa sawit (POME) sebagai agen penurun dan penstabil semula jadi. Dua fotopemangkin pertigaan iaitu, Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> (ABMOT) dan Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> (AGCNT), disintesis melalui kaedah mikrogelombang dan dicirikan menggunakan FTIR, XRD, SEM-EDX dan UV-Vis DRS. Ujian terhadap oksitetrasiklin hidroklorida (OTC) menunjukkan ABMOT dapat mencapai degradasi sebanyak 97.2% dengan dos optimum 0.6 g/L, manakala AGCNT mencatatkan 81.3% di bawah cahaya tampak. Selain itu, kapasiti penyerapan dalam keadaan gelap untuk ABMOT lebih tinggi jika dibandingkan dengan AGCNT. Ini menunjukkan bahawa ABMOT mempunyai toleransi yang lebih tinggi untuk kepekatan bahan cemar yang lebih tinggi kerana afiniti permukaan yang lebih sesuai terhadap oksitetrasiklin. Walau bagaimanapun, kedua-dua nanokomposit menunjukkan kestabilan yang baik untuk empat kitaran degradasi secara berterusan yang menjadi penanda aras kepada keupayaan pengitaran semula. Kedua-dua nanokomposit turut menunjukkan keberkesanan antibakteria terhadap *Escherichia coli* dan *Staphylococcus aureus*. Bagi menangani kekangan pemulihan serbuk, nanokomposit telah dimuatkan ke dalam membran poliakrilonitril (PAN) melalui kaedah kensongsangan fasa. Membran yang tersekatgerak dengan nanokomposit menunjukkan peningkatan hidrofiliti dan porositi, dengan AGCNT@PAN (2 wt%) mencapai fluks air tulen tertinggi (170.94 L·m<sup>-2</sup>·h<sup>-1</sup>) dan kadar penyingkiran sebanyak 90.34% berbanding dengan membran PAN tanpa pengubahsuaian yang hanya mampunyai fluks air tulen sebanyak 149.51 L·m<sup>-2</sup>·h<sup>-1</sup> dan kadar penyingkiran sebanyak 44.88%. Secara keseluruhan, kajian ini membuktikan pendekatan mesra alam dan baharu untuk menghasilkan fotopemangkin pelbagai fungsi melalui pemanfaatan sisa industri agro POME. Penemuan ini memberikan perspektif yang bermakna ke arah kemajuan nanoteknologi hijau bagi aplikasi rawatan air yang lestari.

**Kata Kunci:** Efluen Kilang Kelapa Sawit (POME), sintesis hijau, titanium dioksida, argentum, fotopemangkinan, antibakteria, membran

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## LIST OF ABBREVIATIONS

ABMOT	Ag/Bi <sub>2</sub> MoO <sub>6</sub> /TiO <sub>2</sub>
AGCNT	Ag/Bi <sub>2</sub> MoO <sub>6</sub> /TiO <sub>2</sub>
ATR	Attenuated Total Reflectance
DRS	Diffuse Reflectance Spectroscopy
EDC	Endocrine-Disrupting Chemical(s)
EDX	Energy Dispersive X-ray
FTIR	Fourier Transform Infrared Spectroscopy
GAE	Gallic Acid Equivalent
NCs	Nanocomposite(s)
NPs	Nanoparticle(s)
OTC	Oxytetracycline hydrochloride
PAN	Polyacrylonitrile
PBS	Phosphate Buffer Solution
POME	Palm Oil Mill Effluent
PPCPs	Pharmaceutical and Personal Care Products
PWF	Pure Water Flux
QE	Quercetin Equivalent
RR	Rejection Rate
SEM	Scanning Electron Microscopy
SPR	Surface Plasmon Resonance
TEM	Transmission Electron Microscopy
TPC	Total Phenolic Content
TFC	Total Flavonoid Content
UV-Vis	Ultraviolet-Visible
XRD	X-ray Diffractometry

CHAPTER 1:  
**INTRODUCTION**

## **1.1 Background of Study**

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As population growth and industrialization continue to progress around the world, the inevitable fate of water contamination from the penetration of various emerging pollutant sources poses a threat to both human well-being and the environment. Among various emerging pollutants, pharmaceutical and personal care products (PPCPs) have been recognized as an environmental concern due to their persistence, bioaccumulation potential, and adverse effects on aquatic ecosystems and human health Hena et al. (2021). Despite lacking regulatory status, the European Union (EU) announced that at least 33 compounds were prioritized for removal over the next 2 decades (Arumugam et al., 2025). Some of the compounds that were listed include anti-inflammatory drugs (e.g. ibuprofen, diclofenac) and antibiotics (Kumar et al., 2023). These compounds are well known to be endocrine-disrupting chemicals (EDCs) or carcinogenic. Although they are typically found in trace amounts (~0.1 ng/L to 200 ng/L), the concentrations of these compounds have been increasing in recent years, raising concerns about potential environmental risks (Hena et al., 2021; Kanakaraju et al., 2025; Ślósarczyk et al., 2021).

Despite the existence of conventional technologies in wastewater treatment, such as coagulation, flocculation and biological methods, they are limited in their ability to completely remove decontamination of the said emerging pollutants, including persistent organic/inorganic pollutants, and heavy metals (Ahmed & Haider, 2018; Anvari et al., 2021; Della Rocca et al., 2021). Generally, these processes end up producing secondary pollutants, which require further processing (Kweinor Tetteh & Rathilal, 2021). Therefore, there is a need to develop advanced wastewater treatment methods that efficiently purify contaminated water in a more sustainable way.

Advanced oxidation processes (AOPs) have emerged as effective wastewater treatment technologies for removing persistent, non-biodegradable organic pollutants. AOPs

operate by generating highly reactive oxygen species (ROS), particularly hydroxyl radicals ( $HO \bullet$ ), which have a strong oxidation potential ( $\sim 2.8$  eV) and can degrade a wide range of organic contaminants into simpler and less harmful compounds (Kumari & Kumar, 2023). Unlike conventional treatment methods, AOPs can mineralize recalcitrant pollutants rather than merely transferring them from one phase to another. Common AOPs used for the degradation of organic contaminants include ozonation, Fenton and photo-Fenton processes, electrochemical oxidation, and photocatalysis, the latter of which has gained increasing attention due to its operational simplicity and environmental compatibility.

In recent years, advances in nanotechnology have highlighted the use of AOPs as a promising tool for wastewater treatment across various settings (Kweinor Tetteh & Rathilal, 2021). Photocatalysis has been extensively studied for water treatment because it uses semiconductor materials and light energy to drive redox reactions. In a typical photocatalytic process, irradiation with light energy equal to or greater than the semiconductor's band gap excites electrons from the valence band (VB) to the conduction band (CB), generating electron-hole pairs. The photogenerated holes oxidize water or hydroxide ions to form hydroxyl radicals ( $HO \bullet$ ), while the excited electrons reduce dissolved oxygen to generate superoxide radicals ( $O_2 \bullet$ ). These reactive species play a crucial role in the degradation and mineralization of organic pollutants in aqueous systems.

Among numerous nano-based materials, titanium dioxide ( $TiO_2$ ) is widely applied as a nanocatalyst in the wastewater treatment method known as photocatalysis (Celebi et al., 2021; Dahl et al., 2014). This is primarily due to  $TiO_2$  being non-toxic, chemically stable, and possessing efficient charge separation, which contributes to its high photocatalytic activity (Zoubi et al., 2021). However, pristine  $TiO_2$  has some drawbacks due to its high bandgap, which leads to rapid recombination of photogenerated electron-hole pairs. This limits the photocatalytic effectiveness of  $TiO_2$ , as its photoexcitation is primarily confined to the ultraviolet (UV) spectrum, with minimal functionalization in the visible light region (Lin et al., 2006). Therefore, extensive research has been conducted to address the limitations of pristine  $TiO_2$  by incorporating dopants (e.g. metals and non-metal elements) or co-catalysts (e.g. perovskites, semiconductors, etc.) in the fabrication process (Zhang et al., 2021).

Past researchers have successfully fabricated  $TiO_2$  with plasmonic metals like gold (Au) and silver (Ag) using various methods including sol-gel, photoreduction and solvothermal (Wang et al., 2019; Wu et al., 2021; Yang et al., 2022). These materials can

enhance the photocatalytic performance of TiO<sub>2</sub>, as plasmonic metals cause photoexcitation of the binary composite under visible light through surface plasmon resonance (SPR). Perovskite-based catalysts, such as bismuth (Bi)-based photocatalysts, on the other hand, can align the Fermi energy levels of the binary due to its narrow bandgap with a wide range of light absorption (Arif et al., 2021). Given its lower energy bandgap ( $E_g = 2.5 \text{ eV} - 2.9 \text{ eV}$ ) in comparison to TiO<sub>2</sub>, Bi-based perovskite, such as bismuth molybdate (Bi<sub>2</sub>MoO<sub>6</sub>), has suitable band structures to form heterojunctions which minimize electron-hole recombination, improve charge separation efficiency, and combine the favorable traits of the two photocatalysts (Belousov et al., 2024). Additionally, the incorporation of graphitic carbon nitride (g-C<sub>3</sub>N<sub>4</sub>) has also been widely studied. g-C<sub>3</sub>N<sub>4</sub> is advantageous for its ability to harness visible light and for its suitable energy bandgap ( $E_g = 2.7 \text{ eV}$ ), which enables it to form heterojunctions with TiO<sub>2</sub>, substantially boosting the collective photocatalytic performance of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> heterojunction systems (Nemiwal et al., 2021).

Incorporation of g-C<sub>3</sub>N<sub>4</sub> and Bi<sub>2</sub>MoO<sub>6</sub> in the fabrication of Ag/TiO<sub>2</sub>-based ternary composites has shown potential for the treatment of different pollutants (Ranjithkumar et al., 2023; Yin et al., 2021). However, there remains a need for a greener, simpler fabrication process to enhance photocatalytic performance under visible light. In most studies, the fabrication process uses harsh solvents (Aravind et al., 2021). Hence, the possibility of utilizing plant extracts, microorganisms, or agricultural waste via green synthesis can be performed to mediate the fabrication of ternary heterostructures of Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>, providing a greener and more benign synthesis alternative.

The palm oil industry is among the largest in tropical regions such as Southeast Asia and Africa. Generally, about 10% of palm produce generates oil, and the other 90% ends up as vegetative waste such as empty fruit bunches, palm press fibers, palm kernel shells, and palm oil mill effluent (POME) (Ofori-Boateng & Lee, 2013). These wastes contain an abundance of phytochemicals such as carotenoids, phenolics, sterols, flavonoids and tocopherols, which have been previously studied and applied for the synthesis of various nanomaterials (Gan et al., 2012; Lucas-Gómez et al., 2020; Pradhan et al., 2022; Shankar & Rhim, 2016; Sowani et al., 2016). While the high organic content in POME could pose a threat to the environment, these unutilized compounds could also be a promising resource in the green synthesis of nanomaterials.

While conventional Ag/TiO<sub>2</sub>-based NCs exhibit promising photocatalytic activity, their synthesis often relies on chemical reducing agents and surfactants that may introduce

surface defects, residual toxicity, or poor interfacial stability (Bhardwaj & Singh, 2021). In contrast, POME contains naturally occurring phenolic compounds, flavonoids, and organic acids that can act as mild reducing agents, capping agents, and stabilizers during nanocomposite formation (Imam et al., 2025). These biomolecules promote controlled nucleation, improved particle dispersion, and enhanced interfacial contact between composite components. As a result, POME-derived nanocomposites are expected to exhibit comparable or improved photocatalytic and antibacterial performance while offering a greener, more sustainable synthesis route and improved surface functionality for pollutant interaction.

## 1.2 Problem Statements

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Previous studies have successfully demonstrated the green synthesis of Ag nanoparticles (NPs) (AgNPs) and gold nanoparticles (AuNPs) by utilizing POME as a reducing and stabilizing agent (Aliero et al., 2024; Gan et al., 2012). While these works have greatly contributed to the advancement of POME-derived nanomaterials, they largely focus on individual NPs. The fabrication of more complex ternary NCs requires more nuanced methods, especially when using an unconventional green source such as POME. Despite growing interest in sustainable synthesis routes, few studies explore the formation of ternary nanocomposites using POME as a key component. Furthermore, no comprehensive studies have investigated the photocatalytic and functional performance of POME-derived ternary systems.

Another drawback that needs to be addressed in the application of NCs in photocatalytic treatment is the difficulty of separating their powder post-treatment. One way to overcome this drawback is to immobilize the NC on materials that are easily removed or replaced, which has recently attracted research interest in integrating photocatalytic NCs with polymeric membranes (Elrasheedy et al., 2019). Amongst various membranes, polyacrylonitrile (PAN) is a well-known commercial polymer with good solubility in organic solvents, unique chemical and thermal properties, radiation stability, and low cost. However, PAN-based membranes also have disadvantages, including low chemical stability, fouling, and hydrophobicity. To enhance its properties and address its shortcomings, PAN can be modified with nanomaterials.

Therefore, this study aims to use POME extracts in the green fabrication of ternary Ag/TiO<sub>2</sub>-based NCs, namely Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>, and assess their effectiveness and functionality as visible-light-driven photocatalysts for degrading oxytetracycline hydrochloride (OTC), a model emerging pollutant. OTC has been chosen as the model pollutant owing to its widespread use as a veterinary and human antibiotic, especially in aquaculture. Furthermore, this study emphasizes incorporating POME-derived NCs into PAN membranes via the phase-inversion method. To develop a multifunctional membrane-photocatalyst system for potential wastewater treatment applications, a performance comparison with a powder-based photocatalytic system was also presented.

### 1.3 Research Hypothesis

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This study hypothesizes that Ag/TiO<sub>2</sub>-based ternary NCs synthesized using POME as a green reducing and stabilizing agent will exhibit enhanced physicochemical properties, visible-light photocatalytic activity, and antibacterial performance compared to conventionally synthesized counterparts. It is further hypothesized that immobilizing these NCs into polyacrylonitrile (PAN) membranes will improve material recovery and filtration performance without significantly compromising their functional properties.

### 1.4 Research Question

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With respect to the aforementioned criteria, several research questions were raised:

- i. How does the use of POME extracts influence the morphology, crystallinity and chemical composition of Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> NCs?
- ii. How effective are the POME-derived ternary NCs against OTC?
- iii. How effective are POME-derived ternary NCs against gram-positive and gram-negative bacteria?
- iv. What are the effects of incorporating POME-derived NCs into PAN membranes on the permeability, separation efficiency and physicochemical properties?
- v. Does the use of POME as a green synthesis agent present a viable alternative to chemical synthesis routes in terms of environmental and functional performance?

## 1.5 Research Objectives

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Hence, the objectives of this study were:

- i. To synthesize ternary Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> NCs using POME extracts as a green reducing and stabilizing agent, and characterize their physicochemical and structural properties.
- ii. To evaluate the photocatalytic performance of the synthesized ternary NC in degrading OTC under visible light,
- iii. To assess the antibacterial activity of the synthesized ternary NCs against gram-positive (*E. coli*) and gram-negative (*S. aureus*) bacteria,
- iv. To fabricate PAN membranes embedded with POME-derived ternary NCs via the phase inversion method and determine their permeability and separation efficiency,
- v. To compare the removal efficiency of bare ternary NC powders with membrane-integrated systems for the removal of OTC.

## 1.6 Significance of Study

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This study successfully demonstrated a green and sustainable approach in synthesizing ternary Ag/TiO<sub>2</sub>-based NCs using POME, which is an underutilized agro-industrial waste. By extending the application of POME beyond the synthesis of singular NPs to the fabrication of more complex heterostructure NC systems, this research contributed to the development of an eco-friendly photocatalyst with good functionality under visible light. In addition to evaluating the potential of POME-derived NCs in powder form, this research explored their incorporation into PAN membranes via the phase inversion method. Although the membrane systems were not applied under photocatalytic conditions, their basic filtration performance was evaluated to gain insight into how NC loading influences membrane properties, providing a basis for understanding the effectiveness of incorporating ternary NCs into membrane systems. The outcomes of this study provided useful insights into the viability of using POME as a green route for synthesizing ternary photocatalysts and the potential of integrating ternary NCs into filtration membranes for future multifunctional water treatment applications. This work supports the advancement of sustainable materials and highlights a pathway for the valorization of agro-industrial waste into high-value components in material science and environmental remediation technologies.

## 1.7 Research Hypothesis

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This study focuses on the green synthesis of Ag/TiO<sub>2</sub>-based ternary NCs using POME as a natural reducing and stabilizing agent. Two photocatalysts, Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>, were synthesized via a microwave-assisted method and characterized for their physicochemical, structural, and optical properties using FTIR, XRD, SEM-EDX, and UV-Vis DRS.

The photocatalytic performance of the synthesized NCs was evaluated under visible-light irradiation for the degradation of OTC as a model emerging pollutant. The effects of catalyst dosage, initial pollutant concentration, and reusability were investigated. Antibacterial activity against gram-negative (*E. coli*) and gram-positive (*S. aureus*) bacteria was also assessed.

To address limitations in catalyst recovery, the synthesized NCs were incorporated into PAN membranes via the phase-inversion method. The membrane study was limited to evaluating morphology, permeability, and rejection performance under filtration conditions. Photocatalytic membrane operation, large-scale application, and by-product toxicity analysis are beyond the scope of this study.

# LITERATURE REVIEW

## 2.1 Advanced oxidation processes

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Advanced oxidation processes (AOPs) are powerful chemical treatment technologies that emerged to remediate water, as they can remove a wide range of pollutants without producing secondary pollutants. This advantage addressed the limitations of conventional physicochemical and biological processes, which are further constrained by stringent environmental legislation (Liu et al., 2021). The first mentions of AOPs can be traced back to 1987 for the degradation of organic and non-biodegradable pollutants via the generation of hydroxyl radicals ( $HO \bullet$ ) in place of oxidation with reagents such as potassium permanganate, potassium dichromate and sodium persulfate (Khan et al., 2020). This was because oxidation of pollutants with these reagents only produced partially oxidized intermediates that served as a secondary source of pollution. Instead, the generations of radicals led to the complete mineralization of pollutants into simpler products, such as carbon dioxide and water. Hence, in recent years, various AOPs have been introduced, including Fenton-based, ozone-based, and photocatalytic-based AOPs. In general, all these processes follow the same fundamental steps in which they produce either one or more species of ROS, such as  $HO \bullet$  radical, superoxide radicals ( $O_2 \bullet$ ), peroxy radical ( $RO_2 \bullet$ ), and so forth (Kumari & Kumar, 2023).

### 2.1.1 Photocatalysis

Among AOPs, photocatalysis has attracted considerable attention as an effective and sustainable water treatment technology. In recent years, there has been increased research interest due to its ability to generate ROS in situ (mainly  $HO \bullet$  and  $O_2 \bullet$ ), while operating under relatively mild conditions compared with other conventional AOPs such as Fenton and ozone-based systems (Arifin et al., 2023). Fenton and ozone-based systems generally require strict pH control, continuous chemical input, and complex operational setups. In

photocatalytic systems, semiconductor materials are often employed to absorb photons with energy that is either equal to or greater than the energy band gap,  $E_g$ . This results in the excitation of electrons from the valence band (VB) to the conduction band (CB), forming electron-hole pairs (Abey et al., 2025). These charge carriers subsequently participate in surface redox reactions, thereby producing ROS. The combination of operational simplicity, the catalyst's reusability (due to the heterogeneous nature of the process), and the potential for complete mineralization has positioned photocatalysis as a promising and versatile AOP for the removal of persistent and emerging pollutants.

### 2.1.2 Photocatalytic reaction mechanism

Following the general principles of photocatalysis, the fundamentals of photocatalytic reaction mechanisms in water treatment involve a sequence of steps. The first step often involves charge generation, followed by transport and finally, the interfacial redox reactions of the photogenerated charge carriers on the surface of the catalyst (Chakravorty & Roy, 2024). The final step governs pollutant degradation reactions. Upon irradiation with incident photons, a semiconductor photocatalyst absorbs energy. This promotes electrons from the VB to the CB, resulting in the formation of electron-hole pairs. These charge carriers may undergo recombination or migrate to other catalyst surface, where they participate in simultaneous oxidation and reduction reactions. The photogenerated holes ( $h^+$ ) can oxidize surface-adsorbed water molecules and hydroxide ions to form  $HO \cdot$  radicals while excited electrons ( $e^-$ ) reduce dissolved oxygen to generate  $O_2 \cdot$  radicals. The ROS subsequently attacks organic pollutant molecules through a series of complex reactions, which ultimately end in the mineralization of these complex pollutants to harmless by-products such as  $CO_2$ ,  $H_2O$ , and inorganic ions. However, photocatalytic efficiency is strongly influenced by critical factors such as recombination rate, light absorption capacities, overall surface areas, and the availability of reactive sites on the photocatalyst surface (Bekele & Alamnie, 2025).

### 2.1.3 Application of photocatalyst in wastewater treatment

Metal oxide-based semiconductors are the most extensively investigated class of photocatalysts for wastewater treatment due to their chemical stability, ease of synthesis, and strong redox capability. The most widely studied metal oxides include zinc oxide (ZnO),

tungsten trioxide (WO<sub>3</sub>), iron oxide (Fe<sub>2</sub>O<sub>3</sub>), and titanium dioxide (TiO<sub>2</sub>), each exhibiting distinct band structures and photocatalytic behaviors, producing different treatment outcomes.

ZnO has been reported to exhibit photocatalytic activity comparable to that of TiO<sub>2</sub> under UV irradiation. For example, ZnO yielded degradation efficiencies of 99% for Tartrazine (TRZ), and 98% for naproxen (NAP) after 120 minutes of exposure under UV (Balu et al., 2019; Mohamed et al., 2023). However, its practical application is often limited by photocorrosion and dissolution. Another candidate photocatalyst, WO<sub>3</sub>, has a narrower band gap (approximately 2.4 eV) and therefore exhibits strong visible-light-driven activity. For instance, Manganese and Copper (Mn-Cu) doped WO<sub>3</sub> nanostructures achieved 86.7% degradation of methylene blue (MB) within 175 minutes and 75.9% of levofloxacin (LVF) within 120 minutes under visible light (Rizvi et al., 2024). Similarly, Ag-doped WO<sub>3</sub> composites have shown enhanced performance in degrading chloramphenicol (CLP) (Truong et al., 2025). Fe<sub>2</sub>O<sub>3</sub>, despite being abundant and absorbing visible light, its efficiency is often limited by rapid charge carrier recombination, though it can be improved through heterojunction formation (Hitam & Jalil, 2020).

Among the afore-mentioned semiconductors, TiO<sub>2</sub> remains as the benchmark material due to its high chemical stability, non-toxic nature and resistance to photocorrosion. Modified TiO<sub>2</sub> systems, such as black TiO<sub>2</sub>, have achieved 100% degradation of ciprofloxacin (CIP) under LED irradiation in 70 minutes (Samy et al., 2024). While TiO<sub>2</sub> primarily exhibits UV activity due to its wide band gap (3.2 eV), its consistent performance and environmental compatibility make it the reference material in research on heterogeneous photocatalytic wastewater remediation. A summary of various photocatalysts used for the degradation of organic pollutants is presented in Table 2.1.

**Table 2–1: Summary of selected metal oxide-based photocatalysts for the removal of organic pollutants**

Semiconductor	Dopant/Co-catalyst	Target Pollutant	Conditions	Removal Efficiency	Reference
TiO <sub>2</sub>	Nitrogen	CIP	<ul style="list-style-type: none"> <li>• Visible light</li> <li>• 70 minutes</li> </ul>	100.0%	Samy et al. (2024)
WO <sub>3</sub>	Mn-Cu	MB	<ul style="list-style-type: none"> <li>• Visible light</li> <li>• 175 minutes</li> <li>• pH 11</li> </ul>	86.7%	Rizvi et al. (2024)

WO <sub>3</sub>	Ag	CLP	<ul style="list-style-type: none"> <li>• Visible light</li> <li>• 120 minutes</li> </ul>	~80.0%	Truong et al. (2025)
ZnO	g-C <sub>3</sub> N <sub>4</sub> /α-Fe <sub>2</sub> O <sub>3</sub>	TRZ	<ul style="list-style-type: none"> <li>• Visible light</li> <li>• 120 minutes</li> <li>• pH 7 (Neutral)</li> </ul>	99.34%	Balu et al. (2019)

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ZnO	Pristine	NAP	<ul style="list-style-type: none"> <li>• UV light</li> <li>• 120 minutes</li> </ul>	98.7%	Mohamed et al. (2023)
Fe <sub>2</sub> O <sub>3</sub>	Carbon quantum dots	MB	<ul style="list-style-type: none"> <li>• Visible light (Xenon)</li> <li>• 180 minutes</li> </ul>	~90%	Hitam and Jalil (2020)

## 2.2 Titanium dioxide, TiO<sub>2</sub> as a photocatalyst

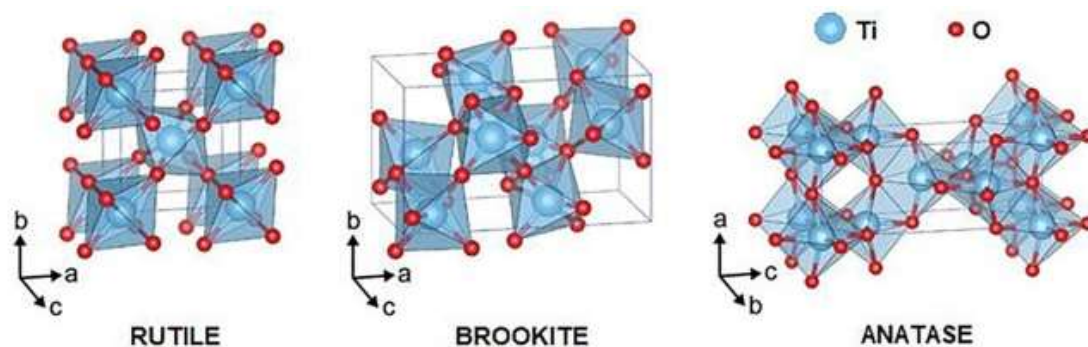
The adaptation of TiO<sub>2</sub> as a photocatalyst has been widely studied ever since the discovery of its water-splitting abilities under the irradiation of UV light by Fujishima and Honda (1972). Following this revelation, a study by Frank and Bard (1977) paved the way for the incorporation of TiO<sub>2</sub> in the photocatalytic oxidation of aqueous sulphite and cyanide ions. This study was one of the earliest mentions of the application of TiO<sub>2</sub> in water purification technology. Subsequently, further studies on the properties and mechanism of TiO<sub>2</sub> single crystals were established by Wang et al. (1999), which paved the path for numerous other studies with a variety of applications in photocatalysis. To date, various studies on the application of TiO<sub>2</sub> photocatalyst have been established, such as its application in water purification, cancer therapy, green energy generation, energy storage and so forth (Çeşmeli & Biray Avcı, 2019; Demir et al., 2019; R. Li et al., 2020; Singh et al., 2020). The versatility of TiO<sub>2</sub> in the aforementioned areas is attributed to the properties of TiO<sub>2</sub> being environmentally friendly, non-toxic, non-corrosive, highly stable, cost-effective and reusable (Qamar et al., 2023).

### 2.2.1 Crystalline properties of TiO<sub>2</sub>

There are three main polymorphs ascribed to TiO<sub>2</sub>, which are anatase, rutile and brookite. These crystalline phases of TiO<sub>2</sub> are widely available in nature and are caused by the distortion of the TiO<sub>2</sub> octahedral forms (Eddy et al., 2023). Based on Figure 2.1, rutile and brookite phase octahedrons exhibit a slight orthorhombic distortion, whereas in anatase, a more significant distortion is observed, making anatase less symmetrical than the other two variations (Chen & Mao, 2007). The Ti–Ti distances in anatase are larger, whereas the Ti–O distances are shorter than those in rutile. In the rutile structure, each octahedron is in contact with 10 neighbor octahedrons (two sharing edge oxygen pairs and eight sharing corner oxygen atoms), while in the anatase structure, each octahedron is in contact with eight neighbors (four sharing an edge and four sharing a corner). Differences in lattice structures

lead to distinct mass densities and electron band structures of the TiO<sub>2</sub> polymorphs (Janczarek et al., 2022).

**Figure 2–1: The three crystal structures of TiO<sub>2</sub> polymorphs Adapted [reprinted] from Haggerty et al. (2017).**

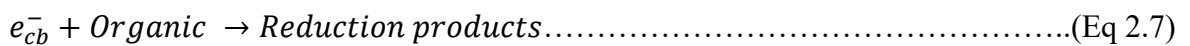
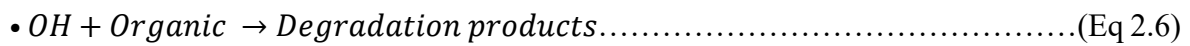
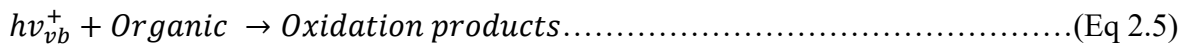
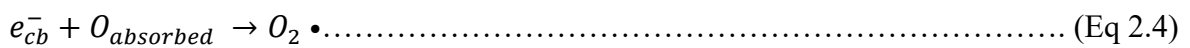
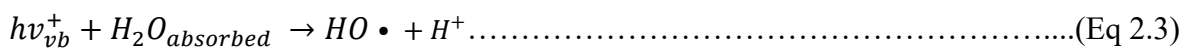
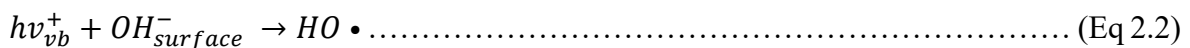
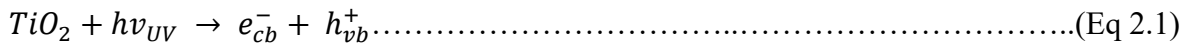


Nevertheless, in photocatalytic water-splitting applications, the rutile and anatase polymorphs of TiO<sub>2</sub> are the most extensively studied, as they exhibit greater photoactivity compared to the brookite phase (Eddy et al., 2023). The difference in their photocatalytic activity is mainly attributed to the difference in their band structure, mass density, and the effective mass of the photogenerated charge carriers (Zhang et al., 2014). Despite the rutile phase having a lower energy bandgap (3.0 eV) than the anatase phase (3.2 eV), past studies have shown that anatase has a smaller effective mass and a longer lifetime of photogenerated electron-hole (Luttrell et al., 2014; Yamakata & Vequzo, 2019; Žerjav et al., 2022). This is a result of having an indirect bandgap in comparison to the other two phases, which have a direct bandgap configuration. This leads to higher photocatalytic reduction activity of anatase TiO<sub>2</sub>. Depending on the goal of its application, the various phases can be well-suited for a variety of photocatalytic reactions.

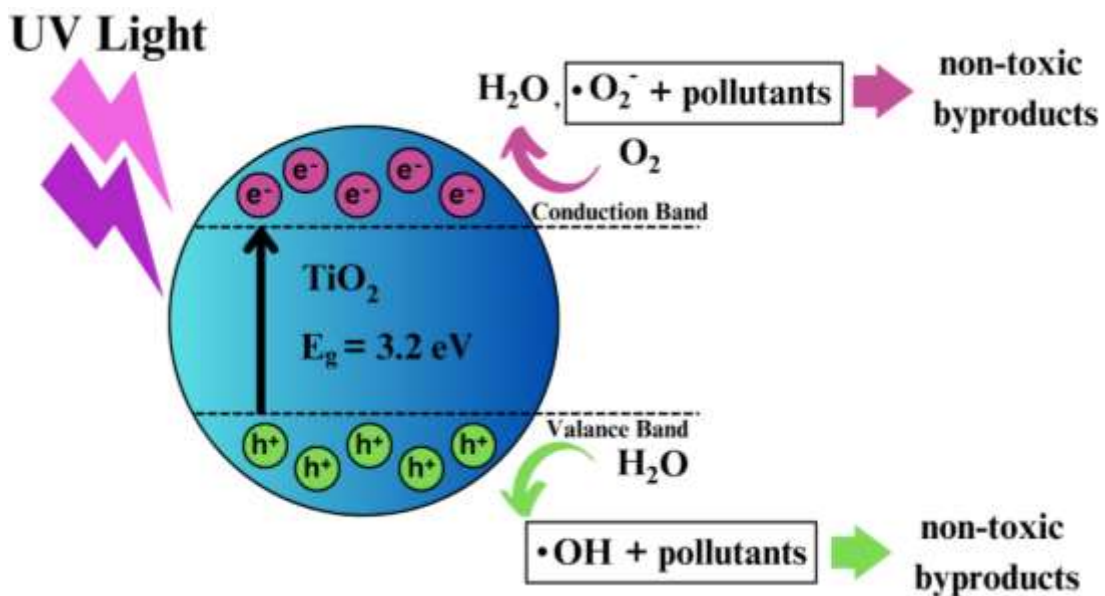
### 2.2.2 Application of TiO<sub>2</sub> in water purification

Given the versatility of the different phases of TiO<sub>2</sub> polymorphs in photocatalysis, the recent decade has witnessed an exponential increase in studies with regard to the water purification process (Armaković et al., 2023; Dharma et al., 2022; Magaña-López et al., 2021; Riaz & Park, 2020; Venkata Reddy et al., 2019; Wetchakun et al., 2019). Photocatalysis has emerged as a promising alternative to conventional approaches for removing pollutants from water, as pollutants can be transformed into non-toxic degradation products. Generally, the photocatalytic mechanism of TiO<sub>2</sub> begins with the generation of

electron-holes in the conduction band ( $e^-_{cb}$ ) and the valence band ( $h^+_{vb}$ ), upon irradiation of UV light, as shown in Figure 2.2 (Chen et al., 2020). According to Paumo et al. (2021), the formation of electron-holes will result in the formation of radicals (hydroxyl radicals,  $HO \bullet$  and superoxide radicals,  $O_2 \bullet^-$ ) through interactions with oxygen and water molecules that are absorbed onto the surface. These radicals will then react with the pollutants to degrade them into simpler and non-toxic byproducts such as carbon dioxide and water (Figure 2.2). The mechanisms involved in the  $TiO_2$  photocatalysis are shown in the equations (Eq) below:



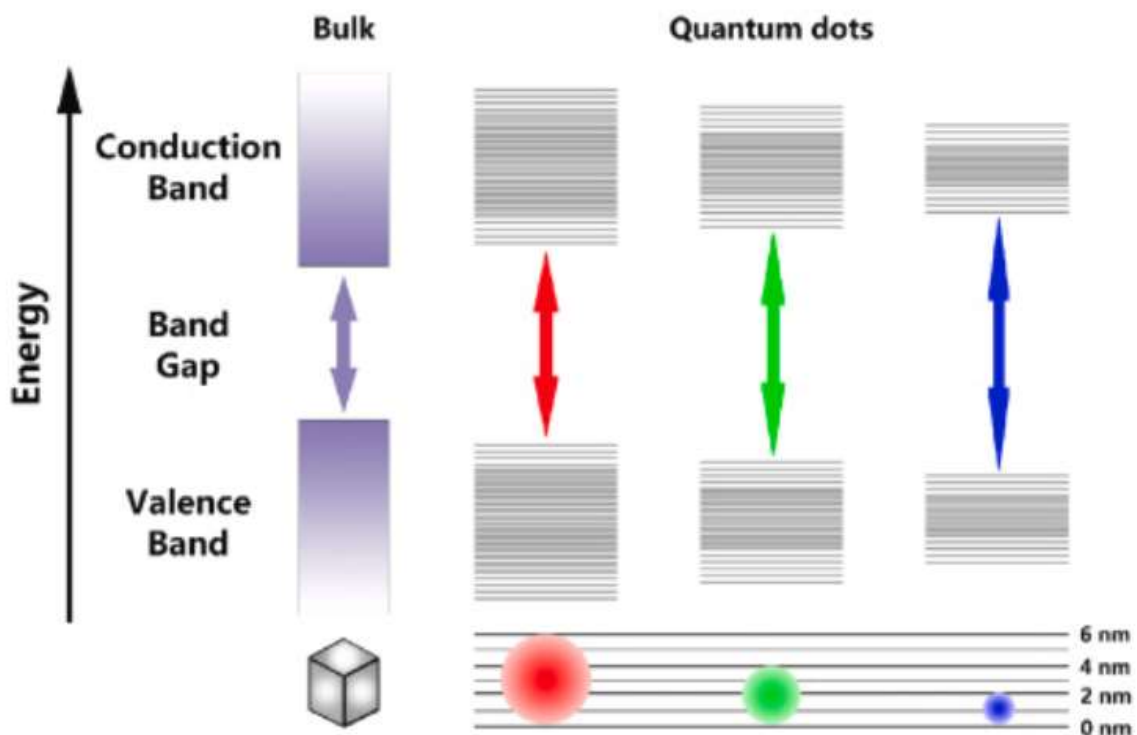
**Figure 2–2: Schematic illustration of photocatalytic activity in the  $TiO_2$  photocatalyst.**



### 2.2.3 Drawbacks of TiO<sub>2</sub>

Despite numerous desirable traits as a photocatalyst, the use of pure TiO<sub>2</sub> for light-induced water purification is hindered because its photocatalytic activity is limited to UV irradiation. Moreover, in most cases, TiO<sub>2</sub> used in photocatalytic water purification is in the nanoscale (Al-hamoud et al., 2022; Alkorbi et al., 2022; Ancy et al., 2022; Kim et al., 2021; Malakootian et al., 2020). For most nano-scaled semiconductors like TiO<sub>2</sub> NPs, the quantum size effect alters semiconductor properties due to the great difference in the size of the particle to the wavelength of the electron (Nunzi & De Angelis, 2022). The quantum size effect is simply the change in optical and electronic properties that is reflected in the energy levels, potential wells, valence bands, conduction bands, and electron energy bandgaps (Neikov & Yefimov, 2019) (Figure 2.3).

**Figure 2–3: Quantum size effect on semiconductor materials of different sizes**  
Adapted [reprinted] from Liu et al. (2022).



Bulk materials of semiconductor metals often exhibit a small bandgap due to defects that cause deep and shallow traps near the band edge of the electronic states (Lin et al., 2006). This is caused by the mild delocalization of molecular orbitals, which, in turn, results in a red shift in the absorbance spectrum (Lin et al., 2006). However, when the size of these

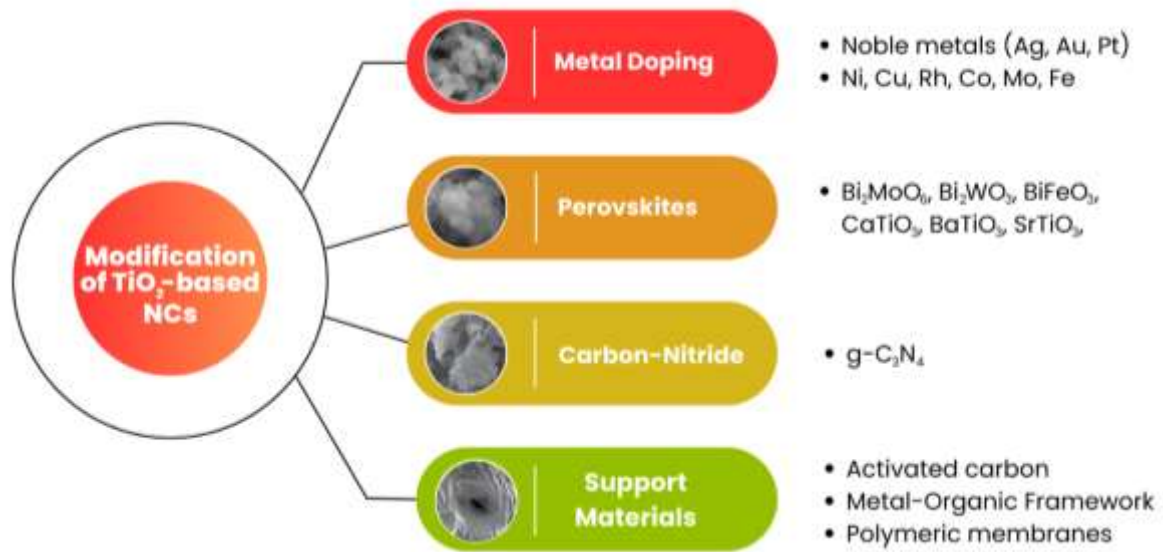
materials is within the nanoscale (<30 nm), quantum size effects have been reported to take place as electrons and holes in the quantum-sized semiconductors are spatially confined in a potential well, which in turn induces a blue shift in the absorbance spectrum (Lin et al., 2006). This phenomenon demonstrates that pure nano-TiO<sub>2</sub> exhibits strong UV absorbance due to the high-energy bands of both rutile and anatase phases, which are 3.0 eV and 3.2 eV, respectively. This absorbance is also dependent on the size of nano-TiO<sub>2</sub> particles. For instance, Jiménez Reinoso et al. (2016), found that as the size of TiO<sub>2</sub> decreases from the microscale to the nanoscale, the bandgap increases from 3.15 eV to 3.37 eV, resulting in a blue shift (391 nm to 357 nm) in the UV absorption edge. Another study by D. Li et al. (2020) also revealed that the absorbance edge of pure nano-TiO<sub>2</sub> with anatase crystallinity showed strong absorption within the UV region at 400 nm. These studies suggest that the particle size of TiO<sub>2</sub> influences the absorption edge; therefore, nano-TiO<sub>2</sub> needs to be modified to enhance its photocatalytic ability, as UV accounts for only 5% of solar radiation.

### **2.3 Current strategies to improve TiO<sub>2</sub>'s photocatalytic performance**

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Various strategies have been developed to further improve the photocatalytic activity of TiO<sub>2</sub>, with current trends toward the functionalization of TiO<sub>2</sub>-based NCs for visible-light and near-infrared regions. This is because in the overall solar spectrum, visible light and infrared radiation account for at least 42% and 49% of the total incident rays, respectively (Bano et al., 2021). This can be achieved by the fabrication of TiO<sub>2</sub> with other materials such as metals, perovskites and carbon-nitride materials. Additionally, in recent years, researchers have shown growing interest in the recyclability, stability, and recovery of TiO<sub>2</sub>-based NCs. Consequently, numerous studies have examined how different support materials influence stability. An illustration of the current strategies employed in modifying TiO<sub>2</sub>-based NCs is shown in Figure 2.4.

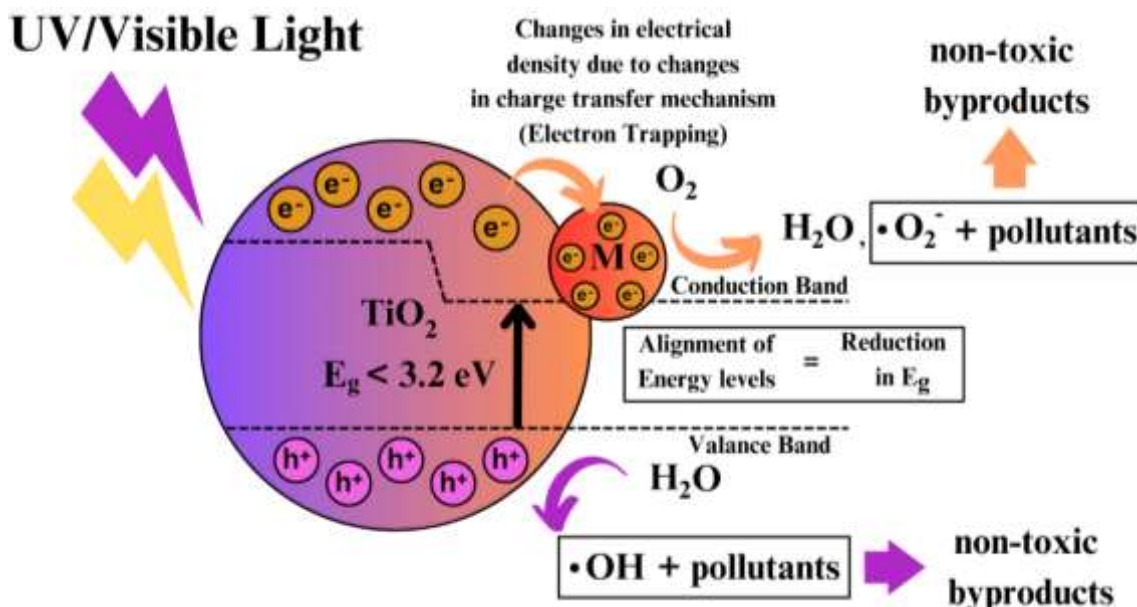
**Figure 2–4: Strategies employed to improve the functionalization and stability of TiO<sub>2</sub>-based nanocomposites under a wider range of light irradiation.**



### 2.3.1 Metal Doping

In general, doping of TiO<sub>2</sub>-based NCs has been done in two ways, namely with anionic non-metal elements (e.g. carbon, nitrogen, sulphur, fluorine) and cationic dopants (e.g. transition metals, noble metals, rare earth elements). Metal doping has been widely studied due to the ability of metal dopants to act as an electron trap to reduce electron-hole recombination, introduce new energy levels in the bandgap structure of TiO<sub>2</sub>, induce SPR effects and create new gap states through their interaction with the VB states of TiO<sub>2</sub> (Kumaravel et al., 2019). Figure 2.5 presents a schematic illustration depicting the effects of metal doping on TiO<sub>2</sub>-based NCs.

**Figure 2–5: Schematic illustration of the photocatalytic mechanism involved in a metal-doped TiO<sub>2</sub> photocatalyst.**



Amongst a myriad of metals explored for enhancing the photocatalytic ability of TiO<sub>2</sub>, the most widely studied dopants namely Au, Ag, platinum (Pt), palladium (Pd), and copper (Cu). The addition of metal dopants is often aimed at enhancing and prolonging the lifetime of the charge carrier. Therefore, the selection of metal dopants is important as the photoactivity of TiO<sub>2</sub>-based photocatalysts that are modified with metal dopants varies depending on how well the overall photocatalytic working system functions, despite the aforementioned enhancements on the photocatalyst. This is because the synergistic effects of metal dopants rely on whether it is an active or inactive dopant in the different crystalline phases of TiO<sub>2</sub>.

Cu is a popular choice among metal dopants because it is abundant as an earth element, making it a highly cost-effective option. Albornoz Marin et al. (2022) synthesized core-shell Cu@TiO<sub>2</sub> through the wet impregnation technique as a means of introducing Cu<sup>2+</sup> ions into the crystal lattice of TiO<sub>2</sub> for the degradation of phenol. The incorporation of Cu into TiO<sub>2</sub> resulted in a photocatalyst with reduced crystalline size, increased lattice strain and a decrease in the bandgap energy,  $E_g$  from 3.01 eV to 2.67 eV for 3.0% Cu. Although the degradation rate under UV irradiation was not particularly high, it doubled under visible light due to the reduced bandgap and the SPR effects of Cu. A similar finding was reported by Pascariu et al. (2022), where Cu/TiO<sub>2</sub> composite nanofibers with Cu-doped onto TiO<sub>2</sub>, achieved nearly complete degradation of 25 ppm amaranth dye, reaching a degradation rate of 99.8% under both UV and visible light. Notably, the crystalline phase of TiO<sub>2</sub> differed

between the two studies whereby Alborno Marin et al. (2022) used a mixed anatase/rutile phase TiO<sub>2</sub>, while Pascariu et al. (2022) used a pure anatase phase TiO<sub>2</sub>. The differences between these two studies highlight the importance of selecting an appropriate metal dopant, depending on the TiO<sub>2</sub> crystalline phase, to achieve optimal photocatalytic degradation properties.

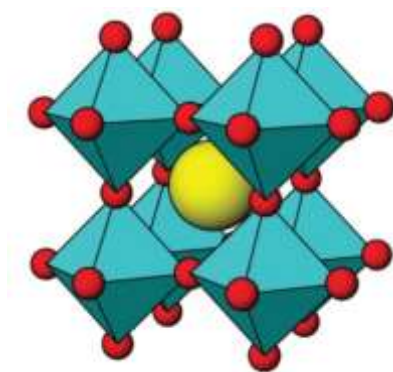
Apart from Cu, various studies on noble metal doping of TiO<sub>2</sub>-based photocatalysts have been conducted. Noble metals such as Ag, Au, Pt and Pd have been shown to induce desirable traits in comparison to other metals due to the conductive properties of these metals. Nevertheless, the application of Au, Pt and Pd in modifying TiO<sub>2</sub>-based photocatalysts is difficult as these elements are expensive and difficult to source out despite being superior to Ag as a dopant. However, Ag excels as a dopant compared to other noble metals. It has superior conductivity, can induce SPR effects similar to those of other metals, is significantly more abundant than the other noble metals, and is well known for its antibacterial properties (Raes et al., 2024). For instance, Gomes et al. (2018) studied the effects of Ag, Pd and Pt loading on the efficiency of TiO<sub>2</sub> for the photocatalytic ozonation of parabens. It was observed that at higher loading, Pd and Pt outperformed Ag in degrading parabens, whereas at lower concentrations, Ag exhibited superior photocatalytic activity. This study highlights the economic advantage of Ag compared to other noble metal dopants with photocatalytic enhancements of TiO<sub>2</sub> that are close to those of the others. However, it is also worth noting that maximizing the performance of Ag-doped TiO<sub>2</sub> photocatalytic materials requires selecting suitable TiO<sub>2</sub> crystalline phases, as Ag exhibits varying synergistic effects depending on whether it is incorporated into pure or mixed phases, as demonstrated in previous studies (Kusdianto et al., 2018).

### **2.3.2 Perovskites**

In recent years, there has been an increase in the fabrication of TiO<sub>2</sub> with co-catalysts such as perovskites. Perovskites are considered as third-generation photocatalysts due to their unique physicochemical characteristics that include high chemical stability, high thermal stability and excellent electronic conductivity. The general chemical formula of perovskite is ABX<sub>3</sub> in which A, B and X sites stand for alkaline-earth/rare-earth cations with large atomic radii such as bismuth (Bi), transition metal cations such as molybdenum (Mo) with smaller atomic radii, and anions, namely, oxygen (O), sulphur (S), nitrogen (N), and

halides (Wang et al., 2021). The formation of perovskites happens when cations A form a 12-fold coordination at the centre of the cube of B cations that are combined with 6 anions X to form  $BX_6$  coordination in an octahedral molecular geometry (Figure 2.6). The differences in the valences and atomic radii of these three components generate distortion in the structure that induces characteristics such as oxygen vacancies, active redox sites, narrow energy bandgap, visible light photocatalytic activity and superior electrical conductivity and thermal stability (Kong et al., 2019; Kumar et al., 2020).

**Figure 2–6: An ideal  $ABX_3$  structure of a cubic perovskite crystal Adapted [reprinted] from Kong et al. (2019).**

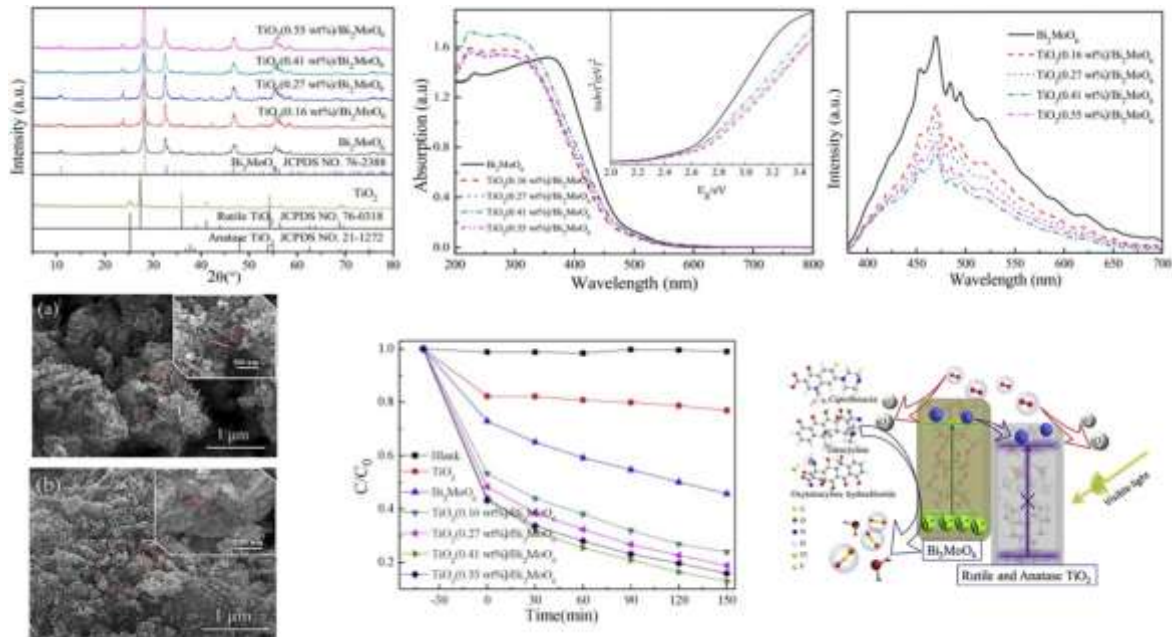


Some of the most common phases in the perovskite family that have been currently studied include the Dion-Jacobson phase, Ruddlesden-Popper phase and Aurivillius phase (Kubacka et al., 2012). However, amongst all the available phases in the perovskite families, the Aurivillius phase is considered the class that has been shown to have the most significant results in terms of its photocatalytic performance (Collu et al., 2022; Guo et al., 2019; Malik et al., 2021; Yao et al., 2024). Given the exceptional properties of perovskite materials as photocatalysts, recent years have seen the use of various perovskite types as co-catalysts to enhance  $TiO_2$ -based photocatalysts under visible light (Fatima et al., 2024).

A study by Liu et al. (2019) successfully fabricated  $Bi_2MoO_6/TiO_2$  via a facile solvothermal-calcination method for the photodegradation of ciprofloxacin (Figure 2.7). A binary anatase/rutile  $TiO_2$  with a rod-like shape, and with nano-plate-like  $Bi_2MoO_6$  was produced. The photocatalyst also exhibited a reduced bandgap of  $\sim 2.60$  eV, along with improved charge-carrier dynamics and reduced electron-hole recombination. The photocatalyst exhibited exceptional performance in degrading ciprofloxacin, achieving nearly 60.0% dark adsorption and close to 99.0% photodegradation under visible light.

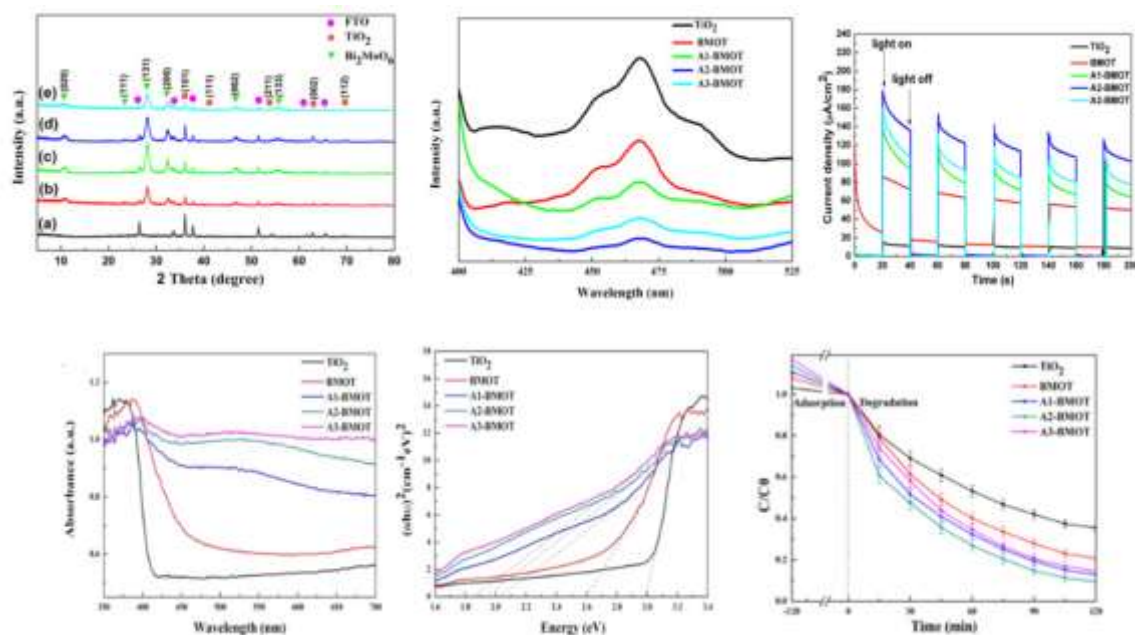
**Figure 2–7: Characterizations and photocatalytic performance of Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>**

Adapted [reprinted] from Liu et al. (2019).



Furthermore, another study by Yin et al. (2021) fabricated Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> with Ag quantum dots (QDs) for the degradation of tetracycline hydrochloride using a multi-step solvothermal-wet impregnation method. In comparison to Liu et al. (2019), the morphology of the synthesized AgQD/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> was controlled by using fluorine-doped tin oxide glass as a template for the synthesis of the photocatalyst. Their method produced single-phase TiO<sub>2</sub> rutile nanorods, a Bi<sub>2</sub>MoO<sub>6</sub> nanosheet photocatalyst, and Ag QDs with sizes less than 1 nm. The ability of the AgQD/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> to degrade tetracycline hydrochloride with an efficiency of 90.8% can be attributed to characteristics reported by Liu et al. (2019). These included a reduction in the energy bandgap, electron-hole recombination rate and enhanced charge separation dynamics with a longer lifetime (Figure 2.8).

**Figure 2–8: Characterizations and photocatalytic performance of AgQD/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> Adapted [reprinted] from Yin et al. (2021).**



In summary, the ability of perovskite-based material such as Bi<sub>2</sub>MoO<sub>6</sub> is shown to be able to reduce the energy bandgap of TiO<sub>2</sub> and further improve its charge dynamics as a result of the different bandgap structures. This shows the potential of fabricating TiO<sub>2</sub> with a perovskite co-catalyst to broaden and enhance the applications of TiO<sub>2</sub>-based photocatalytic nanomaterials in water treatment.

### 2.3.3 Carbon Nitrides

Besides perovskite-based materials, co-catalyst, g-C<sub>3</sub>N<sub>4</sub> is another widely studied candidate for the fabrication of TiO<sub>2</sub>-based photocatalytic materials. Layered g-C<sub>3</sub>N<sub>4</sub>, as another conventional semiconductor, has two building blocks (triazine [C<sub>3</sub>N<sub>3</sub>] and tri-s-triazine/heptazine [C<sub>6</sub>N<sub>7</sub>]) rings, homogeneously distributed in the g-C<sub>3</sub>N<sub>4</sub> microstructure. The most commonly used method for preparing bulk g-C<sub>3</sub>N<sub>4</sub> is the thermal polymerization of nitrogen-rich organic precursors (including dicyandiamide, urea, and melamine) at relatively high temperatures (400–700 °C) (Zhang & Jiang, 2022). g-C<sub>3</sub>N<sub>4</sub> is chemically and thermally stable (up to 600 °C in air) owing to the presence of aromatic C–N heterocycles in the material's structure, and it cannot be dissolved in organic or acid/alkali solvents under ambient conditions (Martin et al., 2014; Teixeira et al., 2018; Wang et al., 2021). Thus, g-C<sub>3</sub>N<sub>4</sub> has been widely studied as a metal-free semiconducting photocatalyst

for a variety of applications over the past decades (Huang et al., 2019). Moreover, g-C<sub>3</sub>N<sub>4</sub> and g-C<sub>3</sub>N<sub>4</sub>-based materials are considered to be chemically active only in the case of photogenerated charge pairs being consumed before the recombination takes place (Ong et al., 2016).

As one of the most stable carbon nitride allotropes, g-C<sub>3</sub>N<sub>4</sub> has been extensively investigated in visible-light-driven photocatalysis-related fields. Despite its various advantageous properties, g-C<sub>3</sub>N<sub>4</sub> is nevertheless still not an ideal photocatalyst. This is due to its irregular morphology (which attenuates charge transfer), low specific surface area (less than 20 m<sup>2</sup>/g), and partially blocked reactive sites of the close-packed system (Huang et al., 2019; Ong et al., 2016).

Therefore, studies have been conducted to improve the performance of g-C<sub>3</sub>N<sub>4</sub> by combining it with TiO<sub>2</sub>, as smaller TiO<sub>2</sub> particles can further increase the composite's total surface area. Given the narrow bandgap of g-C<sub>3</sub>N<sub>4</sub> (~2.7 eV), the synergistic effects of the staggered band positions between TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub> can enable functionalization of g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> under visible light, with enhanced performance compared to their individual counterparts. For instance, a study by Kanakaraju et al. (2024) synthesized TiO<sub>2</sub>/ZnS/g-C<sub>3</sub>N<sub>4</sub> photocatalyst through a facile hydrothermal process. The study revealed that the photocatalyst showed enhanced photodegradation of multiple organic pollutants, with RhB having the highest degradation rate of 90%. The synthesized TiO<sub>2</sub>/ZnS/g-C<sub>3</sub>N<sub>4</sub> showed a remarkable ability to degrade mixed pollutants owing to the enhanced optoelectrical and structural properties of the photocatalyst. The fabrication of g-C<sub>3</sub>N<sub>4</sub> also proved the ability of TiO<sub>2</sub> to increase the surface area of the overall composite which allows for better contact with pollutants, thus enhancing the performance of the overall composite. Other studies have also reported similar results (Gahlot et al., 2021; Kocijan et al., 2022; Monga & Basu, 2019; Sutar et al., 2020).

This shows the potential of fabricating Ag/TiO<sub>2</sub>-based nanocomposite with g-C<sub>3</sub>N<sub>4</sub> to further enhance its functionality under visible light. Combining the SPR effects of Ag and the narrow energy bandgap of g-C<sub>3</sub>N<sub>4</sub> can result in a superior photocatalyst that complements each weakness of its singular components to form a high-performing Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> photocatalyst under a wider spectrum of light sources.

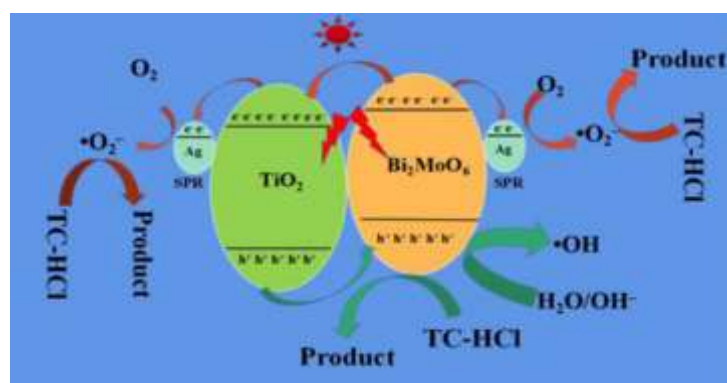
## 2.4 Fabrication of TiO<sub>2</sub> with Ag, Bi<sub>2</sub>MoO<sub>6</sub> and g-C<sub>3</sub>N<sub>4</sub>

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It has been proven that the fabrication of TiO<sub>2</sub> through metal doping or combination with co-catalysts such as perovskites and g-C<sub>3</sub>N<sub>4</sub> has been shown to enhance the photocatalytic properties of TiO<sub>2</sub> as a photocatalyst. Noble metals such as Ag can be used for the modification of TiO<sub>2</sub>-based photocatalysts. By depositing nano-Ag on the surface of TiO<sub>2</sub>, the photocatalytic ability of TiO<sub>2</sub> can be enhanced as Ag would act as an electron trap or charge carrier due to the Schottky barrier between Ag and TiO<sub>2</sub> (Chakhtouna et al., 2021). This is possible as the Fermi energy level of Ag influences the charge separation between the interface of the TiO<sub>2</sub> and Ag dopant (Kanakaraju et al., 2022). Given that the Fermi energy level of Ag is located below that of TiO<sub>2</sub>, this causes the bandgap of TiO<sub>2</sub> to decrease, hence improving charge separation and preventing electron-hole recombination in the TiO<sub>2</sub> crystal lattice.

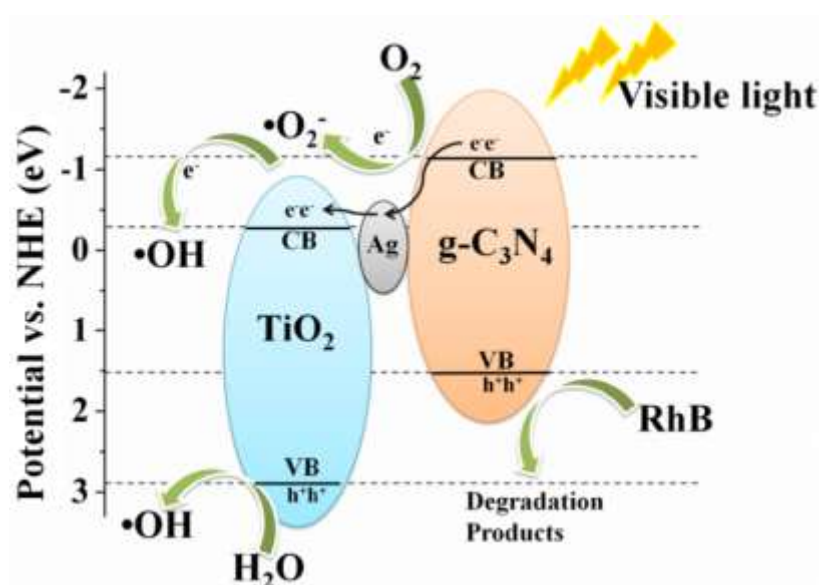
Moreover, as previously mentioned, due to the excellent SPR of Ag, hot electrons upon excitation by UV or visible light are injected via the Schottky junction into the conduction band of TiO<sub>2</sub> (Ding et al., 2014). The hetero-interaction between TiO<sub>2</sub> and nano-Ag, serving as a co-catalyst in the overall mechanism, enhances the reactive sites of the Ag-TiO<sub>2</sub> photocatalyst, leading to improved charge separation and efficient electron transport between Ag and TiO<sub>2</sub> (Shoaib et al., 2016). This is mainly due to the excess negative charge in Ag and the excess positive charge in TiO<sub>2</sub> as a result of the Schottky barrier formed at the heterojunction of the Ag-TiO<sub>2</sub> surface. Additionally, fabrication with bismuth molybdate (Bi<sub>2</sub>MoO<sub>6</sub>) could provide additional benefits to the ternary structure of Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>. Given that Bi<sub>2</sub>MoO<sub>6</sub> have a narrow bandgap with a wide range of light absorption, this allows for an alignment of the Fermi energy levels of TiO<sub>2</sub>, which thus enables the reduction of the energy bandgap of the overall ternary composites (Figure 2.9). This allows for a more efficient charge transfer mechanism as well as better visible light response of the photocatalyst.

**Figure 2–9: Proposed photocatalytic degradation of tetracycline hydrochloride by Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> under visible light irradiation Adapted [reprinted] from Yin et al. (2021)**



Similar to fabrication with Bi<sub>2</sub>MoO<sub>6</sub>, combining TiO<sub>2</sub> with g-C<sub>3</sub>N<sub>4</sub> can also form an effective heterojunction, enabling improved charge transfer dynamics, introducing additional energy levels, and enhancing the visible light functionality of TiO<sub>2</sub> (Acharya & Parida, 2020) (Figure 2.10). Moreover, given that the downside of g-C<sub>3</sub>N<sub>4</sub> is its low surface area, fabrication with TiO<sub>2</sub> could also result in a NC with a higher relative surface area which can allow for more contact with pollutants, thus enhancing adsorption capacity and increasing the number of active sites for photocatalysis to occur.

**Figure 2–10: Proposed photocatalytic degradation of rhodamine B by Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> under visible light irradiation Adapted [reprinted] from Zhou et al. (2019).**



In general, the SPR of Ag and the narrow bandgap of Bi<sub>2</sub>MoO<sub>6</sub> and g-C<sub>3</sub>N<sub>4</sub> will help enhance the photocatalytic ability of TiO<sub>2</sub> in the visible light region whilst reducing the recombination rate of photogenerated electron holes by providing an alternate pathway for electron transfer within the NC. In addition to this, given the distinctive morphology of both co-catalyst Bi<sub>2</sub>MoO<sub>6</sub> and g-C<sub>3</sub>N<sub>4</sub>, this can allow for an increase in the surface area and provide more active sites for photocatalysis to occur on the surface of the photocatalyst.

## 2.5 Method of fabricating Ag/TiO<sub>2</sub>-based NCs

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The effective implementation of enhancement strategies, such as noble metal deposition, heterojunction engineering, and coupling with visible-light-responsive semiconductors, depends strongly on the selected fabrication method. In Ag/TiO<sub>2</sub>-based NCs, the synthesis route plays a critical role in governing particle size distribution, crystallinity, and interfacial contact, which are essential for harnessing the SPR effect of Ag and promoting efficient charge separation within TiO<sub>2</sub>. Consequently, various chemical and physical approaches have been developed to construct Ag/TiO<sub>2</sub>-based binary and ternary systems, each imparting distinct structural features and photocatalytic performances.

### 2.5.1 Sol-gel synthesis

The sol-gel method is one of the most established approaches for producing Ag/TiO<sub>2</sub>-based NCs, as it enables high purity and molecular-level homogeneity (Chen et al., 2025). This method involves transitioning a system from a liquid sol to a solid gel phase, typically via hydrolysis and polycondensation of metal alkoxides or inorganic salt precursors (Sadek et al., 2022; Utomo et al., 2024). In the context of Ag/TiO<sub>2</sub> fabrication, the sol-gel process enables the uniform distribution of Ag within the TiO<sub>2</sub> matrix. For instance, a previous study used sol-gel techniques combined with chemical deposition to fabricate a NiO/Ag-TiO<sub>2</sub> NC electrode (Ravishankar et al., 2022). In this study, the sol-gel method enabled the formation of uniform TiO<sub>2</sub>, facilitating the in-situ deposition of Ag and NiO within the matrix in a two-step synthesis process. This method was particularly effective for creating multi-component systems with high surface area and stable crystalline phases, which are essential for producing an efficient photocatalyst for water remediation.

### 2.5.2 Hydrothermal/Solvothermal synthesis

Hydrothermal and solvothermal methods are highly regarded for their ability to produce well-crystallized Ag/TiO<sub>2</sub>-based heterostructures in a single step, without requiring further calcination. This is due to the high pressure and temperature in a sealed autoclave synthesis environment (Feng & Yam, 2024a). Unlike ambient methods, hydrothermal and solvothermal routes can facilitate and control the growth of specific crystal facets such as nanorods, nanoplates, and nanosheets. These unique morphological characteristics are often linked to enhanced charge carrier properties due to the presence of higher active sites for photocatalysis reactions to take place (Feng & Yam, 2024b). In a past study, the hydrothermal route was employed to fabricate a complex ternary system of g-C<sub>3</sub>N<sub>4</sub>/Ag-TiO<sub>2</sub> NCs (Sewnet et al., 2023). This method was deemed suitable for combining the materials, as it allows interfacial contact between TiO<sub>2</sub> and promotes Ag nucleation for deposition, whilst simultaneously promoting exfoliation of g-C<sub>3</sub>N<sub>4</sub> to form nanosheets. Additionally, changing solvent environments in solvothermal processes can promote fine-tuning of the energy band gap and induce oxygen vacancies, which boost visible-light response and increase the number of active sites in NCs (Jiang et al., 2024).

### 2.5.3 Microwave-assisted synthesis

Microwave-assisted synthesis has recently emerged as a rapid and efficient alternative to conventional heating methods. It is highly compatible with most conventional methods such as chemical deposition and sol-gel synthesis and allows for a significant reduction in reaction time and energy consumption. In this approach, microwave irradiation is used to introduce TiO<sub>2</sub> and Ag into precursor solutions, enabling the facile in situ fabrication of ternary ZnO/TiO<sub>2</sub>/Ag NCs (Mohan et al., 2021). This method is often paired with green synthesis techniques, such as using *Morinda citrifolia* fruit extract as a stabilizing or reducing agent, to produce biocompatible and bactericidal NCs. The localized and rapid heating provided by microwaves helps in achieving uniform distribution of AgNPs across the TiO<sub>2</sub> matrix, which is critical for enhancing the material's photocatalytic and antibacterial properties. The compatibility of microwave-assisted synthesis with bio-derived reducing and stabilizing agents has accelerated the shift towards sustainable nanomaterial fabrication. As a result, green chemistry has recently emerged as a viable alternative to conventional

chemical synthesis by leveraging naturally occurring biomolecules to synthesize nanomaterials (Kirubakaran et al., 2026).

## 2.6 Green synthesis of Ag/TiO<sub>2</sub>-based nanocomposite

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Recently, green chemistry has opened an alternative pathway to standard chemical synthesis by utilizing naturally occurring biomolecules for the synthesis of nanomaterials in a method called green synthesis (Devatha & Thalla, 2018). Nature provides a wide range of naturally occurring biomolecules that can be used in very diverse ways. These biomolecules, found in microorganisms and plants, are used to drive the synthesis of nanomaterials (Rana et al., 2020). However, in most cases, plant extract-mediated green synthesis reactions have been more frequently utilized than microorganism-based green synthesis. This is because plants are easier to use, readily available, and have a higher potential in the mass production of nanomaterials due to their fast reaction (Guerra et al., 2018). In addition, plant extract-mediated green synthesis provides precise control over the production of high-yield nanomaterials with well-defined sizes and varying morphologies in a one-pot reaction (Zhu et al., 2019).

Various studies have shown the successful synthesis of Ag-TiO<sub>2</sub> NCs with well-defined sizes and morphologies. For instance, green synthesized Ag-TiO<sub>2</sub> by Jayapriya and Arulmozhi (2021) using *Beta vulgaris* peel extract had spherical morphology with a reduced size of about 40 nm as observed by SEM and TEM. Another study by Nguyen et al. (2022) reported that green synthesized Ag-TiO<sub>2</sub> via *Cleistocalyx operculatus* leaf extract also demonstrated spherical surface morphology with sizes ranging from 20 nm to 40 nm. Various other studies also revealed similar surface morphology of spherical Ag NPs covering the surface of TiO<sub>2</sub> with promising application in the photodegradation of organic pollutants (Jiang et al., 2019; Kumar et al., 2016; Mohapatra et al., 2021; Nethravathi et al., 2022; Saeed et al., 2019; Tavakoli et al., 2019). This suggests that green synthesis via plant extracts can form Ag-TiO<sub>2</sub> NCs with controlled morphology. A compilation of green-synthesized Ag-TiO<sub>2</sub> NCs can be seen in Table 2.2.

**Table 2–2: Summary of selected studies of TiO<sub>2</sub>-based nanocomposite synthesized via green synthesis for the photocatalytic degradation of organic pollutants**

Type of plant extract	Photocatalyst	Morphology and size	Organic pollutant	Major findings	Reference
Beta vulagris peel	Ag-TiO <sub>2</sub>	Spherical 25 ± 74 nm	Methylene Blue, Congo Red, Methyl Orange	Degradation efficiency for removal of Methylene Blue (92% in 9 min), Congo Red (84% in 20 min) and Methyl Orange (88% in 10 min).	Jayapriya and Arulmozhi (2021)
Cleistocalyx operculatus leaf	Ag-TiO <sub>2</sub>	Spherical 20 – 40 nm	Rhodamine B	Up to 91.4% of Rhodamine B was successfully removed within 180 min.	Nguyen et al. (2022)
<i>Mangifera indica</i> leaf	Ag-TiO <sub>2</sub>	Spherical	Methylene Blue	68% photodegradation was achieved for 7.81 × 10 <sup>-5</sup> mol/L initial MB concentration at pH 8 by using 0.19 g/L photocatalyst	Mohapatra et al. (2021)
<i>Cucumis melo</i> juice	Ag-TiO <sub>2</sub>	Spherical 9.12 nm	Methylene Blue	95.3% of Methylene Blue was degraded in 120 min under irradiation of visible light.	Nethravathi et al. (2022)
<i>Azadirachta indica</i> leaf	Ag-TiO <sub>2</sub>	Spherical	Methylene Blue, Rhodamine B	More than 90% of Methylene Blue and Rhodamine B dyes were degraded in 120 min	Saeed et al. (2019)
<i>Pranus Cerasus</i> seed	Ag-TiO <sub>2</sub> -G	Spherical Ag-TiO <sub>2</sub> distributed onto graphene sheets 10-20 nm	Acid Orange 7 (AO7)	Graphene with 35 wt% Ag exhibits 99% AO7 degradation within 20 min	Tavakoli et al. (2019)

				but only 61% of degradation for bare TiO <sub>2</sub>	
<i>Cinnamomum camphora</i> leaf	Au-Ag-TiO <sub>2</sub>	Spherical 12.6 ± 1.7 nm	Methyl Orange	Degradation rate of Au-Ag-TiO <sub>2</sub> reached ~90% within 30 min of UV-vis light irradiation but pure-TiO <sub>2</sub> only had 50% after 60 min	Jiang et al. (2019)
<i>Nephelium lappaceum</i> L. peel	Ag-TiO <sub>2</sub>	Spherical 100-180 nm	Methylene Blue	>81% degradation rate of Methylene Blue in 600 min under sunlight for Ag-TiO <sub>2</sub> .	Kumar et al. (2016)

Therefore, despite the prevalence of reports on the successful green synthesis of binary TiO<sub>2</sub>-based NCs, the synthesis of ternary NCs using the same methods remains feasible. This potential is supported by the binding properties of naturally occurring compounds in plant-based extracts. These compounds are also prevalent in palm oil mill waste, and it could prove as a possible source to obtain extracts that are also rich in phytochemicals for the green synthesis of nanomaterials.

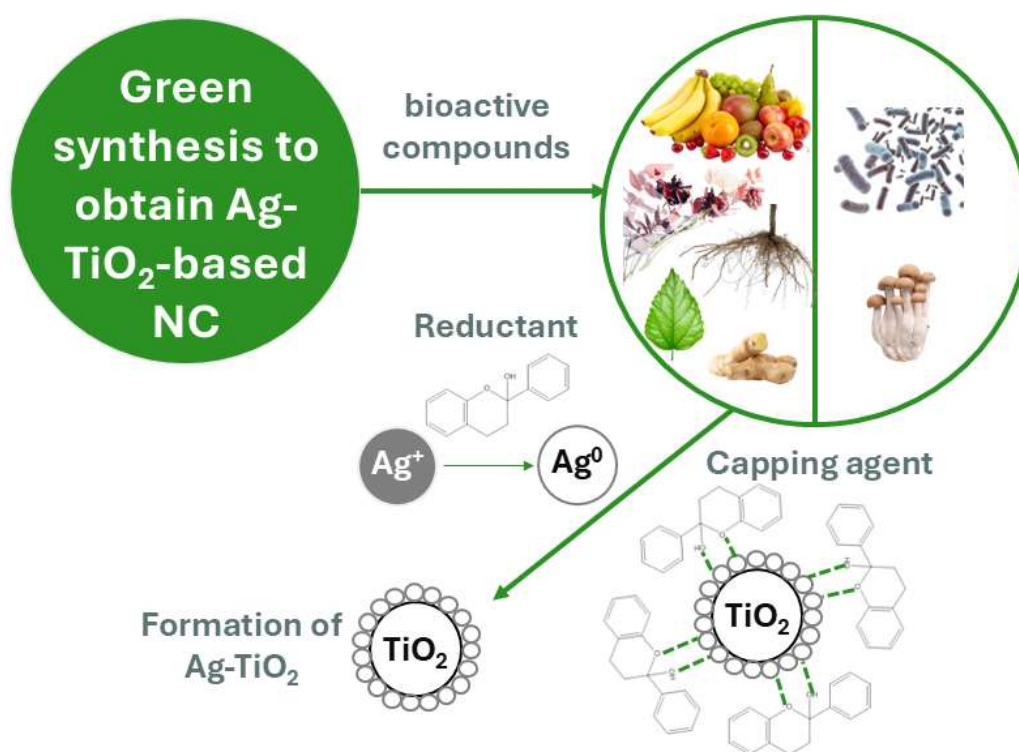
### 2.6.1 Principles and mechanisms of green synthesis

Green synthesis of nanomaterials is fundamentally governed by redox reactions mediated by naturally occurring biomolecules, offering an environmentally benign alternative to conventional chemical synthesis routes. In this approach, bio-derived compounds function as reducing agents, stabilizers, and, in some cases, capping agents, thereby eliminating the need for toxic chemicals and energy-intensive processes (Zhu et al., 2019). The general mechanism of green synthesis involves three main stages: the reduction of metal precursor ions, the nucleation and growth of nanoparticles, and the stabilization of the resulting nanostructures to prevent agglomeration.

In plant extract-mediated green synthesis, a diverse mixture of phytochemicals including polyphenols, flavonoids, polysaccharides, terpenoids, alkaloids, proteins, and

organic acids, collectively drives these processes (Bawazeer et al., 2021). These biomolecules donate electrons to metal ions such as  $\text{Ag}^+$  ions, facilitating their reduction to metallic nanoparticles, while functional groups in the phytochemicals adsorb onto the nanomaterial surface, providing steric or electrostatic stabilization. Figure 2.11 illustrates the general mechanism involved in the green synthesis of  $\text{Ag}/\text{TiO}_2$ -based NCs. The process is relatively simple and does not require the separation or purification of individual bioactive compounds prior to synthesis, making it attractive for scalable and sustainable nanomaterial production.

**Figure 2–11: General mechanism involved in the green synthesis of  $\text{Ag}/\text{TiO}_2$ -based NCs**



Despite these advantages, the heterogeneous nature of plant extracts introduces variability in reaction kinetics, nanoparticle size, and morphology. This variability arises from differences in phytochemical composition, concentration, and functional group distribution among plant species, plant parts, and extraction conditions. Spectroscopic analyses such as FTIR have provided insight into the functional groups involved in nanoparticle formation. For example, Chand et al. (2019) reported that flavonoids and terpenoids present in onion and tomato extracts were primarily responsible for the reduction

and stabilization of AgNPs. However, the precise contribution of individual phytochemicals often remains difficult to isolate due to synergistic interactions within the complex NP matrix.

As phytochemical profiles differ significantly across botanical sources, the selection of plant material plays a crucial role in determining the efficiency, reproducibility, and functionality of green-synthesized nanomaterials. While fresh plant extracts have been widely investigated, increasing attention is being directed toward agricultural waste materials such as fruit peels and plant residues, which are rich in bioactive compounds yet remain underutilized (Ying et al., 2022). These waste-derived resources offer a sustainable, cost-effective alternative while supporting circular-economy principles through biomass valorization. Therefore, exploring agricultural waste as a reducing and stabilizing agent offers a promising pathway to the controlled synthesis of Ag/TiO<sub>2</sub>-based NCs with enhanced photocatalytic and antibacterial performance.

### **2.6.2 POME as a green reducing and stabilizing agent**

POME is a highly concentrated wastewater generated during the final stages of palm oil processing, characterized by significant levels of organic pollutants, solids, and oils that pose substantial environmental risks if not properly managed. Despite its reputation as a problematic industrial waste, POME has emerged as a viable and sustainable alternative to conventional plant extracts for the green synthesis of nanomaterials (Aliero et al., 2024; Gan et al., 2012). As a waste-derived resource, POME aligns with circular economy principles by providing a low-cost, eco-friendly source of bioactive compounds needed for the biogenic reduction of metal ions. The transition from fresh botanical extracts to POME is increasingly seen as a strategic pathway to mitigate environmental pollution while simultaneously producing high-value nanomaterials for biomedical and industrial applications.

The viability of POME as a reducing and stabilizing agent is primarily attributed to its rich profile of phenolic compounds and other bioactive phytochemicals. High-Performance Liquid Chromatography and Gas Chromatography Mass Spectroscopy analyses have identified several key phenolic acids in POME, including gallic acid, caffeic acid, ascorbic acid, vanillic acid, coumaric acid, and myricetin (Aliero et al., 2024). These compounds possess potent antioxidant properties and contain functional groups such as

hydroxyl, carbonyl, and carboxylic groups that actively facilitate the reduction of  $\text{Ag}^+$  into  $\text{Ag}^0$  (Aliero et al., 2025). Furthermore, these phytochemicals act as capping agents, forming a protective layer around the nanoparticles that prevents aggregation and ensures long-term stability.

In addition to phenolic acids, POME contains various other phytonutrients, including carotenoids, tocopherols (vitamin E), sterols, and squalene, which can be extracted using green techniques such as microwave- or ultrasound-assisted extraction (Ofori-Boateng & Lee, 2013; Teh et al., 2017). The synergistic interaction of these diverse bioactive components enables POME to serve as an effective all-in-one matrix for nanomaterial formation, often yielding AgNPs with superior stability and significant biological activity (Aliero et al., 2025). For instance, AgNPs synthesized from POME have demonstrated high crystalline stability and remarkable antibacterial and antibiofilm efficacy against pathogens such as *Staphylococcus aureus* and *Pseudomonas aeruginosa*. By harnessing these discovered bioactive compounds, the utilization of POME not only addresses industrial waste management challenges but also provides a robust, reproducible platform for the controlled synthesis of functional Ag/TiO<sub>2</sub>-based NCs.

## 2.7 Immobilization of Ag/TiO<sub>2</sub>-based nanocomposites onto membranes

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Regardless, the main issue in the actual application of TiO<sub>2</sub>-based NCs is the difficulty of separating them after post-treatment. Therefore, the immobilization of TiO<sub>2</sub>-based NCs has been actively studied to overcome this problem. Various materials, particularly polymers and ceramics, have been widely used as supports for TiO<sub>2</sub>-based UV-responsive photocatalytic membranes (PMs). Some of the immobilized TiO<sub>2</sub> PMs include TiO<sub>2</sub>-coated and TiO<sub>2</sub>-blended PMs. This is achieved by various methods such as dip-coating (Djafer et al., 2010; Fischer et al., 2015), layer-by-layer self-assembly (Starr et al., 2016), electrospraying (Lang et al., 2022), atmospheric plasma spraying (APS) (Lin et al., 2012), as well as chemical vapor deposition (CVD), which are summarized in Table 2.3.

**Table 2–3: Fabrication methods of TiO<sub>2</sub>-based UV-responsive photocatalytic membranes.**

	Membrane type	TiO <sub>2</sub> precursor/TiO <sub>2</sub> type	Membrane fabrication method	Remarks	Reference
<b>TiO<sub>2</sub>-coated membranes</b>	Al <sub>2</sub> O <sub>3</sub> UF tubular membrane	Titanium isopropoxide (TTIP)	Chemical vapor deposition (CVD)	<ul style="list-style-type: none"> <li>• A gaseous stream enriched with TTIP vapor.</li> <li>• Pyrolytic decomposition of TTIP vapor in a 600 °C furnace and TiO<sub>2</sub> NPs formation.</li> <li>• TiO<sub>2</sub> nanoparticles deposited on the membrane surface.</li> </ul>	Athanasekou et al. (2012)
	Al <sub>2</sub> O <sub>3</sub>	TiO <sub>2</sub>	Atmospheric plasma spraying	<ul style="list-style-type: none"> <li>• TiO<sub>2</sub> powder preparation.</li> <li>• The plasma power (21 kW) was used to spray TiO<sub>2</sub> powder at a spraying distance (10 cm) to coat TiO<sub>2</sub> on Al<sub>2</sub>O<sub>3</sub> support.</li> </ul>	Lin et al. (2012)
	Nylon-6	TiO <sub>2</sub> NPs	Electrospraying	<ul style="list-style-type: none"> <li>• The coating was constructed by coupling the electrospinning of a polymer solution with the electrospray of TiO<sub>2</sub>.</li> </ul>	Daels et al. (2014)
<b>TiO<sub>2</sub> blended membranes</b>	PVDF	TiO <sub>2</sub> NPs	Phase inversion	<ul style="list-style-type: none"> <li>• Casting solutions included PVDF and TiO<sub>2</sub> particles in n-methyl-2-pyrrolidone (NMP) solvent at 60–65 °C.</li> <li>• Casting solutions were cast onto the non-woven sheet as a base substrate.</li> <li>• Membranes were immersed in a</li> </ul>	Damodar et al. (2009)

			23–25 °C tap water coagulation bath for 1 day and were washed with distilled water.	
Cellulose triacetate membrane (CTA)	TiO <sub>2</sub> NPs	Phase inversion	<ul style="list-style-type: none"> <li>• Casting solutions included CTA, plasticizer, and TiO<sub>2</sub> in CH<sub>3</sub>Cl solvent.</li> <li>• Casting solutions were mixed and placed to allow the solvent evaporation and polymerization occurrence.</li> </ul>	Molinari et al. (2004)
PVDF/SPES	TiO <sub>2</sub> NPs	Phase inversion	<ul style="list-style-type: none"> <li>• Polyethersulfone sulfonation treatment (SPES).</li> <li>• Casting solutions included PVDF, SPES and TiO<sub>2</sub> in DMAc solvent with polyvinyl pyrrolidone (PVP) as the pore former.</li> <li>• Casting solutions were cast on polyester non-woven fabric, and were immersed in water, and washed.</li> </ul>	Rahimpour et al. (2011)

Whether they are for ceramic or polymer supports, the bonding strength of the photocatalysts and membrane supports is a key criterion, influencing the membrane's normal operation. Several studies have shown that the attachment of photocatalysts to membranes is often insufficient to enable the photocatalytic membrane to withstand multiple cycles without a decline in performance (Leong et al., 2014). To achieve better adhesion, membrane modification has been targeted and has attempted to introduce the key functional groups on the membrane surface that can stably hold TiO<sub>2</sub>-based catalysts. It is well known that the surface of polymers with –COOH, –SO<sub>2</sub>OH, and ether bonds exhibits self-assembly behaviors of TiO<sub>2</sub> through two different adsorption schemes (Shi et al., 2019). One scheme

is that  $\text{TiO}_2$  is bound to the oxygen atoms of these groups via coordination with the ion  $\text{Ti}^{4+}$ . The other scheme is to form a hydrogen bond between these groups and the hydroxyl group of the  $\text{TiO}_2$  surface. For instance, the plasma-enhanced graft of polyacrylic acid (PAA) with  $-\text{COOH}$  groups on the surface of the membrane, including PVDF and PTFE, has been reported to facilitate  $\text{TiO}_2$  to deposit onto functionalized membrane surfaces successfully (You et al., 2012). Another approach was attempted by blending other polymers, such as poly- (styrene-alt-maleic anhydride) (SMA) with  $-\text{COOH}$ , and the sulfonated polyethersulfone (SPES) with sulfone group and ether bond for  $\text{TiO}_2$  self-assembly.

Through green synthesis, bonding between the  $\text{TiO}_2$ -based NC and the membrane supports may be achieved, as plant extracts attached to the green-synthesized NC surface may strengthen interactions with functional groups on the membrane surface. Moreover, the addition of fabricated  $\text{TiO}_2$  could enable a responsive photocatalytic membrane in the visible-light region. This can promote the application of these membranes under solar-irradiated systems.

# RESEARCH METHODOLOGY

## 3.1 Materials

Titanium butoxide,  $\text{Ti}(\text{OCH}_2\text{CH}_2\text{CH}_2\text{CH}_3)_4$  (97% reagent grade), bismuth (III) nitrate pentahydrate,  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  (ACS reagent,  $\geq 98.0\%$ ), Sodium molybdate dihydrate,  $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$  (ACS reagent,  $\geq 99\%$ ), urea,  $\text{NH}_2\text{CONH}_2$  (ACS reagent, 99.0-100.5%), and polyvinylpyrrolidone,  $(\text{C}_6\text{H}_9\text{NO})_n$  (average  $M_w \sim 1,300,000$ ) were purchased from Sigma-Aldrich. Hydrochloric acid (HCl), Sodium hydroxide (NaOH), ethylene glycol ( $\text{HOCH}_2\text{CH}_2\text{OH}$ ,  $\geq 99.5\%$ ) and absolute ethanol (EtOH) were obtained from Merck. Silver nitrate,  $\text{AgNO}_3$  (AR, 99.98%), was procured from Bendosen. Other materials include polyacrylonitrile  $(\text{C}_3\text{H}_3\text{N})_n$  (average  $M_w \sim 150,000$ ), ultrapure water, UPW (HPLC grade) and dimethyl sulfoxide,  $(\text{CH}_3)_2\text{SO}$  (AR,  $\geq 99.9\%$ ) that were purchased from Shanghai Macklin Biochemical Co., Ltd., Fisher Chemical, and RCI Labscan Ltd., respectively. A schematic overview of the experimental methodology performed in this study is illustrated in Figure 3.1.

**Figure 3–1: Schematic overview of the overall experimental methodology**



## 3.2 Preparation and analysis of POME extracts

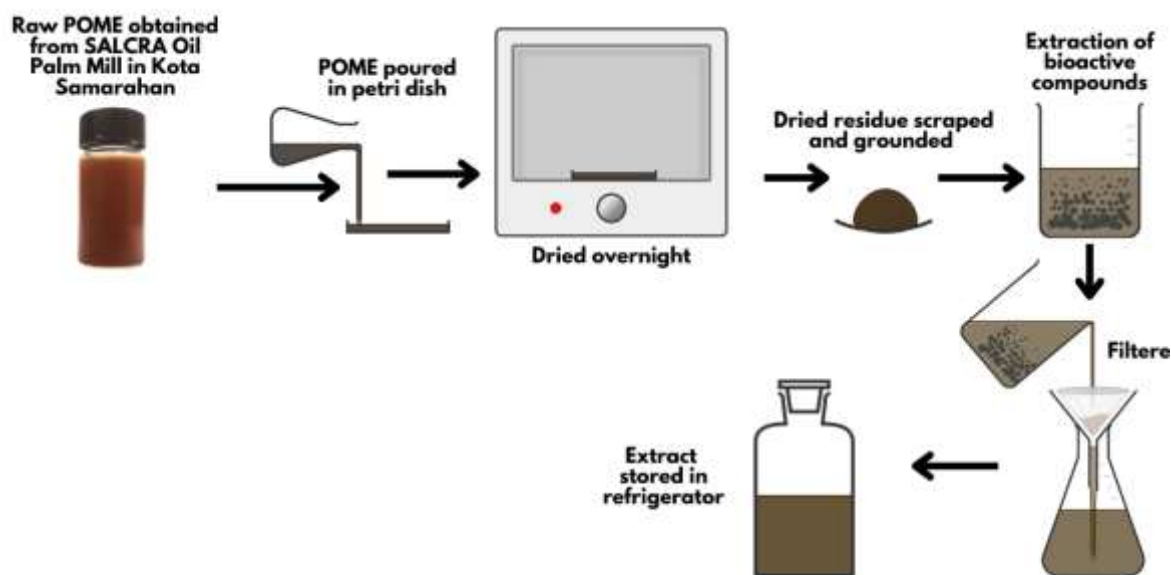
Raw POME was obtained from the SALCRA Oil Palm Mill in Kota Samarahan, Sarawak. POME extract was prepared according to a previous study (Gan et al., 2012).

Sampled raw POME was dried at 60 °C for 24 hours in a glass container. After drying, the POME samples were scraped, ground into powder, and sieved to remove coarse particles. Bioactive compounds were extracted from the dried POME by mixing 3 g of the dried POME powder with 100 mL of distilled water and stirring the mixture at 80 °C for 30 minutes. The mixture was then filtered using Whatman filter paper to attain the POME extract. The extract was kept in a refrigerator at 4 °C until further use. Figure 3.2 shows the extraction process of raw POME.

Total phenolic content (TPC) and Total flavonoid content (TFC) analysis was carried out to confirm the presence of bioactive compounds that were responsible for the reduction and stabilization of the POME-derived ternary NCs. TPC and TFC analysis was carried out using the Follin-Ciocalteu method and aluminum chloride colorimetric method as described by Nuzul et al. (2022) with some slight modifications. Prior to TPC and TFC analysis, POME extracts were subjected to a vacuum filter to obtain the crude extracts. For TPC analysis, about 5 mg of the samples was dissolved in 5 mL of EtOH. A total of 300 uL of the dissolved samples was taken and put in a test tube. Then, 2250 uL of Folin–Ciocalteu reagent was added and allowed to rest for about 10 minutes before 2250 uL of 6% sodium carbonate solution was added to the mixture. The mixture was gently mixed and left to further react for 60 minutes. This was followed by measuring the absorbance value via UV-Vis spectrometry at  $\lambda$  max of 765 nm by using gallic acid standards (100, 200, 300, 400, 500 ug/mL) as the reference standard. The results were expressed as mg GAE/g extract.

For TFC analysis, 5 mL of the dissolved sample was mixed with 5 mL of 2% aluminum chloride solution in a test tube. The mixture was shaken and left to stand for 10 minutes before being analyzed with UV-Vis spectrometry. Flavonoid detection was performed at the  $\lambda$  max of 415 nm, with quercetin (25, 50, 75, 100, 125 ug/mL) as the reference standard. The results were expressed as mg QE per g of extract.

Figure 3–2: Extraction of bioactive compounds from raw POME



### 3.3 Synthesis of $\text{TiO}_2$ -g- $\text{C}_3\text{N}_4$ and $\text{Bi}_2\text{MoO}_6$

$\text{TiO}_2$  was synthesized using the sol-gel method with titanium (IV) butoxide as the precursor, while g- $\text{C}_3\text{N}_4$  was prepared via pyrolysis of urea, following our previously published work (Kanakaraju et al., 2024). The sol-gel synthesis of  $\text{TiO}_2$  was performed by mixing titanium (IV) butoxide with EtOH in a 1:2 ratio. The mixture was stirred for 30 minutes, after which 100 mL of distilled water was added dropwise. The resulting solution was then stirred for an additional 2 hours, followed by filtration and washing three times with distilled water and EtOH. The gel was then dried at 110 °C for 2 hours before calcination in a furnace (Protherm PLF 110/30) at 300 °C for 5 hours.

For g- $\text{C}_3\text{N}_4$  synthesis, approximately 10 g of urea was placed in a crucible and subjected to pyrolysis at 550 °C for 3 hours, resulting in bulk g- $\text{C}_3\text{N}_4$ . The g- $\text{C}_3\text{N}_4$  obtained was dispersed in deionized water, then filtered and washed to remove impurities. The resulting material was then dried in an oven to obtain the yellow bulk g- $\text{C}_3\text{N}_4$ .

$\text{Bi}_2\text{MoO}_6$  was synthesized via a green solvothermal method with slight modifications from reported procedures (Kanwal et al., 2023; Selvamani et al., 2023). In this process, 2 mmol of  $\text{Bi}(\text{NO}_3)_3 \cdot 5\text{H}_2\text{O}$  and 1 mmol  $\text{Na}_2\text{MoO}_4 \cdot 2\text{H}_2\text{O}$  were each dissolved in 30 mL of ethylene glycol under magnetic stirring to obtain Solution A and B, respectively.

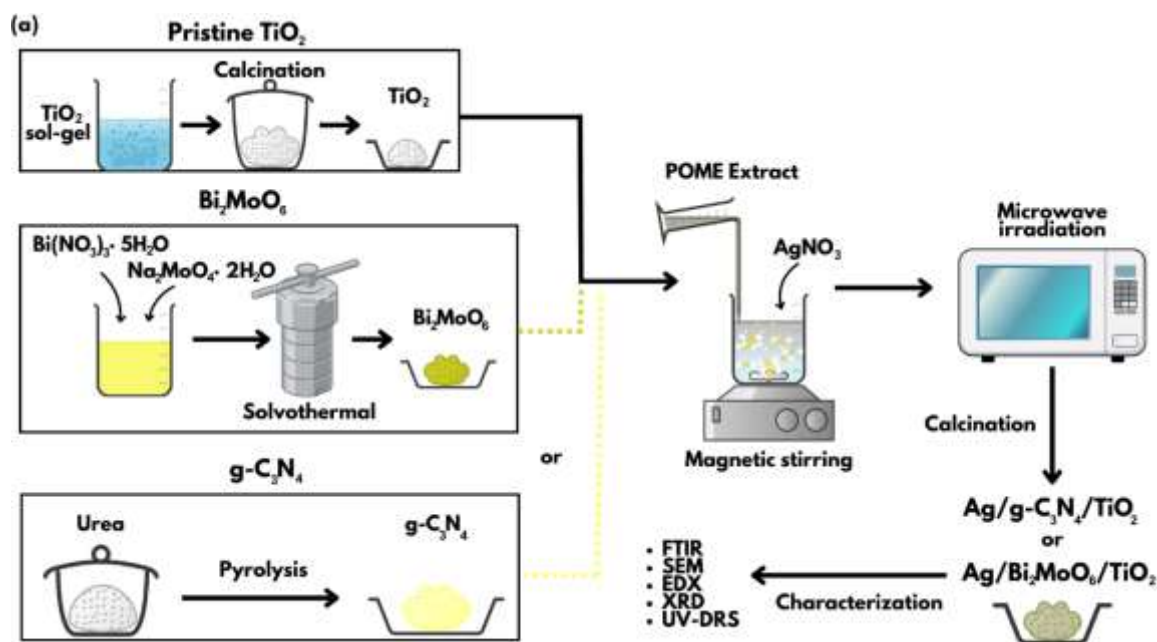
Subsequently, Solution A and Solution B were combined, and 30 mL of EtOH was added. A 3 mL of POME extract (10% v/v relative to the ethanol volume) was then added to the mixture as a stabilizing agent, and the solution pH was adjusted to 8, resulting in the precipitation of  $\text{Bi}_2\text{MoO}_6$ . The solution was transferred into a hydrothermal reactor and kept at 150 °C for 12 hours. After the reaction was completed, the precipitate was filtered and washed with deionized water and EtOH several times. Finally, the precipitate was dried in an oven at 110 °C for 2 hours.

### 3.4 Green synthesis of $\text{Ag}/\text{Bi}_2\text{MoO}_6/\text{TiO}_2$ and $\text{Ag}/\text{g-C}_3\text{N}_4/\text{TiO}_2$ NCs

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The synthesis of  $\text{Ag}/\text{Bi}_2\text{MoO}_6/\text{TiO}_2$  (ABMOT) NC was performed by mixing  $\text{TiO}_2$  and  $\text{Bi}_2\text{MoO}_6$  in a 1:0.3 ratio in 50 mL of distilled water. The mixture was then left under continuous stirring for 30 minutes. Ag deposition was carried out by dissolving  $\text{AgNO}_3$  in 50 mL of water, followed by the addition of 5 mL of POME extract (10% v/v relative to the water volume) to facilitate Ag reduction. The POME extract used for Ag deposition was prepared from the same raw material, with the same extraction conditions applied throughout, ensuring a consistent bioactive composition across batches. This solution was then combined with the  $\text{Bi}_2\text{MoO}_6/\text{TiO}_2$  mixture and subjected to microwave irradiation in a commercial microwave oven at 100 W for 6 minutes to promote the deposition of AgNPs onto the binary system. The resulting mixture was subsequently stirred magnetically for another 2 hours, then filtered, washed and dried at 100 °C for 2 hours. The resulting ABMOT ternary NC was then calcinated at 450 °C for 3 hours and stored for further testing. The synthesis of  $\text{Ag}/\text{g-C}_3\text{N}_4/\text{TiO}_2$  (AGCNT) was done in the same manner as the above-mentioned method, but by mixing  $\text{g-C}_3\text{N}_4$  with  $\text{TiO}_2$  in a 1:1 ratio. A schematic summary of the overall synthesis methodology for the prepared NCs is presented in Figure 3.3.

Figure 3–3: Schematic summary for the synthesis of ABMOT and AGCNT NCs.



### 3.5 Characterization of POME-derived NCs

Scanning electron microscopy (SEM) coupled with Energy-dispersive X-ray spectroscopy (EDX) (Model: JEOL, JSM-IT500HR) was used for the determination of morphological properties and elemental composition of the NCs. Analysis of functional groups that were present in the NCs was done via Fourier transform infrared spectroscopy (FTIR) (Model: Thermo Nicolet iS10) by using KBr pellets as a standard at a wavelength of  $4000 - 400 \text{ cm}^{-1}$ . X-ray diffraction analysis (XRD, PANalytical X'pert Pro) was done to determine the crystalline structure and phase of the green-synthesized samples. UV-vis diffuse reflectance spectroscopy (UV-Vis DRS) (Model Carry 5000) was employed to determine the effective absorption spectra of NCs, and the Kubelka-Munk plot was established to confirm any changes in the energy bandgap.

### 3.6 Evaluation of photocatalytic activity

The photocatalytic efficiency of the synthesized NCs was determined by studying their ability to degrade OTC antibiotics under visible light (Figure 3.4). First, a fixed amount of the ternary NC was added to 10 mg/L of OTC solution and mixed with a magnetic stirrer. The suspension was left to stir in the dark for 1 hour to allow for the adsorption-desorption

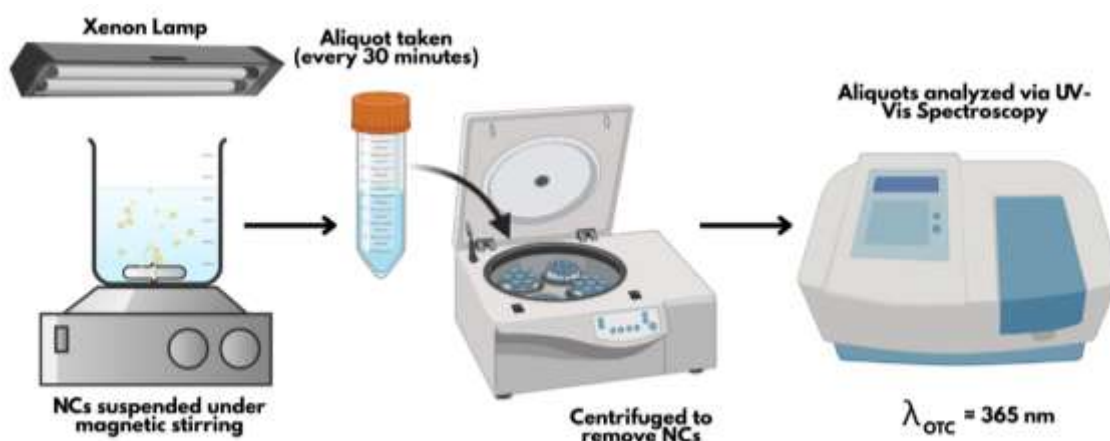
equilibrium of OTC onto the photocatalyst. After 1 hour under dark conditions, the suspension was irradiated under the illumination of Xenon Lamp (300 W) at 5,800,000 lux to simulate sunlight. The photocatalytic degradation experiment was done for another 2 hours. A fixed aliquot was taken every 30 minutes over the 3-hour photocatalytic experiment under simulated sunlight. All aliquots of the suspension were spun with a tabletop centrifuge (Hettich, EBA 200) at 6000 rpm for 10 minutes before being subjected to analysis with a UV-Vis spectrophotometer. The analysis was done by monitoring the changes at  $\lambda_{\text{max}}$  of 352 nm, the OTC removal efficiency was calculated as follows:

$$\text{Removal efficiency (\%)} = \frac{C_i - C_f}{C_i} \times 100\% \dots\dots\dots(\text{Eq 3.1})$$

where  $C_i$  is the initial concentration of OTC and  $C_f$  is the final concentration of OTC. The photocatalytic treatment was conducted by varying the catalyst dosage (0.3 g/L, 0.6 g/L, 1.2 g/L) and the initial concentration of OTC (10 mg/L, 15 mg/L, 20 mg/L).

Additionally, a recyclability study was conducted to assess the stability of AGCNT and ABMOT over 4 consecutive rounds. This was done by running photocatalysis at optimized conditions of 0.6 g/L catalyst dosage with an initial concentration of 10 mg/L for OTC. After one cycle, the photocatalyst was recovered through centrifugation, washed with distilled water and ethanol to remove residual pollutants, followed by drying and subsequently reusing them under the same experimental conditions.

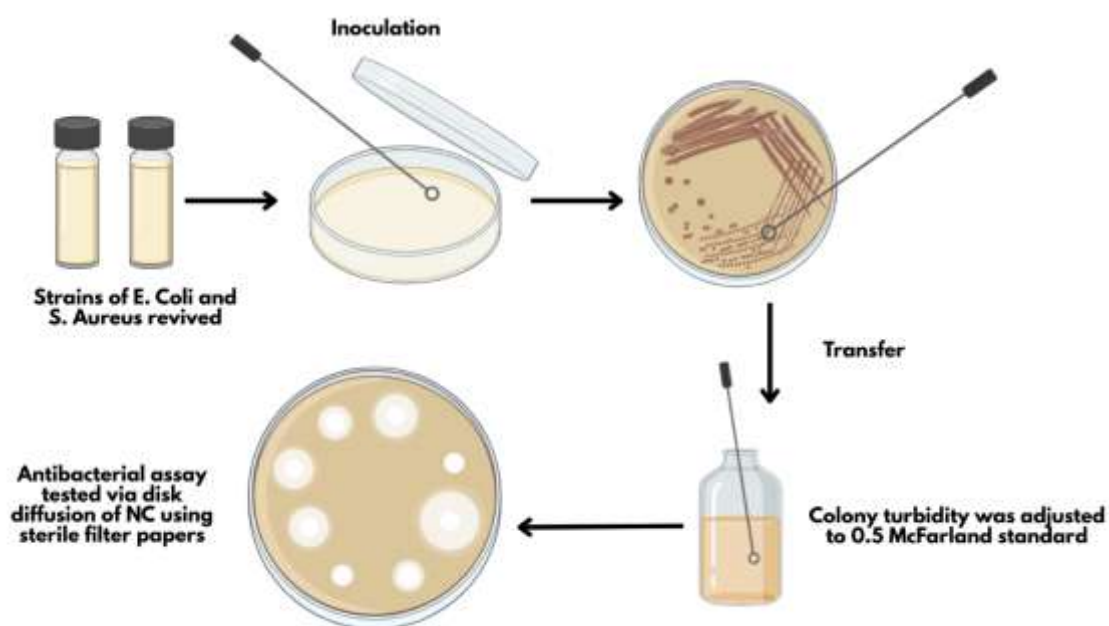
**Figure 3–4: Photocatalytic evaluation of ABMOT and AGCNT NCs for the degradation of OTC under visible light.**



### 3.7 Evaluation of antibacterial activity

The antibacterial study was conducted via the disc diffusion method to determine the antibacterial effectiveness of both ABMOT and AGCNT NCs against both gram-positive bacteria, *E. coli* and gram-negative bacteria, *S. aureus* (Figure 3.5). The bacterial strains were obtained from the Microbiology Laboratory 2, Faculty of Resource Science and Technology, UNIMAS. The bacterial strains were revived in nutrient broth (NB) and inoculated onto nutrient agar (NA) plates. Next, the inoculum was standardized by transferring several colonies into fresh NB and adjusting the culture to a turbidity that is equivalent to 0.5 McFarland standard. The antibacterial assay was conducted by first comparing the effectiveness of ABMOT and AGCNT with their unary and binary counterparts in inhibiting the growth of the bacterial samples. All samples were fixed at a concentration of 1 g/L. A determined amount of the samples mentioned was mixed in a phosphate buffer (PBS) solution. Sterile filter papers were then dipped into the PBS solution containing these samples. After that, the standardized inoculum was then swabbed onto NA plates using sterile cotton swabs and the treated filter papers were then placed on the surface of the agar and incubated at 37 °C for 24 hours. The diameter of the inhibition zones was measured after the incubation period. This experiment was repeated by varying the concentration of ABMOT and AGCNT at 1, 2, 4, 8, and 10 g/L, respectively.

**Figure 3–5: Disk-diffusion method for the assessment of antibacterial activity of ABMOT and AGCNT NCs.**



### 3.8 Fabrication of POME-derived ternary NC into PAN membranes

The fabrication of NCs into the PAN membrane was done via the phase inversion method. In this process, 85 g of DMSO was measured and poured into a beaker. This was followed by the addition of 1% of ABMOT NC based on the weight of PAN into the beaker containing the solvent. The solution was stirred for 1 hour at 80 °C. After thorough mixing, the solution was transferred into a conical flask, where 15 g of PAN powder was added. The mixture was then continuously stirred at 80 °C for 24 hours until complete dissolution and homogenization were achieved. The final solution was then cast onto a clean glass plate using a casting rod to obtain a uniform membrane film. The glass plate was then immersed in a deionized water coagulation bath to induce phase inversion and form the solid membrane structure. After complete phase separation, the membrane was carefully peeled off and further soaked in ultrapure water to remove residual solvents. The final membranes were stored in ultrapure water until further use. A similar method was used to fabricate AGCNT. A bare PAN membrane was also prepared in the same manner, except without the addition of NC. A summary of the composition used to prepare the bare PAN membrane and NC membranes is summarized in Table 3.1.

**Table 3–1: Composition of ABMOT and AGCNT NCs-embedded PAN membranes**

Membrane	PAN (g)	DMSO (g)	NC loading (g)
PAN	15.00	85.00	0.00
1% ABMOT@PAN	15.00	85.00	0.15
2% ABMOT@PAN	15.00	85.00	0.30
1% AGCNT@PAN	15.00	85.00	0.15
2% AGCNT@PAN	15.00	85.00	0.30

### 3.9 Characterization of membranes

The membranes were characterized using SEM (Model: JEOL JSM-IT500HR), coupled with EDX, to determine their surface morphology and cross-section. Additionally, FTIR spectroscopy (Model: Thermo Nicolet iS10, equipped with attenuated total reflectance

(ATR)) was used to confirm the chemical interaction of NCs within the membrane matrix in the spectral range of 2000–400 cm<sup>-1</sup>.

### 3.10 Evaluation of fabricated membranes

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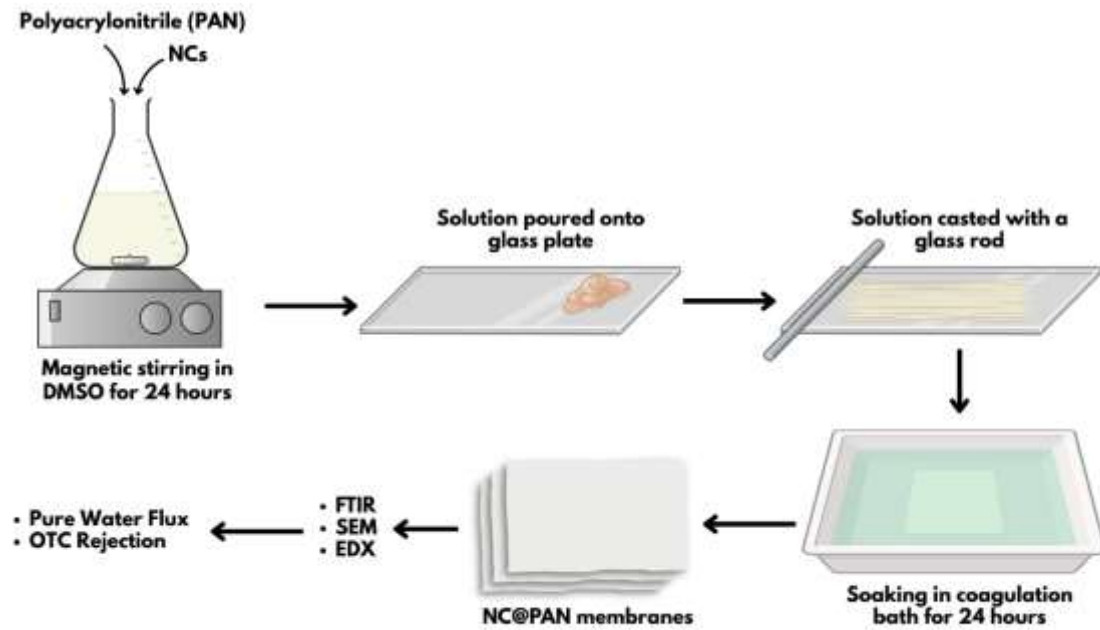
The membrane permeation test served as the primary method for evaluating the performance of pristine PAN membranes and those incorporating ABMOT and AGCNT ternary NCs. This test consisted of pure water flux and OTC rejection. For the pure water flux test, membranes were compacted using a backflow that was fed back into the feed of the filtration system setup. Pure water flux was determined by pumping ultrapure water to the system at a pressure of 2 bar for 1 hour for stabilization of the membrane system. Then, the water flux test was conducted at a pressure of 1 bar for another hour, and the results were recorded every 10 minutes. The procedure was repeated 3 times to ensure accurate reading and replicability of water flux results. Similarly, the OTC rejection test was performed following the same procedure but with 10 g/L of OTC solution as the feed. The concentrations of the initial feed and permeate were measured using a UV-Vis spectrophotometer (Model: Perkin-Elmer LS55) at a  $\lambda$  max of 352 nm. The water flux and OTC rejection were obtained following the equation as described by Huang et al. (2022):

$$J_{flux} = \frac{V}{A \times t \times P} \text{ (L} \cdot \text{m}^{-2} \cdot \text{h}^{-1}\text{)} \dots\dots\dots(\text{Eq 3.2})$$

$$\text{RR (\%)} = \frac{C_{feed} - C_{permeate}}{C_{feed}} \times 100\% \dots\dots\dots(\text{Eq 3.3})$$

where,  $J_{flux}$  represents the water flux (L·m<sup>-2</sup>·h<sup>-1</sup>), V is the volume of permeate, A, is the effective area of membrane (m<sup>2</sup>), P is the operational pressure (bar), RR is rejection rate (%),  $C_{feed}$  is the concentration of OTC in the feed and  $C_{permeate}$  is the concentration of OTC from the permeate collected. A summary of the membrane workflow is shown in Figure 3.6.

Figure 3–6: Schematic overview of the synthesis, characterization, and evaluation of NC-embedded PAN membranes.



## FINDINGS AND DISCUSSION

### 4.1 Analysis of POME extracts

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#### 4.1.1 TPC and TFC analysis of POME extracts

To verify the suitability of POME as a green reducing and stabilizing agent in NC synthesis, compound confirmation tests were conducted to determine the key phytochemicals, with particular attention to phenolic and flavonoid compounds. These compounds are well-known for their role in the green synthesis of NPs, as they can both reduce metal ions and prevent agglomeration, thereby promoting the formation of stable NPs. The TPC content of POME extracts obtained was  $240.70 \pm 2.64$  gallic acid equivalent (GAE) mg/g (Table 4.1). These values align well with past studies that focused on the extraction of phenolics from POME, confirming that this waste stream harbors significant quantities of polyphenols (Ofori-Boateng & Lee, 2013). Notably, the phenolic content observed is comparable to that of other plant extracts that were previously studied, highlighting the potential of POME as a possible green source for nanomaterial synthesis. (Bhardwaj & Singh, 2021; Nguyen et al., 2022; Rahmawati et al., 2023). This opens up possibilities for agro-industrial waste to be utilized and become added-value products, thus promoting a circular economy.

The TFC in the extracts was found to be 42.50 quercetin equivalents (QE) mg/g (Table 4.1). Torres-Limiñana et al. (2022) reported the green synthesis of Ag-TiO<sub>2</sub> using an extract of *Eucalyptus globulus* and assessed the presence of secondary metabolites, namely the phenolics and flavonoid contents. Compared to the current study, their findings indicated that a higher flavonoid content relative to phenolics. This variation could be attributed to the differences in solvent selection, which may have favored the extraction of phenolics over flavonoids. Nevertheless, given that phenolics and flavonoids were detected in the POME extracts, their presence makes them well-suited for the synthesis of ABMOT and AGCNT.

Additionally, while there is no universal threshold for TPC and TFC values required for green nanomaterial synthesis, previous studies have shown that extracts containing sufficient phenolic and flavonoid compounds can effectively function as reducing and stabilizing agents (Hussain et al., 2025). The effectiveness of these extracts depends not only on their total content but also on the chemical nature and redox activity of the bioactive compounds present. In this study, the high TPC and appreciable TFC values obtained for POME fall within, and in some cases exceed, the ranges of 20 mg/L for TPC and TFC reported for successful green synthesis of nanomaterials (Vera et al., 2023). This further confirms the suitability of POME as a viable green source for the fabrication of ABMOT and AGCNT NCs.

**Table 4–1: TPC expressed in gallic acid equivalent (GAE) (mg/g) and TFC expressed in quercetin equivalent (QE) (mg/g) of the POME extracts.**

Extract	TPC as GAE (mg/g)	TFC as QE (mg/g)
POME Extract	240.70 ± 2.64	42.50 ± 0.01

## 4.2 Characterization of ABMOT and AGCNT

### 4.2.1 FTIR spectra

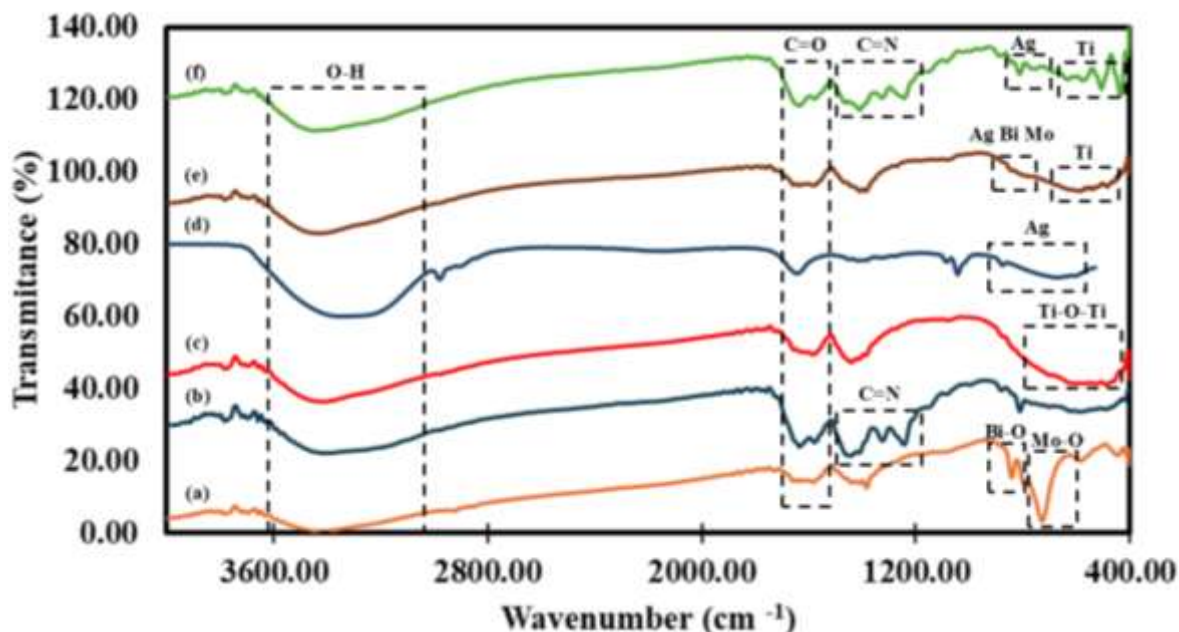
FTIR spectra of unary components, TiO<sub>2</sub>, Bi<sub>2</sub>MoO<sub>6</sub>, g-C<sub>3</sub>N<sub>4</sub>, AgNP, as well as the ternary heterostructure of ABMOT and AGCNT, were analyzed within the range of 400 – 4000 cm<sup>-1</sup> to identify the functional groups present in the NCs. Through FTIR analyses, it is possible to evaluate any potential chemical interactions between each component in the NCs matrix.

Figure 4.1 shows the FTIR spectra of both ABMOT and AGCNT, compared with their individual components. FTIR analysis shows that pristine TiO<sub>2</sub> had broad absorption peaks at 3425 cm<sup>-1</sup> and strong peaks at 640 cm<sup>-1</sup> corresponding to -OH stretching and Ti-O-Ti bonds, respectively (Figure 4.1c). Bi<sub>2</sub>MoO<sub>6</sub> showed strong characteristic peaks at 1107 cm<sup>-1</sup>, 813 cm<sup>-1</sup> and 729 cm<sup>-1</sup> corresponding to Bi-O bond, Mo-O stretching vibration modes and MoO<sub>6</sub> asymmetric stretching, respectively, as shown in Figure 4.1b (Cai et al., 2019;

Zhang et al., 2016). The characteristic peaks for g-C<sub>3</sub>N<sub>4</sub> were mainly detected from 800-1700 cm<sup>-1</sup>, with the s-triazine ring peak detected at 810 cm<sup>-1</sup> (Figure 4.1a). The bands at 1639, 1566 and 1458 cm<sup>-1</sup> are attributed to aromatic C-N vibrations, while bands at 1323 and 1246 cm<sup>-1</sup> belong to the vibration of fully condensed N-(C)<sub>3</sub> and partially condensed C-N-H link, respectively (Xiao et al., 2021). The characteristic bands of AgNPs at 1384 cm<sup>-1</sup> and 1620 cm<sup>-1</sup> are mainly ascribed to C=C stretching of aliphatic alkenes and C=O from aldehydes or ketones, respectively, as seen in Figure 4.1d (Das et al., 2025). The presence of C=C and C=O peaks is ascribed to the phenolic compounds that are present on the surface of the AgNPs. Additionally, the C=O peak observed in TiO<sub>2</sub>, Bi<sub>2</sub>MoO<sub>6</sub>, and ABMOT NCs may also arise from weakly adsorbed carbon dioxide from the surrounding environment, which can interact with surface hydroxyl groups to form surface carbonate species. The peaks observed at approximately 671 and 779 cm<sup>-1</sup> are attributed to Ag-O-Ag stretching vibrations.

In the case of ABMOT (Figure 4.1e), most of the characteristic bands of TiO<sub>2</sub> were maintained, except for the appearance of broader bands between 600-800 cm<sup>-1</sup>, a distinct peak at 1384 cm<sup>-1</sup> and a widened peak at 3142 cm<sup>-1</sup>. The observed could be linked to the presence of Bi-O and Mi-O bonds and bound organic compounds as described. Similarly, changes in the AGCNT peaks (Figure 4.1f) were detected. The broader peak at 3240 cm<sup>-1</sup> is attributed to overlapping O-H stretching vibrations of TiO<sub>2</sub> and the N-H bonds of g-C<sub>3</sub>N<sub>4</sub>. The peaks in the 800-1700 cm<sup>-1</sup> range associated with g-C<sub>3</sub>N<sub>4</sub> remained intact, while the broad peaks observed between 600 and 800 cm<sup>-1</sup> were attributed to the overlapping of s-triazine ring vibrations with Ti-O-Ti and Ag-O-Ag stretching modes, each within that range. Thus, the FTIR spectra of the samples confirm the presence of all key functional groups associated with ABMOT and AGCNT ternary heterostructures.

**Figure 4–1: FTIR spectra of (a) Bi<sub>2</sub>MoO<sub>6</sub>, (b) g-C<sub>3</sub>N<sub>4</sub>, (c) TiO<sub>2</sub>, (d) AgNP, (e) ABMOT NC and (f) AGCNT NC**



#### 4.2.2 XRD analysis

XRD analysis was conducted to investigate the crystalline structure and phase composition of the synthesized samples. This is essential in confirming the successful formation of heterojunctions and the presence of individual components in ABMOT and AGCNT NC matrix. The diffraction patterns were compared with standards from the Joint Committee on Powder Diffraction Standards (JCPDS) files to identify the characteristic peaks corresponding to each phase. Additionally, shifts, broadening or disappearances of the peaks were analyzed to gain insights into the structural interactions among the components within the ternary heterostructures of ABMOT and AGCNT NCs.

The XRD patterns of Ag, Bi<sub>2</sub>MoO<sub>6</sub>, g-C<sub>3</sub>N<sub>4</sub>, TiO<sub>2</sub> and the ternary heterostructures of ABMOT and AGCNT are depicted in Figures 4.2 and 4.3, respectively. For individual samples of TiO<sub>2</sub> (Figures 4.2a and 4.3a), it was observed that characteristic peaks at  $2\theta = 25.4^\circ, 38.5^\circ, 48.1^\circ, 55.1^\circ, 63.0^\circ$  (JCPDS 21-1272) correspond to anatase phase (101), (004), (200), (211), (204), and (220) crystal faces, respectively. The anatase phase of TiO<sub>2</sub> is well known to have the highest photocatalytic activity in comparison to the other polymorphs (Katal et al., 2020). This makes it suitable to be combined with other materials to further enhance its properties. Given the dominant formation of (101) crystal facets, this indicates

that the TiO<sub>2</sub> synthesized is thermodynamically stable and makes it favorable for long-term structural stability (Ye et al., 2013).

On the other hand, individual samples of AgNPs exhibited peaks of Ag (200) at  $2\theta = 44.6^\circ$ , Ag (220) and  $2\theta = 64.6^\circ$ , and Ag (311) at  $2\theta = 77.4^\circ$  (JCPDS 04-0783) (Figure 4.2c and 4.3b). Similar peaks were observed in a past study by Aliero et al. (2024) in which the characteristic peaks of AgNPs that were synthesized with POME, also exhibited peaks corresponding to (200), (220), and (311). However, the major peaks at  $2\theta = 28^\circ$ ,  $32^\circ$ , and  $38^\circ$  were absent in the POME-derived AgNPs. This suggests that Ag species present are poorly crystalline or partially oxidized, potentially indicating the formation of Ag<sub>2</sub>O or Ag-O related phases (Tan Sian Hui Abdullah et al., 2021). During the drying process, exposure to elevated temperatures may have disrupted the stabilizing interactions between AgNPs and POME-derived organic moieties, leading to structural disorder and reduced crystallinity, which in turn suppressed the appearance of distinct Ag<sup>0</sup> diffraction peaks.

Moreover, the diffraction peaks of g-C<sub>3</sub>N<sub>4</sub> corresponded to a hexagonal crystal system (space group P31c), with peaks at  $2\theta = 27.5^\circ$  and  $38.3^\circ$  (JCPDS 01-087-1522) which matched the planes (002) and (100), respectively (Figure 4.3c). The peak corresponding to the (002) plane indicates the interlayer stacking of aromatic s-triazine units, which formed a layer, graphitic-like structure that promotes  $\pi$ - $\pi$  conjugation and facilitates charge carrier transport (Wudil et al., 2023). The notably intense (100) reflection suggests a high degree of in-plane structural ordering, likely due to enhanced periodicity within the polymeric framework (Muhmood et al., 2024). These XRD patterns of the g-C<sub>3</sub>N<sub>4</sub> obtained coincide with FTIR spectra in which aromatic s-triazine units are detected, hence validating the succession of g-C<sub>3</sub>N<sub>4</sub> formation with good structuring order and crystallinity.

Next, the diffraction peaks of pure Bi<sub>2</sub>MoO<sub>6</sub> showed distinctive peaks around  $2\theta = 28.3^\circ$ ,  $32.5^\circ$ ,  $47.1^\circ$ , and  $55.8^\circ$  corresponding to (131), (200), (260) and (331) planes, respectively (Figure 4.2b). The XRD pattern suggests the formation of the orthorhombic Aurivillius phase of  $\gamma$ - Bi<sub>2</sub>MoO<sub>6</sub> (JCPDS 21-0102). Thus, confirming the succession of crystalline Bi<sub>2</sub>MoO<sub>6</sub>. This particular phase is desirable because it has a greater capacity to capture visible light than its fluorite-structured counterparts,  $\alpha$ -Bi<sub>2</sub>Mo<sub>3</sub>O<sub>12</sub> and  $\beta$ -Bi<sub>2</sub>Mo<sub>2</sub>O<sub>9</sub> (Lin et al., 2023).

Upon combining the individual components to form ABMOT and AGCNT ternary NCs, distinct changes were observed in their characteristic peaks. In the case of ABMOT

ternary NC, Ag peaks at  $2\theta = 44.6^\circ$  (200) and  $2\theta = 64.8^\circ$  (220) were detected, confirming that AgNPs were successfully deposited into the ternary NC matrix (Figure 4.2d). Similarly, all characteristic peaks of  $\text{Bi}_2\text{MoO}_6$  and  $\text{TiO}_2$  were retained, albeit with reduced intensity. This indicates the coexistence of different phases within the ternary system and reflects increased structural distortion, a typical feature of most NC systems.

A similar trend was observed in the AGCNT ternary NC, where the diffraction peaks of g- $\text{C}_3\text{N}_4$  and  $\text{TiO}_2$  were preserved but with reduced intensity. However, a key distinction between the two ternary NCs was the form of Ag present. In the case of AGCNT ternary NC, peaks corresponding to typical metallic Ag were absent. However, a distinct peak at  $2\theta = 32.7^\circ$  (111) corresponding to  $\text{Ag}_2\text{O}$  was present (JCPDS 031-1104) (Figure 4.3d). Based on this observation, it can be concluded that the variation in Ag phases between ABMOT and AGCNT NCs is attributed to the difference in the composite matrix. Past studies have shown that g- $\text{C}_3\text{N}_4$ -based photocatalyst predominantly stabilizes Ag in the form of  $\text{Ag}_2\text{O}$  due to a more oxidative environment. Thus, the interaction of  $\text{Ag}^+$  ions favors the formation of  $\text{Ag}_2\text{O}$  as compared to metallic Ag, allowing the AGCNT composite to benefit from p-n junction formation, which enhances the visible light absorption (Wang et al., 2022; Wu et al., 2015).

In contrast, when  $\text{Bi}_2\text{MoO}_6$  is introduced into the NC matrix, the formation of metallic Ag is favorable, likely due to the narrow bandgap and efficient electron transfer which supported the full reduction of  $\text{Ag}^+$  to  $\text{Ag}^0$  (Lin et al., 2015; Wang et al., 2018). In general, the XRD analysis confirms the successful synthesis and crystallinity of each individual component and their respective incorporation into ABMOT and AGCNT ternary NCs. Furthermore, the difference in Ag species in ternary ABMOT and AGCNT highlights the critical influence of the composite matrix in forming varying species of Ag as a result of varying redox environments provided by  $\text{Bi}_2\text{MoO}_6$  and g- $\text{C}_3\text{N}_4$ , respectively. Overall, the XRD result validates the structural integration of each component into the desired ternary NC architectures.

Figure 4-2: XRD spectra of (a) TiO<sub>2</sub>, (b) Bi<sub>2</sub>MoO<sub>6</sub>, (c) AgNP and (d) ABMOT NC

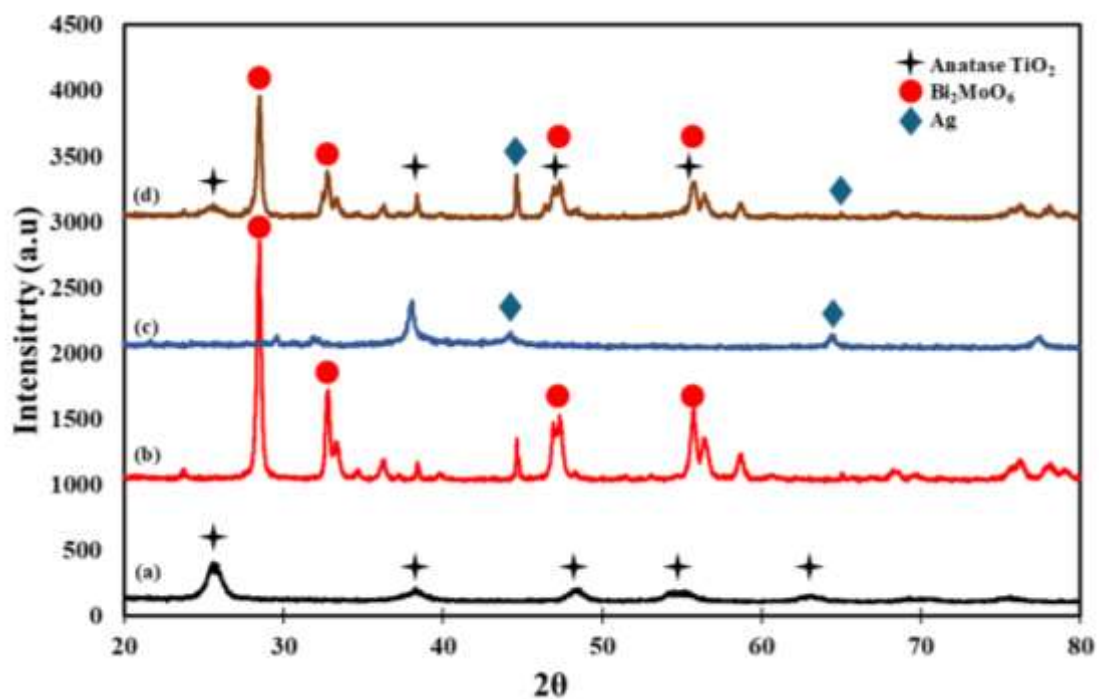
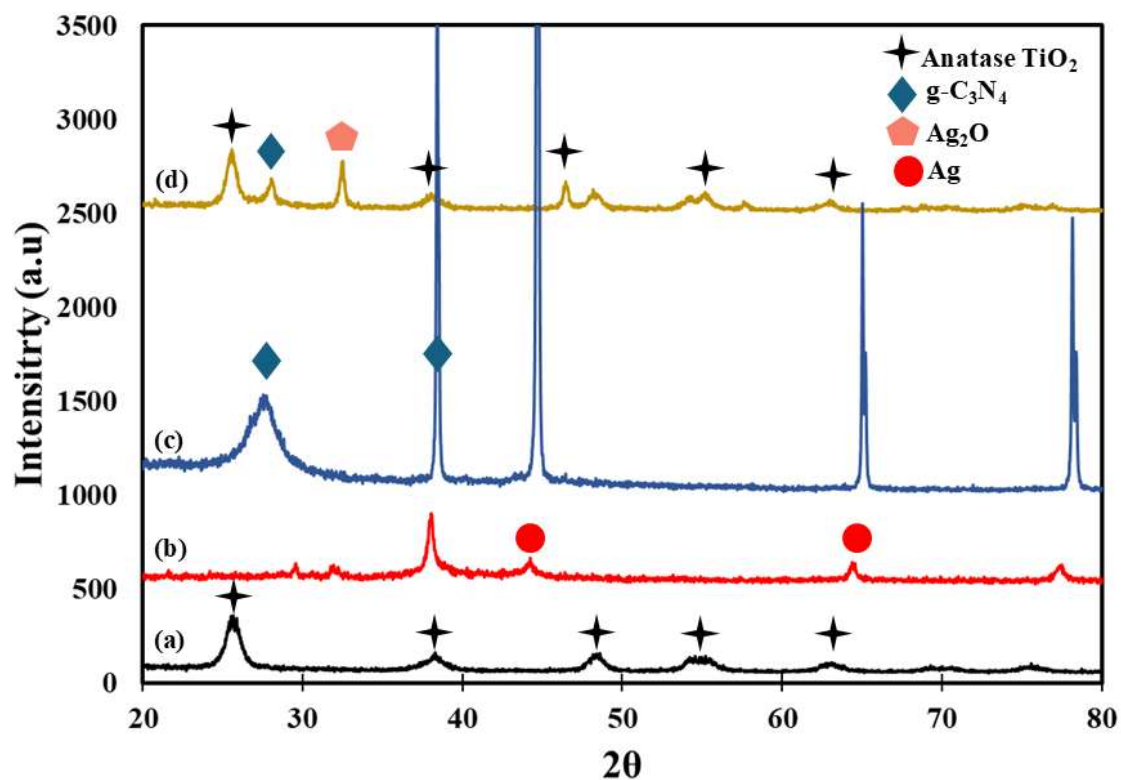


Figure 4-3: XRD spectra of (a) TiO<sub>2</sub>, (b) AgNP, (c) g-C<sub>3</sub>N<sub>4</sub> and (d) AGCNT NC



### 4.2.3 SEM-EDX analysis

To complement the structural insights provided by XRD analyses, further morphological and elemental characterization was carried out via SEM-EDX. This characterization step is essential to visually verify the successful incorporation of each component in the ternary heterostructures of ABMOT and AGCNT matrices. SEM images were acquired to observe morphological changes resulting from heterojunction formation in ABMOT and AGCNT ternary NC matrices, as shown in Figure 4.4. Moreover, elemental mapping aids in evaluating the homogeneity of ternary NC architecture. For Ag/TiO<sub>2</sub>-based composites, morphology plays an important role in light scattering, active site accessibility, and the formation of effective interfacial contacts between the components (Ahmed et al., 2025; Khan et al., 2019).

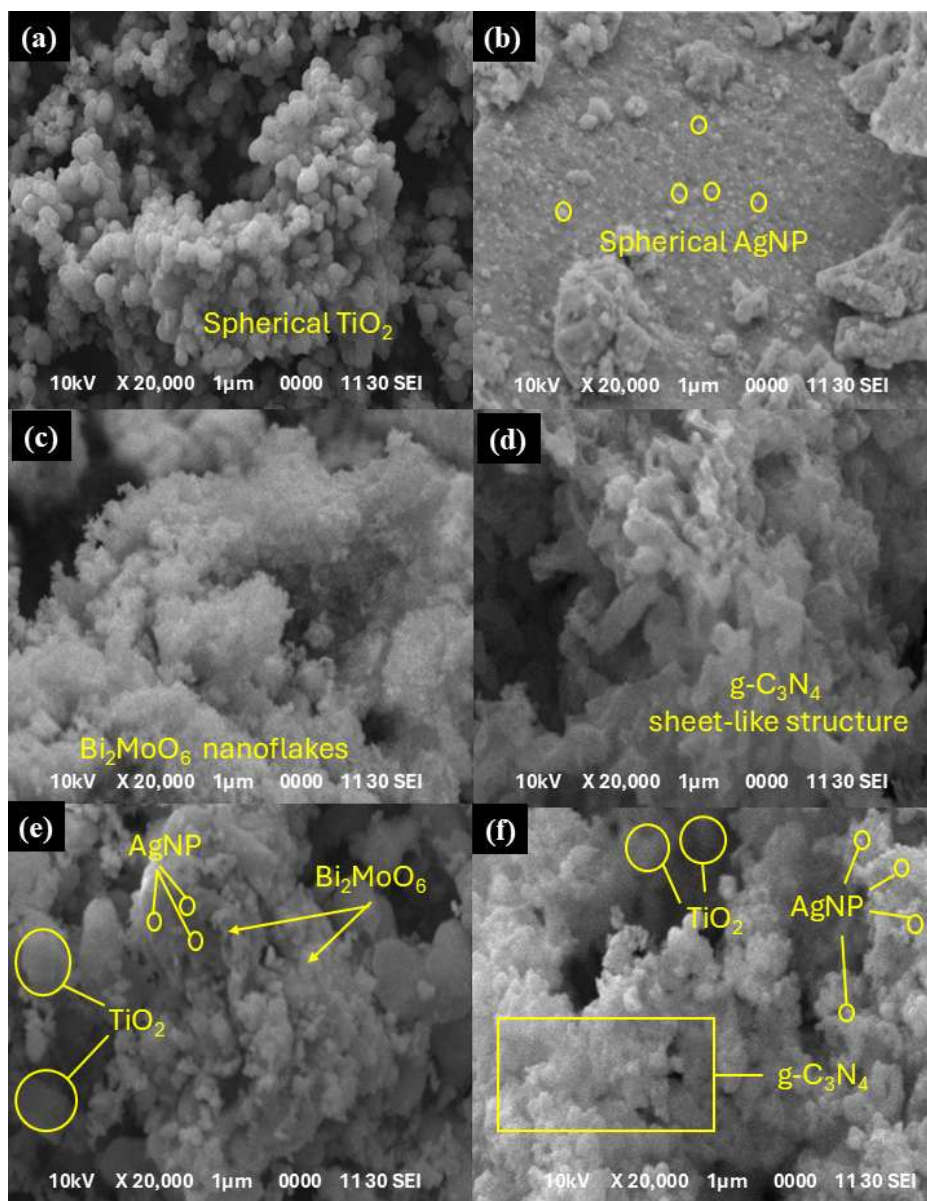
The surface morphology of pristine TiO<sub>2</sub> synthesized via the sol-gel method exhibits smooth, spherical NPs that tend to agglomerate (Figure 4.4a), a common feature in nano-TiO<sub>2</sub> due to its high surface energy, van der Waals forces and hydrogen bonding, which drive the NPs to cluster together (Ahmad et al., 2021; Rachmaniar et al., 2024). Like pristine TiO<sub>2</sub>, POME-derived AgNPs also form spherical NPs that tend to agglomerate (Figure 4.4b). The distribution of these AgNPs is largely influenced by bioactive molecules, as evidenced by the TPC and TFC analyses, where polyphenols and flavonoids act as capping agents to stabilize the aggregates. Similar results were also reported by Gan et al. (2012), who observed spherical POME-derived AuNPs with uniformly distributed aggregates. More recently, another study also reported that POME-derived AgNPs exhibited evenly distributed spherical NPs, stabilized by specific bioactive compounds such as ascorbic acid, gallic acid, and vanillic acid, which facilitated controlled particle size and shape (Alierio et al., 2025).

On the contrary, Bi<sub>2</sub>MoO<sub>6</sub> formed very distinctive morphologies. Bi<sub>2</sub>MoO<sub>6</sub> displayed nanoflake-like morphology, with thin plate-like structures that overlap each other (Figure 4.4c). Similar morphology of green hydrothermal synthesized perovskite Bi<sub>2</sub>WO<sub>6</sub> was observed in a study by Kanwal et al., (2023). In their study, the perovskite nanomaterial was synthesized with the aid of *Azadirachta indica* leaves which facilitated controlled nucleation and prevented uncontrolled aggregation of the growing crystallites, thus allowing for the formation of plate-like nanostructures that aggregate to have a flower-like aggregate (Kanwal et al., 2023). In addition to forming stable AgNPs, the current study further highlights the ability of POME to be used in more complex synthesis systems, such as in the formation of perovskite-based materials. Next, g-C<sub>3</sub>N<sub>4</sub> was observed to form sheet-like

morphology with loosely stacked layers and irregular edges (Figure 4.4d). Due to thermal polymerization, g-C<sub>3</sub>N<sub>4</sub> forms nanosheets that are composed of s-triazine units. The formation of these stacked sheets validates the XRD from peaks of (002) plane attributed to s-triazine unit stacking, hence proving the succession of g-C<sub>3</sub>N<sub>4</sub> formation.

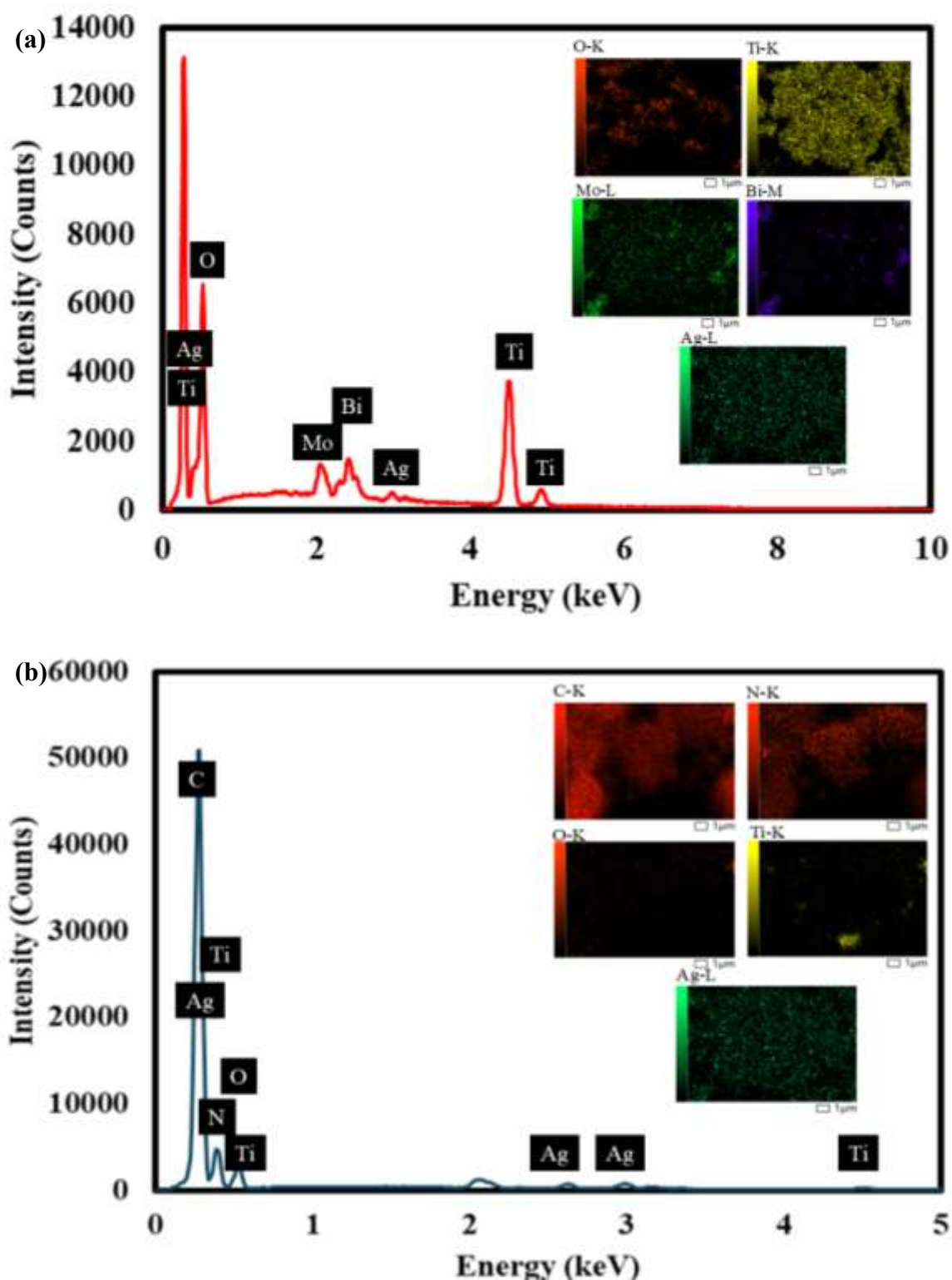
In the ternary systems ABMOT and AGCNT, the formation of NCs resulted in a uniform dispersion of TiO<sub>2</sub> and AgNPs, which were well anchored to the surface of Bi<sub>2</sub>MoO<sub>6</sub> and g-C<sub>3</sub>N<sub>4</sub> (Figure 4.4e and 4.4f). The ternary composites appeared more compact and homogeneous, suggesting the successful assembly of components through microwave-assisted routes. This method provides a favorable environment for uniform dispersion and anchoring of the components by directly interacting with polar molecules (Matias et al., 2023). This enables an even removal of dispersants and allows for consistent nucleation throughout the system (Liu et al., 2020).

**Figure 4–4: SEM micrograph of (a) TiO<sub>2</sub>, (b) AgNP, (c) Bi<sub>2</sub>MoO<sub>6</sub>, (d) g-C<sub>3</sub>N<sub>4</sub>, (e) ABMOT, and (f) AGCNT.**



Further validation by EDX elemental mapping confirmed the uniform distribution of each element across the surfaces of ABMOT and AGCNT (Figure 4.5). Elemental mapping of ABMOT revealed a Ti to Bi and Mo ratio of approximately 1:0.3, with Ag forming around 2.97% by mass as shown in Table 4.2. In the case of AGCNT, although the C and N to Ti composition was expected to be approximately equal based on the 1:1 synthesis ratio, EDX mapping revealed actual ratios of 55.87%, 30.96%, and 1.88%, respectively (Table 4.3). The mass % of Ag in AGCNT was 4.89%.

Figure 4–5: EDX analysis and elemental mapping of (a) ABMOT and (b) AGCNT



These data support the XRD findings of successful material integration. Nevertheless, it is worth noting that the overall composition and distribution of the elements may not directly translate into the mass percentage of elements detected through the EDX

mapping due to the possibility of surface coverage of different elements within the ternary system's matrix (Gui et al., 2023).

**Table 4–2: Elemental composition of ABMOT and AGCNT NCs**

Elements	Mass (%)	
	ABMOT	AGCNT
Ti	52.00±0.26	1.88±0.05
O	30.23±0.13	8.39±0.06
Ag	2.97±0.08	4.89±0.07
Bi	10.87±0.16	-
Mo	3.93±0.08	-
C	-	53.87±0.04
N	-	30.96±0.13

Note: “-” symbolizes elements that are not present in the composition of the material

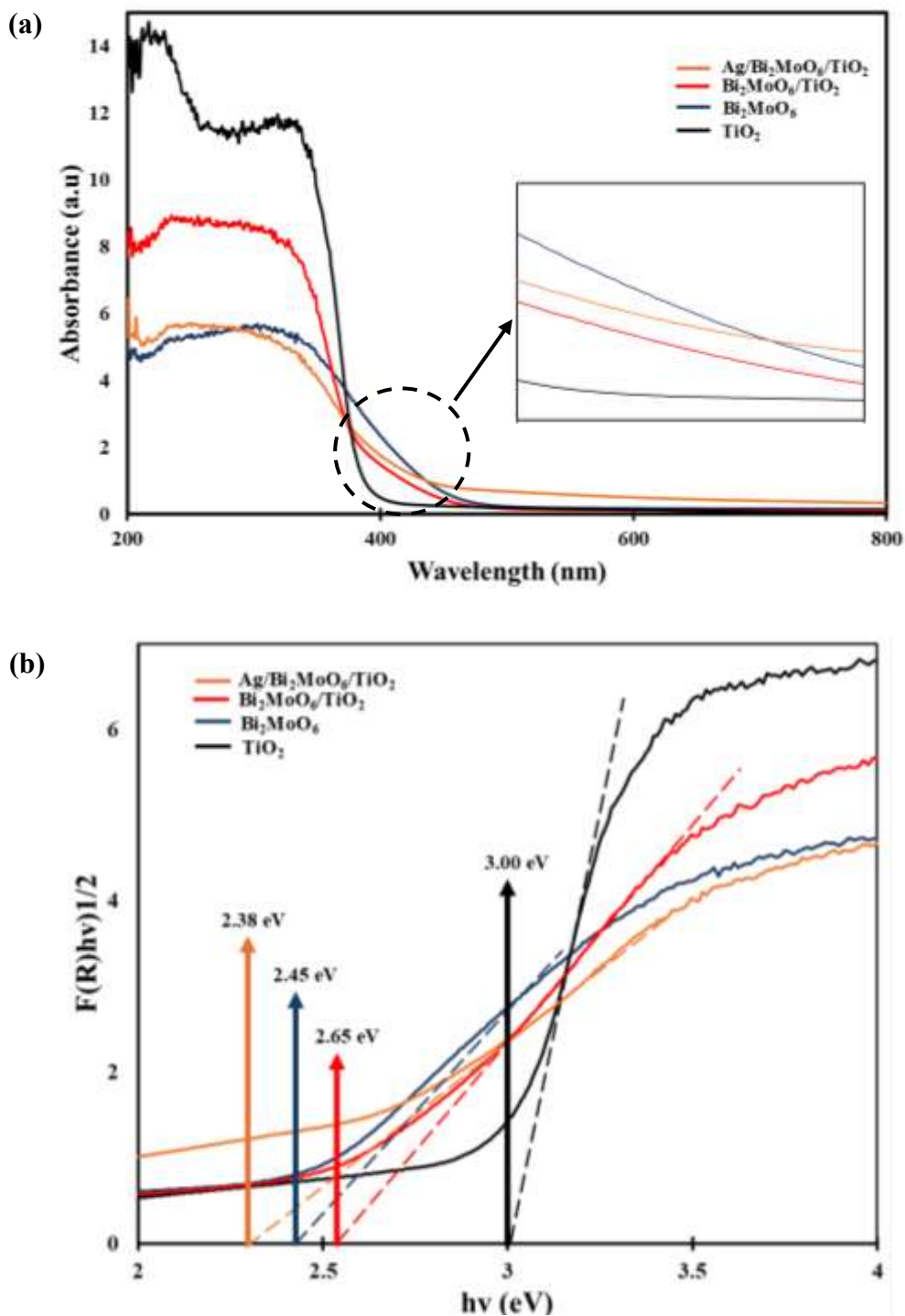
#### 4.2.4 Spectral analysis of UV-Vis DRS

The optical absorption behavior of the synthesized ABMOT and AGCNT NCs was evaluated using UV-Vis spectroscopy and Kubelka-Munk plot analysis, as shown in Figures 4.6 and 4.7. These results provide valuable insights into the materials' light harvesting capability and the modifications in their electronic structures resulting from heterojunction formation and the incorporation AgNPs.

In the ABMOT system (Figure 4.6a and 4.6b), pure TiO<sub>2</sub> showed a sharp absorption edge at around 400 nm with a bandgap of 3.00 eV, confirming that it mainly absorbs UV light. In contrast, Bi<sub>2</sub>MoO<sub>6</sub> demonstrated broader absorption up to about 470 nm with a narrower bandgap of 2.45 eV, which aligns with its known ability to absorb visible light. The Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> composite displayed improved absorption between 300 and 450 nm, indicating effective heterojunction formation that enhances charge separation and broadens the absorption spectrum. Notably, the Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> NC had the highest absorption intensity, with a broad band extending beyond 500 nm. This improvement is mainly due to

the localized SPR effect of AgNPs, which strengthens local electromagnetic fields and extends light absorption into the visible range (Ahmadi et al., 2023; Yin et al., 2021).

Figure 4–6: UV-DRS spectra and (b) Kubelka-Munk plot of Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>, Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and its individual components



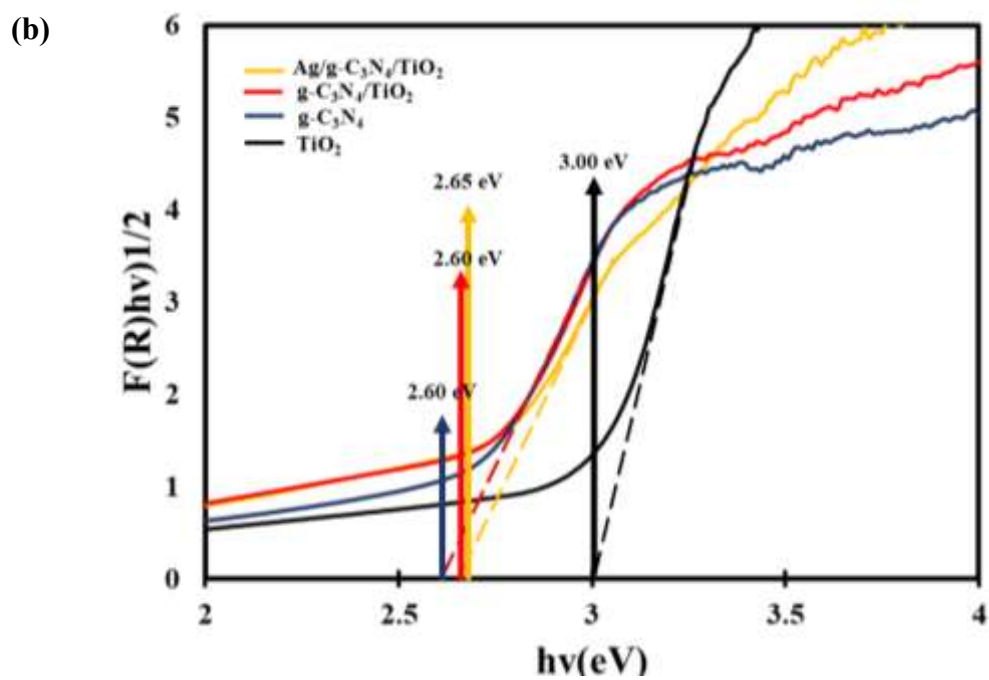
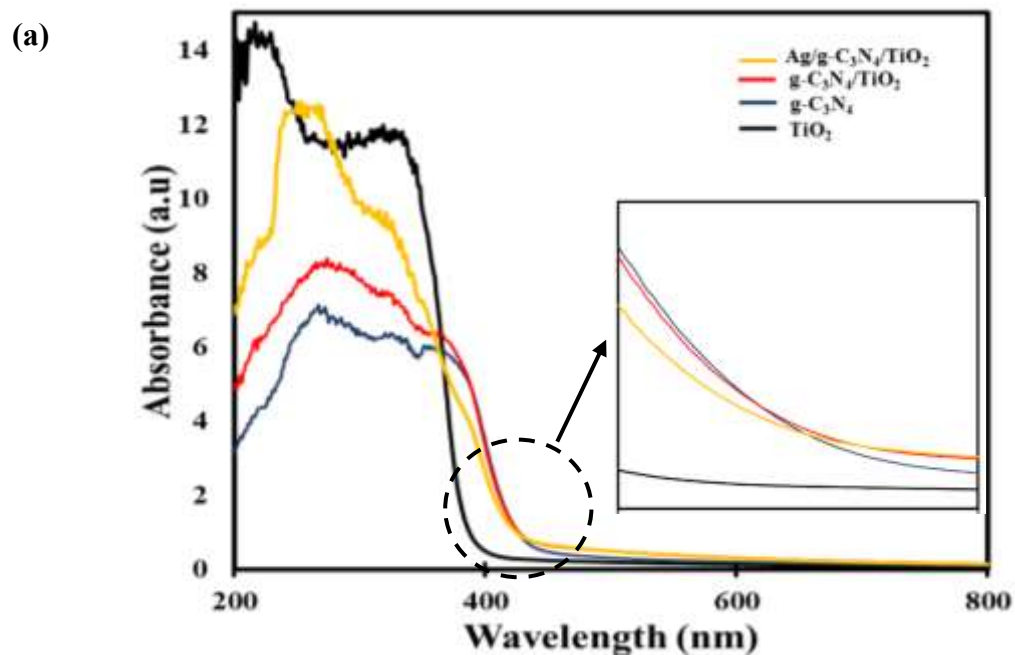
In the AGCNT system (Figures 4.7a and 4.7b), pure TiO<sub>2</sub> maintained its typical UV absorption, while g-C<sub>3</sub>N<sub>4</sub> showed broad absorption up to around 450 nm, reflecting its  $\pi$ - $\pi^*$  electronic transitions and a measured bandgap of 2.60 eV. The TiO<sub>2</sub>/g-C<sub>3</sub>N<sub>4</sub> heterojunction (Figure 4.7) showed further improved absorption across the 300–450 nm range, confirming

a strong interfacial interaction and better light-harvesting ability. With the incorporation of AgNPs, the Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> (Figure 4.7) composite exhibited the highest overall absorption, particularly enhanced in the visible range due to the SPR effect of Ag. The slight increase in the apparent bandgap to 2.65 eV is likely attributed to the minor changes in the electronic environment and optical transitions induced by Ag, rather than intrinsic shifts in the semiconductor's band structure (Sui et al., 2020).

The slope of the Kubelka-Munk plot relates directly to the material's absorption coefficient and the nature of the electronic transitions. In both ABMOT and AGCNT composites, the addition of AgNPs produced steeper slopes than those observed in their respective heterojunctions without Ag. This suggests an enhanced absorption coefficient, indicating a greater probability of photon absorption and electronic transitions within the material (Volpp, 2023). The steeper slope also indicates improved crystallinity and fewer defect states within the composites, which enhance light-matter interactions (Tao et al., 2021). In the ABMOT system, the Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> heterojunction displayed a steeper slope compared to pure Bi<sub>2</sub>MoO<sub>6</sub>, suggesting that coupling with TiO<sub>2</sub> increased the overall electronic transition probabilities by providing more effective charge transfer pathways. Conversely, in the AGCNT system, the g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> composite exhibited a slope comparable to that of g-C<sub>3</sub>N<sub>4</sub> alone; however, the incorporation of Ag further steepened the slope, highlighting Ag's role in promoting more efficient light absorption through plasmonic effects.

Overall, both the absorption spectra and Kubelka-Munk plot analyses confirmed that forming heterojunctions and incorporating AgNPs work together to enhance the optical properties of the ABMOT and AGCNT NCs. These improvements are expected to significantly boost their photocatalytic performance by enabling broader light absorption and more effective charge carrier generation

Figure 4-7: UV-DRS spectra and (b) Kubelka-Munk plot of Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>, g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub> and its individual components



## 4.3 Photocatalytic performance of ABMOT and AGCNT

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### 4.3.1 Parameters affecting the removal of OTC

#### 4.3.1.1 Effect of catalyst dosage

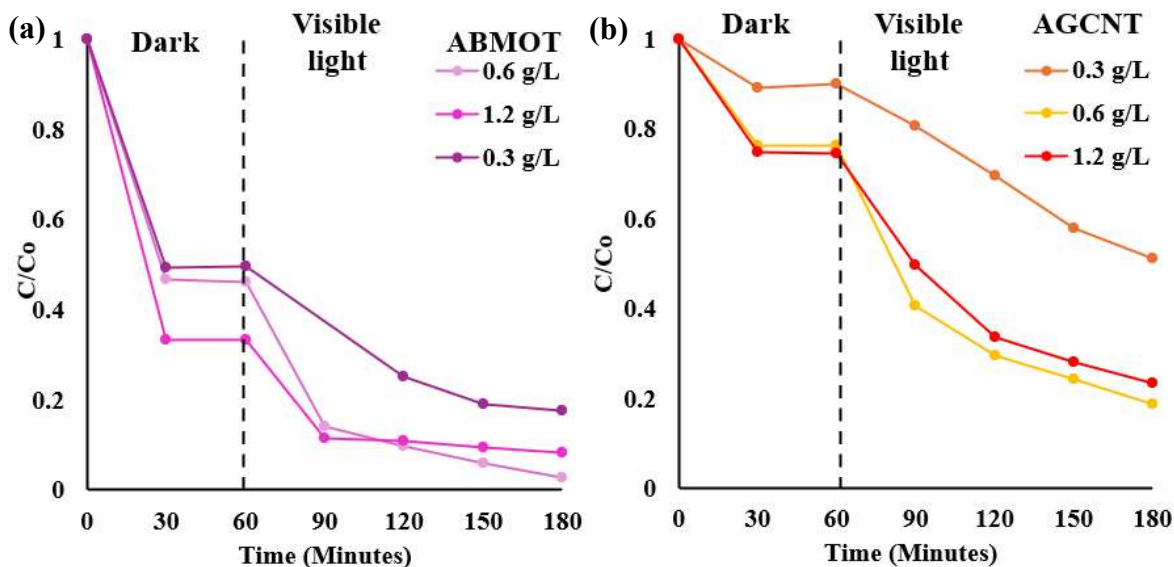
Further investigation was conducted to determine the optimal dosage for ABMOT-AGCNT ternary NCs to degrade OTC under a xenon lamp as a visible-light source simulating sunlight for over 120 minutes (Figure 4.8). Varying catalyst loadings of 0.3 g/L, 0.6 g/L and 1.2 g/L were tested to evaluate the effect of catalyst dosages on the degradation process (Figure 4.8a and b). For both ternary composites, increasing the catalyst dosage from 0.3 g/L to 0.6 g/L enhanced the photocatalytic degradation of OTC. Specifically, OTC degradation increased from 88.3% to 97.2% for ABMOT and 48.7% to 81.3% for AGCNT. However, when the dosage increased to 1.2 g/L, a slight reduction in OTC degradation was observed for ABMOT and AGCNT at 91.8% and 76.6%, respectively.

This indicates that exceeding the optimal catalyst loading can saturate the photocatalyst surface, thereby reducing the overall removal efficiency. Although increasing the photocatalyst dosage up to a certain level can increase the number of reactive sites for photocatalysis, excessive loading could diminish the total surface area available for pollutant adsorption (Khader et al., 2024). Increasing the catalyst dosage beyond its optimal level can also increase opacity and light scattering, which in turn reduce the number of photons that can penetrate the catalyst surface, a crucial factor in initiating photocatalytic reactions (Santana et al., 2023).

Notably, ABMOT exhibited stronger adsorption than AGCNT during the dark phase, likely due to more abundant hydroxyl groups and the layered structure of  $\text{Bi}_2\text{MoO}_6$  as observed in FTIR and SEM analysis. This results in an abundance of surface-active sites and facilitates stronger interactions with OTC molecules prior to visible light irradiation (Duong et al., 2024). In contrast, while AGCNT offers  $\pi$ - $\pi$  interactions with aromatic antibiotic molecules, g- $\text{C}_3\text{N}_4$  is known to have a lower surface area and limited hydroxyl surface density (Muhmood et al., 2024). Hence, the weaker adsorption effect. The stronger initial adsorption of ABMOT increases pollutant concentration near active sites, thus allowing higher degradation of the adsorbed OTC on the surface of the ABMOT NCs. Therefore, the

optimal dosages for both ternary photocatalysts were determined to be 0.6 g/L and were chosen for further study.

**Figure 4–8: Photocatalytic degradation of OTC with different dosages of (a) ABMOT and (b) AGCNT**



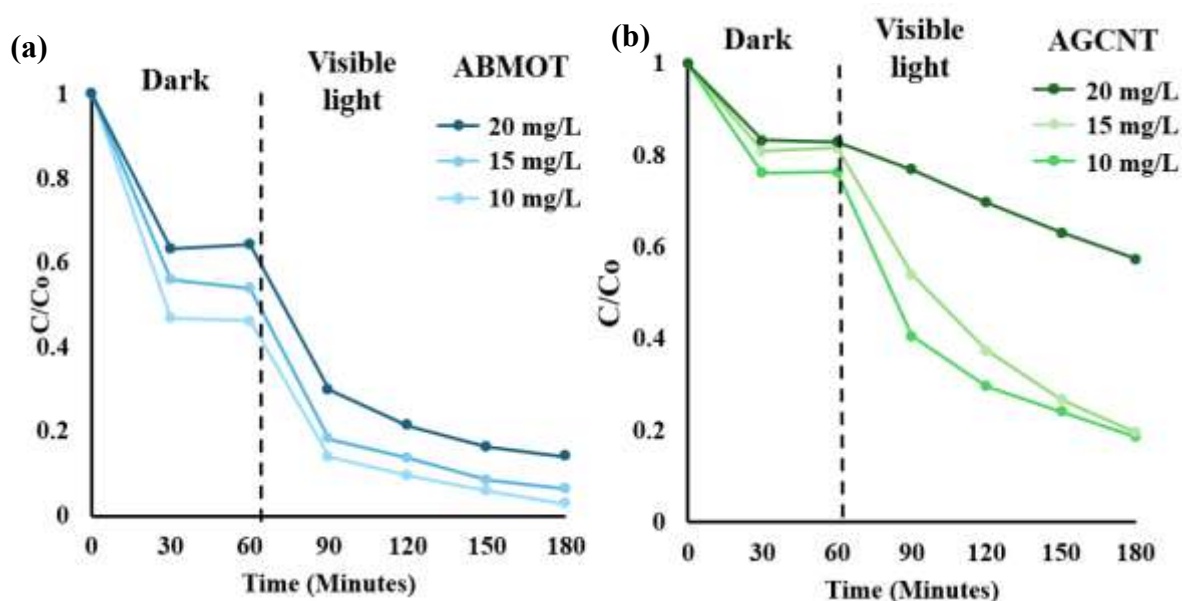
#### 4.3.1.2 Effect of initial concentration

The effect of initial OTC concentration (10–20 mg/L) at a fixed dosage of 0.6 g/L was evaluated for both ternary NCs (Figure 4.9). Minimal changes were observed when the concentration increased from 10 to 15 mg/L, with ABMOT decreasing slightly from 97.2% to 93.5% and AGCNT from 81.3% to 80.3%. However, at 20 mg/L, AGCNT showed a marked decline in degradation efficiency, dropping sharply to 42.5%, contrary to ABMOT, which only decreased to 85.9%. The reduced photodegradation rate with increasing OTC concentrations could be due to excessive OTC loading on the photocatalyst surface, coupled with the accumulation of intermediates that block active sites and thereby suppress the formation of reactive oxygen species (ROS). A similar finding was also observed in a previous study involving the degradation of tetracycline hydrochloride by using Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>-based photocatalyst, whereby the degradation efficiency declined from ~87.5% to 20.0% with the increasing initial concentration from 10 mg/L to 30 mg/L (Qi et al., 2025).

Furthermore, the dark adsorption phase also revealed a notable difference between the two photocatalysts at varying OTC concentrations. For ABMOT, a significant decrease

in the degradation rate was observed, particularly when the concentration was increased to 20 mg/L, further demonstrating that the layered structure of  $\text{Bi}_2\text{MoO}_6$  and its higher hydroxyl density enable strong adsorption, allowing better removal of OTC even at high concentrations. Conversely, AGCNT shows weaker adsorption trends with increasing concentration, indicating the limited adsorption capacity. This is most evident at a concentration of 20 mg/L, where rapid saturation of surface-active sites restricts pollutant-catalyst interaction. Hence, the optimal initial concentration for the degradation of OTC was fixed as 10 mg/L.

**Figure 4–9: Photocatalytic degradation of OTC with different initial concentrations for (a) ABMOT and (b) AGCNT**

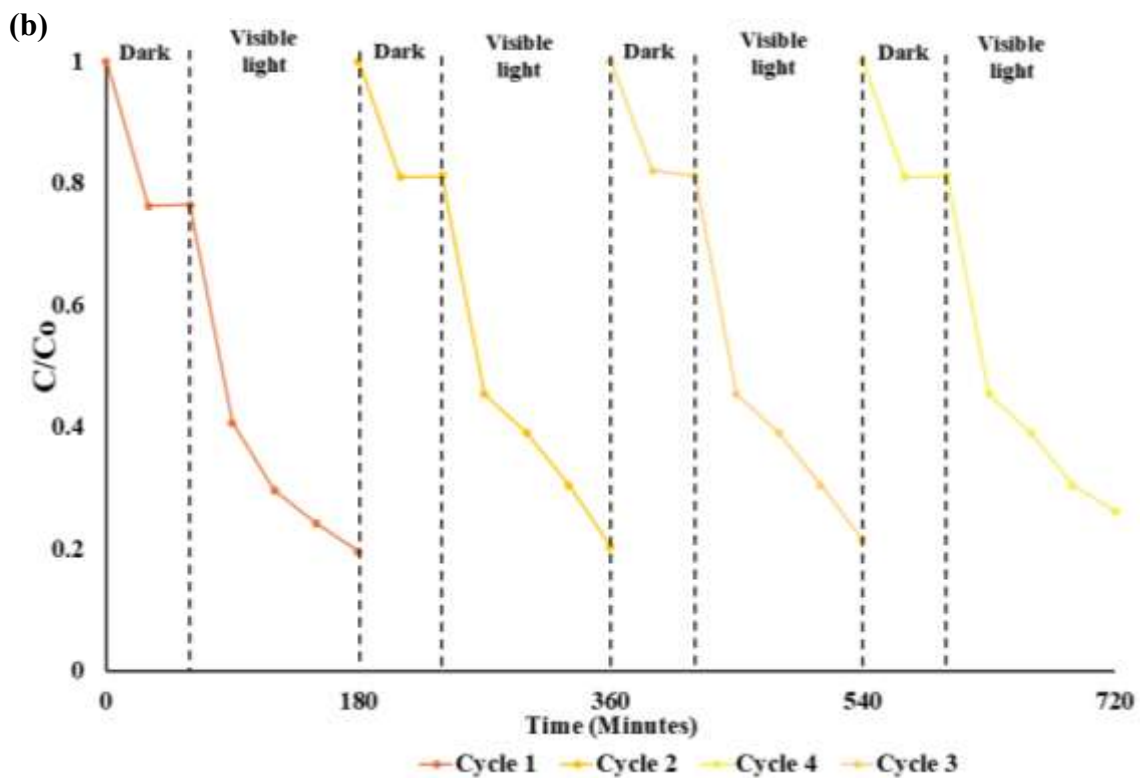
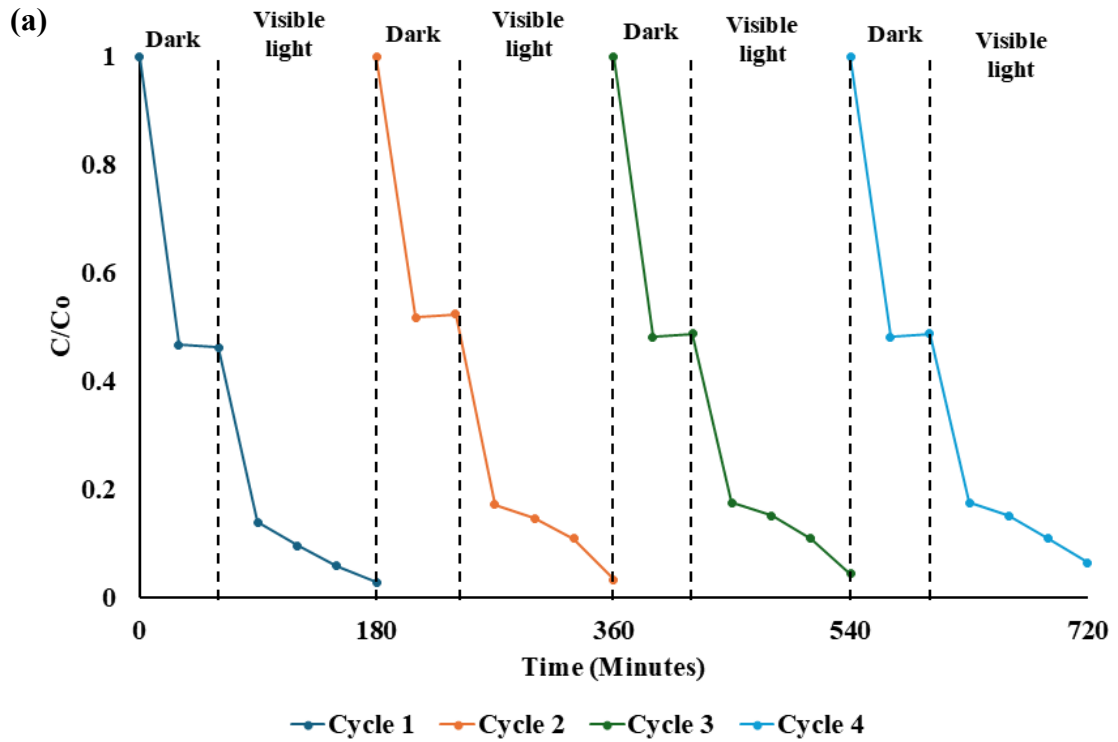


#### 4.3.1.3 Recyclability potential of ABMOT and AGCNT

Recyclability study was conducted to determine the efficiency of the ABMOT and AGCNT ternary NCs in degrading OTC over 4 consecutive cycles (Figure 4.10). The stability of a photocatalyst is of utmost importance as it determines the scalability for larger industrial applications that require cost-effective processes. The recyclability study was conducted under optimized conditions, with ABMOT and AGCNT dosages of 0.6 g/L to treat 10 mg/L OTC for 120 minutes under visible-light irradiation. After 4 consecutive cycles, the degradation efficiency of AGCNT decreased slightly from 80.3% to 73.7% (Figure 4.10b) while ABMOT showed a minor reduction from 97.2% to 93.5% (Figure

4.10a). These results highlight the stability and efficiency of ternary ABMOT and AGCNT NCs.

**Figure 4–10: Photocatalytic degradation of OTC by (a) ABMOT and (b) AGCNT after four cycles**



#### 4.4 Antibacterial properties of AGCNT and ABMOT

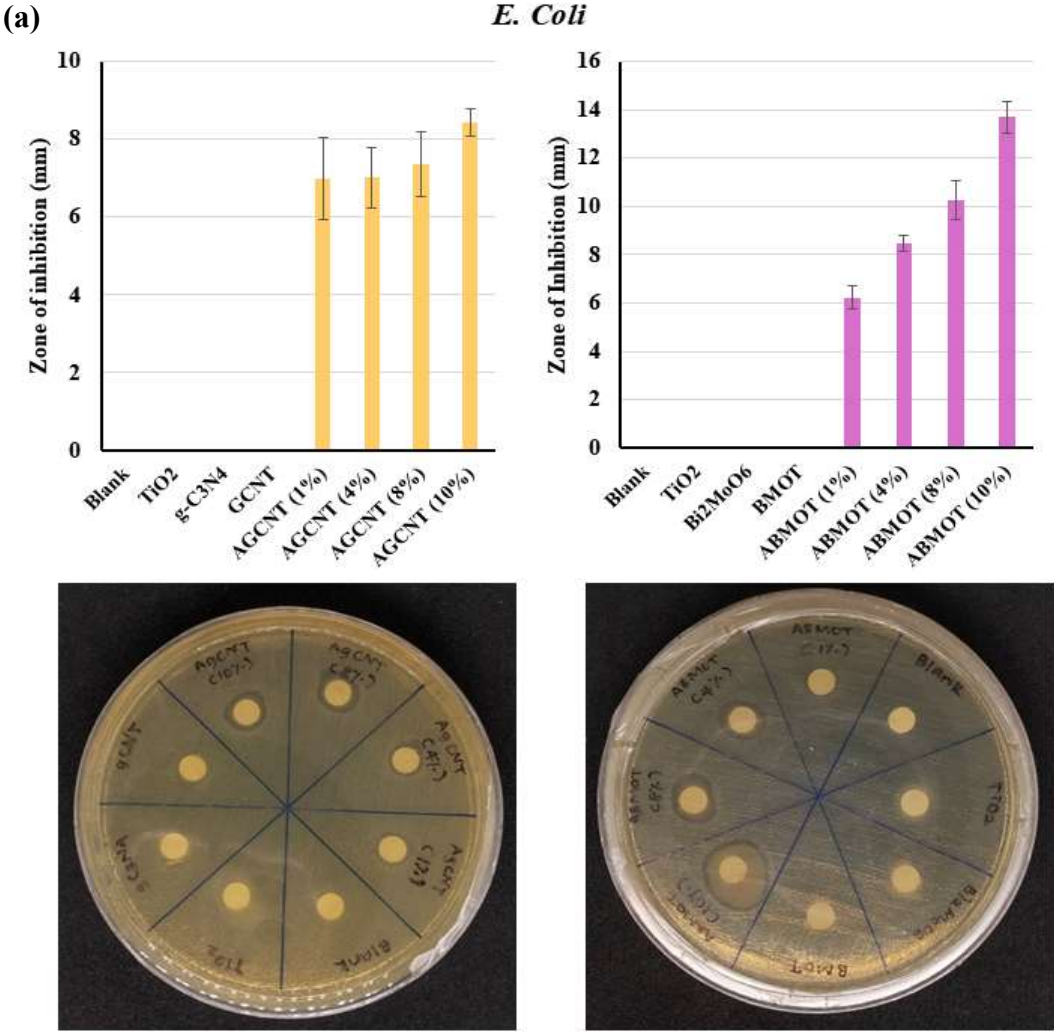
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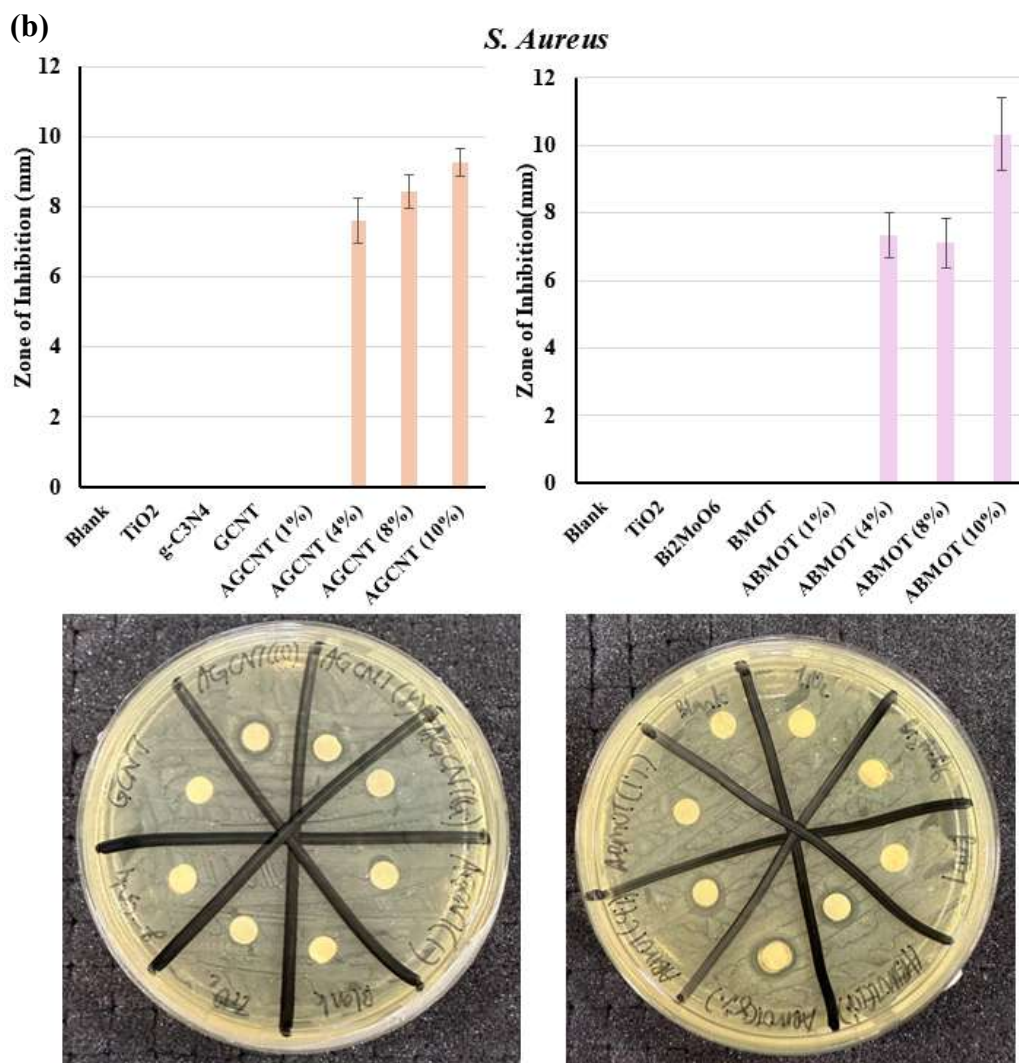
The antibacterial properties of both ternary ABMOT and AGCNT were tested against gram-negative *E. coli* and gram-positive *S. aureus* via the disk diffusion method. Antibacterial activity was evaluated concerning the zone of inhibition (ZOI) induced by the photocatalyst (Figure 4.11). To evaluate the antibacterial activity, ABMOT and AGCNT were compared with their unary and binary counterparts at a fixed concentration. It was found that for all unary and binary samples of TiO<sub>2</sub>, Bi<sub>2</sub>MoO<sub>6</sub>, g-C<sub>3</sub>N<sub>4</sub>, Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub> and g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>, no measurable clear ZOIs were found, suggesting their negligible antibacterial properties. These results corroborate with past studies utilizing NCs containing TiO<sub>2</sub>, Bi<sub>2</sub>MoO<sub>6</sub> and g-C<sub>3</sub>N<sub>4</sub> without the presence of other metal-based dopants or oxides in the ternary photocatalyst matrix (Ma et al., 2024; Vignesh et al., 2018). The main reason for this observation is due to the non-toxic nature of Ti, Bi and Mo-based oxides. In most cases, the antibacterial activity attributed to Ti, Bi and Mo-based metal oxides lie in the generation of reactive species to cause cell damage. However, in the disc diffusion method, the absence of light significantly dampens the antibacterial activity.

The presence of Ag in ABMOT and AGCNT produced clear ZOIs, confirming that the AgNPs induced antibacterial activity within the composite matrix. Ag is known to exhibit antibacterial properties due to the dissolution of Ag<sup>+</sup> ions, which readily attack cell membranes of microbes (Anees Ahmad et al., 2020). Increasing the Ag dosage from 1% to 10% increased the ZOI, with ABMOT increasing from  $6.23 \pm 1.05$  mm to  $13.68 \pm 0.36$  mm against *E. coli*, while AGCNT increased from  $7.00 \pm 0.48$  mm to  $8.43 \pm 0.63$  mm (Figure 4.11a). In the case of *S. aureus*, increasing Ag content in ABMOT yielded similar results with an increase in ZOIs from  $7.35 \pm 0.66$  mm to  $10.31 \pm 1.07$  mm, while AGCNT showed an increase from  $7.6 \pm 0.65$  mm to  $9.26 \pm 0.40$  mm (Figure 4.11b).

These results suggest that the antibacterial activity is predominantly driven by AgNPs, which likely enhance oxidative stress and disrupt bacterial cell membranes, thereby disrupting essential cellular functions and leading to cell death (Bahadur et al., 2016; Chen et al., 2013). Additionally, the antibacterial modes of action between the two ternary composites differ, as reflected in their varying ZOIs. Compared to AGCNT, ABMOT exhibited stronger antibacterial activity, which could be due to the higher rate of Ag dissolution resulting from Ag dispersion in the NC matrix.

**Figure 4–11: Antibacterial activity of ABMOT and AGCNT against (a) gram-negative *E. coli* and (b) gram-positive *S. aureus* bacteria.**

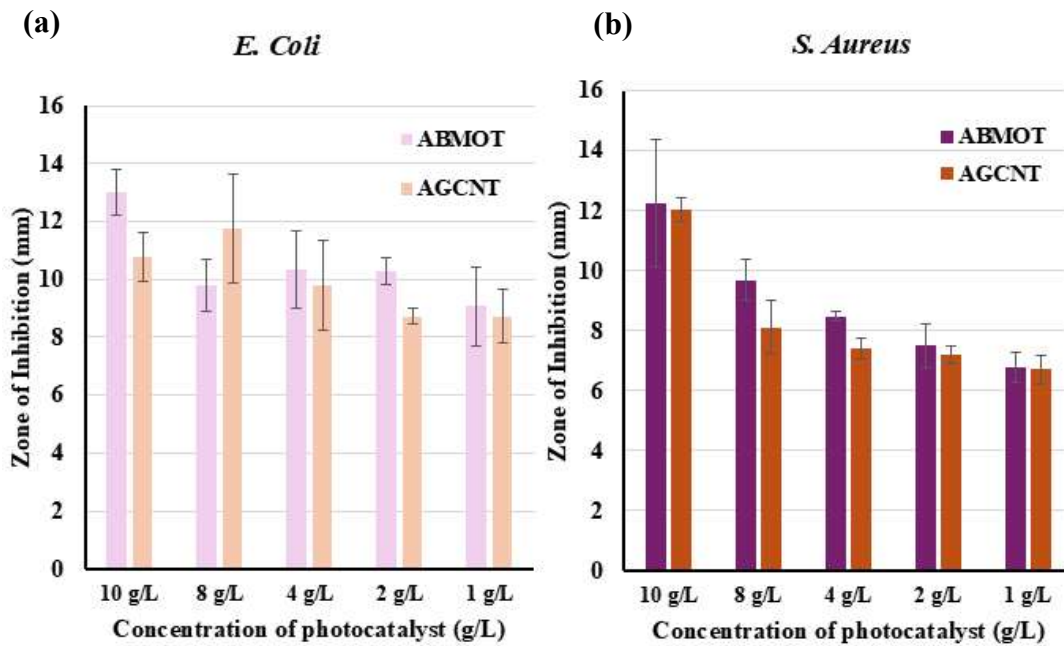




Moreover, a study on the effect of catalyst dosage on antibacterial activity was also conducted to determine the optimal dosage needed for bacterial inhibition (Figure 4.12). For both ABMOT and AGCNT samples, an increase in the catalyst dosage led to larger ZOIs. Nevertheless, in the case of AGCNT, the ZOI against gram-negative *E. coli* began to decline once the dosage exceeded 8 g/L (Figure 4.12). A similar trend was observed for ABMOT when the dosage exceeded 6 g/L, though the ZOI increased again up to 10 g/L. This could be attributed to the saturation point being reached, which limited the interactions between nanomaterials and bacterial cells. On the contrary, when similar conditions were conducted with gram-positive *S. aureus*, a proportional relationship between concentration and ZOIs was observed. This is because higher concentration of nanomaterials is needed for a more effective disruption of bacterial activities, as gram-negative bacteria possess a thicker peptidoglycan layer compared to gram-positive bacteria (Dhanalekshmi et al., 2013). From a collective perspective, the overall photocatalytic performance and antibacterial activity of

both ABMOT and AGCNT ternary composites have been shown to be promising. Therefore, application of these ternary NCs was applied to a membrane matrix to study how they perform as photocatalytic membranes.

Figure 4–12: Effect of catalyst dosage on antibacterial activity by ABMOT and AGCNT against (a) gram-negative *E. coli* and (b) gram-positive *S. aureus* bacteria.



## 4.5 Immobilization of ABMOT and AGCNT into PAN membrane

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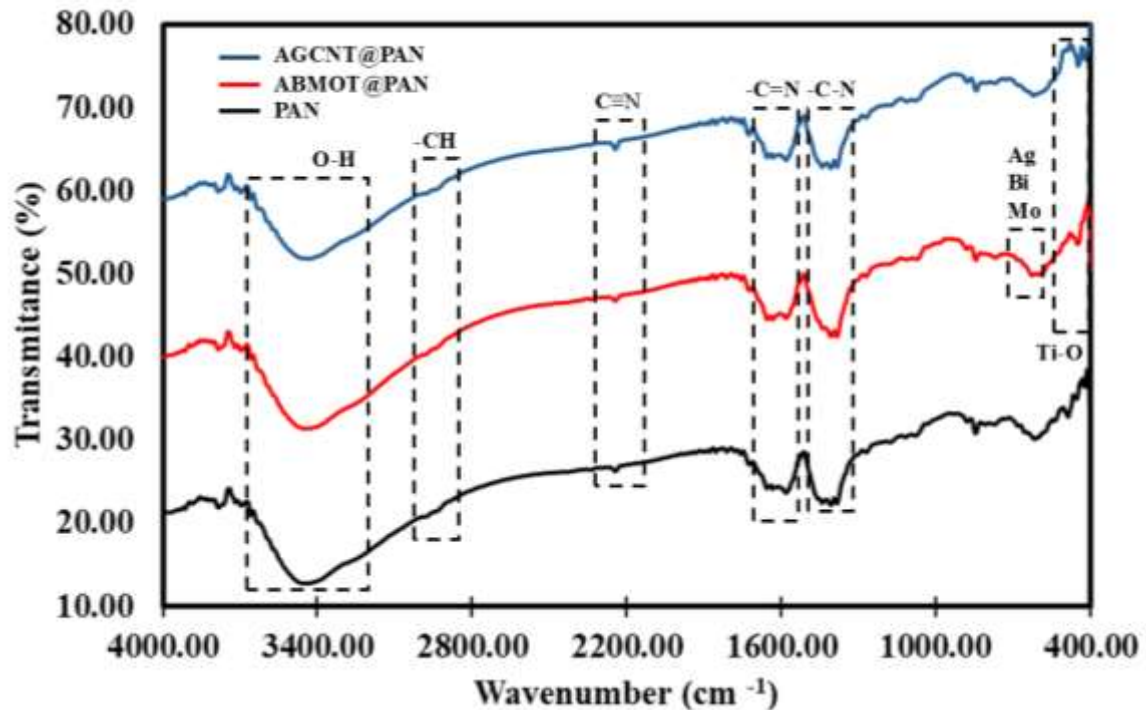
### 4.5.1 Characterization of membranes

#### 4.5.1.1 FTIR spectra

To investigate the physicochemical changes upon NC incorporation, FTIR analysis was employed to evaluate the chemical interactions and functional group retentions of NCs within the membrane matrix. Comparison between bare PAN membranes, ABMOT@PAN and AGCNT@PAN membranes were carried out within the range of 400 – 4000  $\text{cm}^{-1}$  as observed in Figure 4.13. All samples displayed characteristic PAN absorption bands (Figure 4.13a), characteristic bands at  $\sim 2242 \text{ cm}^{-1}$  ( $-\text{C}\equiv\text{N}$  stretching),  $\sim 1450 \text{ cm}^{-1}$  (C–H bending), and  $\sim 3400 \text{ cm}^{-1}$  (O–H stretching). Upon integration of ABMOT (Figure 4.13b) and AGCNT NCs (Figure 4.13c), slight shifts and changes in transmittance intensity were observed in the regions around  $1630 \text{ cm}^{-1}$  and  $1000\text{--}700 \text{ cm}^{-1}$ . These alterations suggest interactions between PAN polymer chains and the NCs' surface functionalities (Hartati et al., 2022). For instance, the broad band around  $3400 \text{ cm}^{-1}$ , associated with O–H bending, showed enhanced intensity in the ABMOT@PAN and AGCNT@PAN samples, possibly due to increased surface hydroxylation from the incorporated metal oxides. In the fingerprint region ( $\sim 700\text{--}1000 \text{ cm}^{-1}$ ), additional features may correspond to stretching vibrations of metal–oxygen (M–O) bonds from Bi–O, Mo–O, Ti–O, or Ag–O linkages, further supporting successful NC integration. The C=N peaks associated with s-triazine stacking units in AGCNT@PAN are not clearly distinguishable, as they overlap with the PAN -C=N bending vibrations.

This result confirms chemical interactions between NCs and PAN within the NC membrane matrices, thereby validating the succession of ternary ABMOT and AGCNT NCs within PAN membranes. These modifications are expected to enhance membrane properties such as hydrophilicity and permeability properties of NC membranes (Sahu et al., 2023).

**Figure 4–13: FTIR spectra of (a) PAN, (b) ABMOT@PAN and (c) AGCNT@PAN membranes**



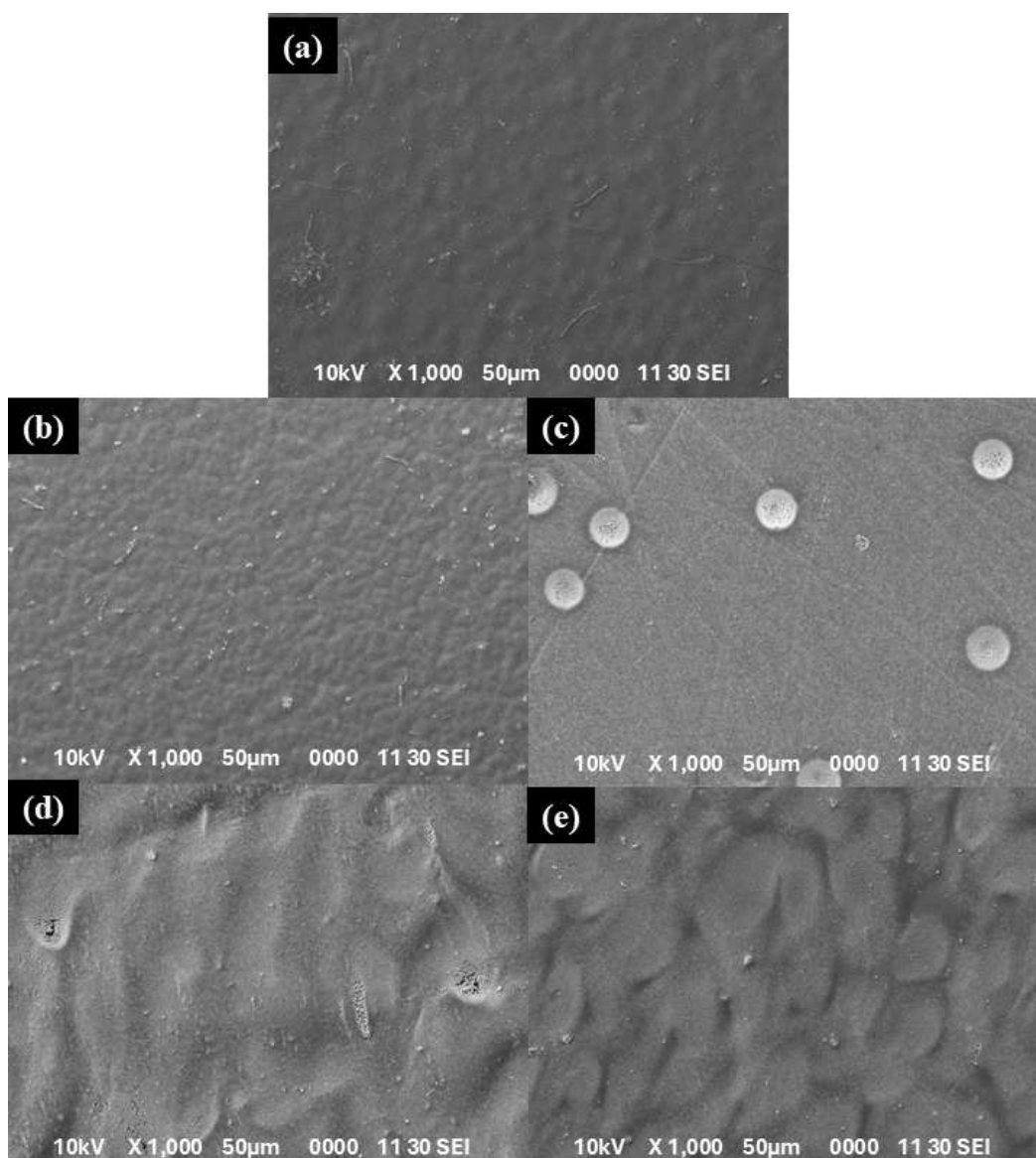
#### 4.5.1.2 SEM-EDX analysis

Further validation of ternary NC incorporation in the PAN membrane was achieved through SEM-EDX analysis (Figure 4.14). The bare PAN membrane exhibited a relatively smooth, featureless surface, typical of pristine PAN prepared via phase inversion, with no visible particulate deposits or irregular surface features (Figure 4.14a). This smoothness indicates the formation of a uniform polymer film in the absence of NC additives (Xie et al., 2023).

The 1 wt% AGCNT@PAN membrane resulted in a noticeably rougher surface with fine granules distributed across it, suggesting that the NC particles were successfully embedded within and partially exposed on the membrane surface (Figure 4.14b). In contrast, the 1 wt% ABMOT@PAN membrane displayed discrete spherical particles and aggregates that were more clearly distinguishable compared to AGCNT@PAN (Figure 4.14c). These protrusions are attributed to the relatively larger particle size and agglomeration tendency of ABMOT, which may cause surface deposition during the casting and solvent exchange stages.

Increasing the NC loading to 2 wt% further intensified the observed surface texturing for both composites. The 2 wt% AGCNT@PAN sample exhibited more pronounced surface undulations and irregular granular coverage, indicating increased NC migration toward the surface (Figure 4.14d). The 2 wt% ABMOT@PAN membrane surfaces appeared even more irregular, with a combination of textured wave-like features and larger exposed particle clusters (Figure 4.14e).

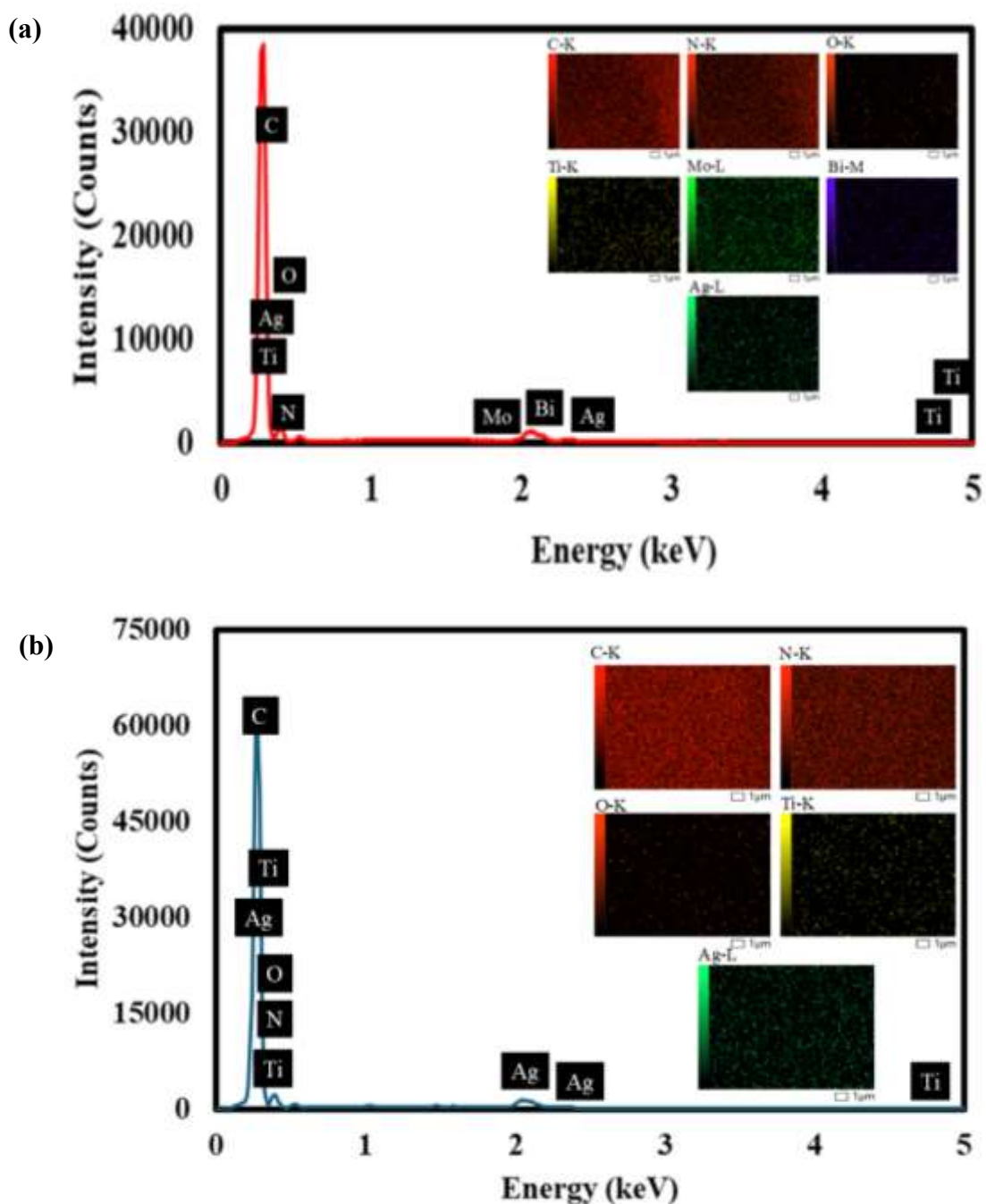
**Figure 4–14: SEM micrograph of (a) PAN, (b) 1% AGCNT@PAN, (c) 1% ABMOT@PAN, (d) 2% AGCNT@PAN and (e) 2% ABMO@PAN**



Furthermore, EDX elemental mapping verified the presence of each element in both ABMOT@PAN and AGCNT@PAN NC membranes (Figure 4.15). The absence of some elements, such as Ti within AGCNT@PAN and Ag/Bi within ABMOT@PAN membranes,

is attributed to the low amount of NCs compared to the mass of PAN in the NC-membrane matrix (Quadri et al., 2023). Nevertheless, trace amounts are still detected in the system, confirming the sequence of NC incorporation. A summary of the elemental composition is shown in Table 4.3.

**Figure 4–15: EDX analysis and elemental mapping of (a) ABMOT@PAN and (b) AGCNT@PAN membranes**



Overall, the SEM-EDX observations confirm that incorporating NCs into PAN membranes alters the surface morphology, with increasing NC content leading to greater

roughness and more pronounced surface features. These changes are more pronounced for ABMOT-loaded membranes compared to AGCNT-loaded membranes, likely due to differences in particle morphology, size distribution, and surface chemistry between the two NC types. The increased surface roughness is expected to enhance surface hydrophilicity and potentially improve photocatalytic activity under light exposure (Vatanpour et al., 2024).

**Table 4–3: Elemental composition of ABMOT@PAN and AGCNT@PAN membranes**

Elements	ABMOT@PAN	AGCNT@PAN
	Mass%	Mass %
Ti	0.24±0.04	Not detected
O	2.92±0.05	2.38±0.04
Ag	Not detected	0.22±0.04
Bi	Not detected	-
Mo	2.45±0.06	-
C	72.45±0.07	74.33 ± 0.05
N	21.94±0.17	23.06±0.15

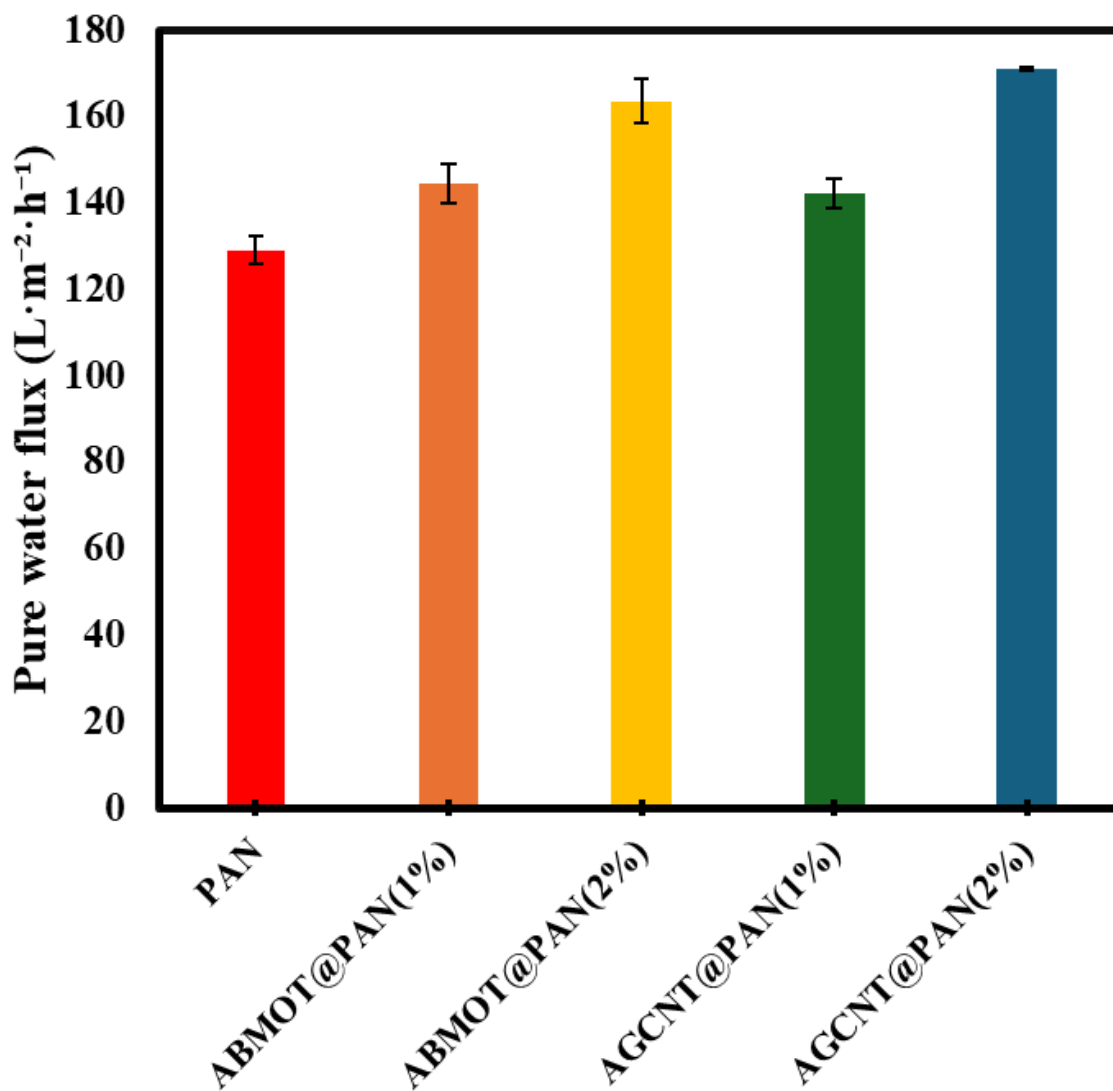
Note: “-“symbolizes elements that are not present in the composition of the material

#### 4.5.2 Permeation test of NC membranes

Pure water flux (PWF) analysis was then carried out to evaluate the permeability of bare PAN and NC-integrated PAN membranes (Figure 4.16). The bare PAN membrane exhibited the lowest flux value of  $129.17 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ , attributable to its relatively smooth and less porous surface morphology observed in SEM images (Figure 4.14a). The incorporation of NCs resulted in a consistent increase in PWF across all tested membranes. With a 1 wt% loading, ABMOT@PAN achieved a flux of  $144.39 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ , whereas AGCNT@PAN recorded  $142.21 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ . Increasing NC loading to 2 wt% further enhanced permeability, with ABMOT@PAN reaching  $163.48 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$  and AGCNT@PAN peaking at  $170.94 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ . The increase in flux with higher NC loading is attributed to the increased surface roughness and hydrophilicity imparted by the NCs, which promote efficient water transport pathways (Ramon & Hoek, 2013). The comparatively higher PWF of AGCNT@PAN (2

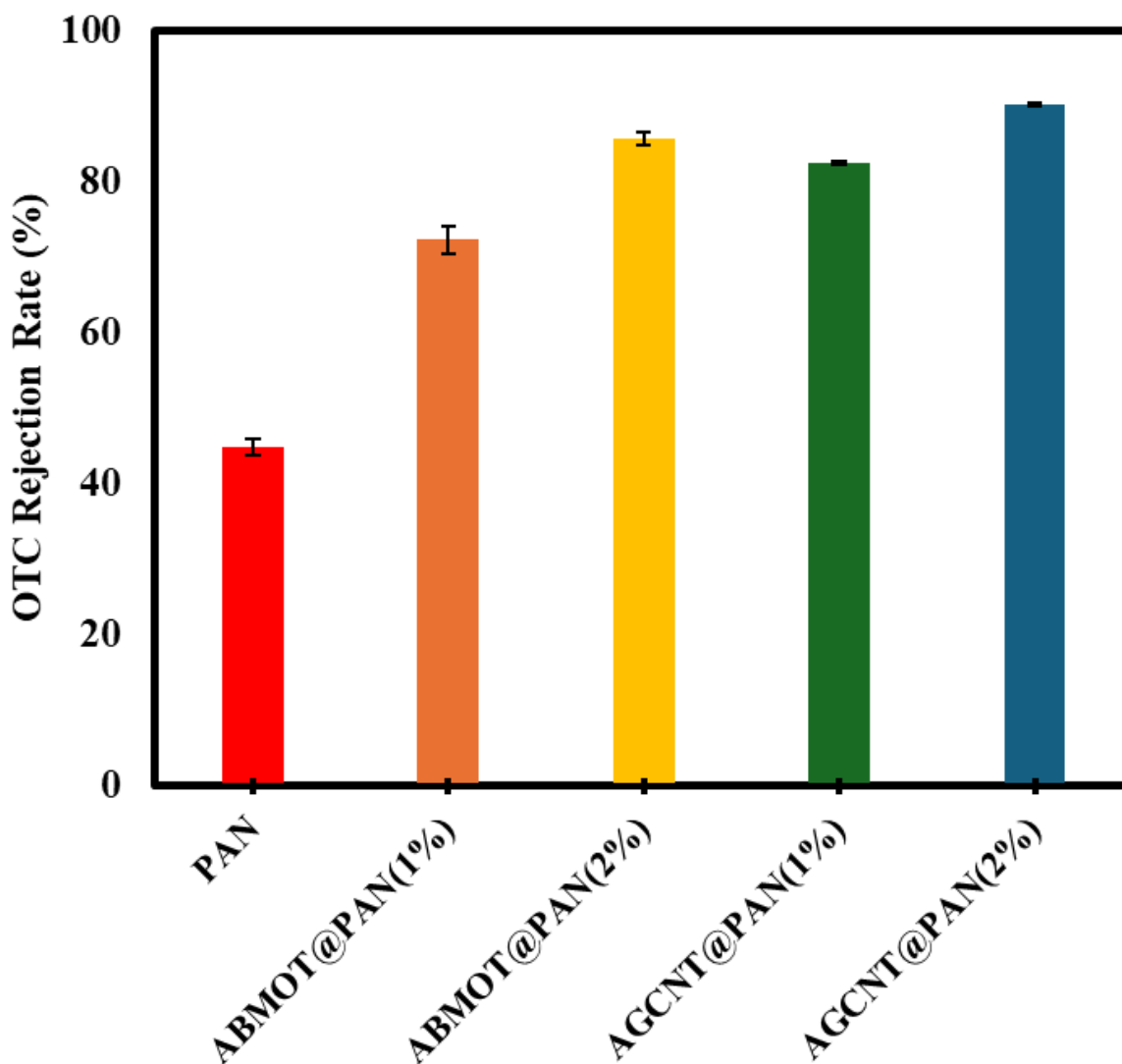
wt%) suggests that the sheet-like morphology of g-C<sub>3</sub>N<sub>4</sub> promotes better water channel formation and interfacial compatibility within the PAN matrix.

**Figure 4–16: Pure water flux of pure PAN, ABMOT@PAN and AGCNT@PAN membranes**



Rejection rate testing (Figure 4.17) revealed that bare PAN had the lowest contaminant removal efficiency (44.88%). The incorporation of NCs improved the rejection performance, with ABMOT@PAN achieving 72.30% at 1 wt% and 85.69% at 2 wt%, while AGCNT@PAN achieved 82.57% and 90.34% at the same loadings. The superior rejection performance of AGCNT-based membranes is attributed to their higher hydrophilicity, greater surface active site density, and improved NC dispersion, which collectively enhance size-exclusion and adsorption mechanisms (Arundhathi et al., 2024; Wu et al., 2022).

Figure 4–17: OTC rejection rate of pure PAN, ABMOT@PAN and AGCNT@PAN membranes



Although the NC@PAN membranes in this study were tested solely under filtration conditions and not for photocatalytic degradation, the superior performance of AGCNT in its powdered form suggests that, if integrated into photocatalytic membrane systems, AGCNT@PAN could potentially achieve higher contaminant removal rates while maintaining favourable permeability. This alignment between powdered catalyst activity and membrane separation efficiency highlights the potential of AGCNT as a more effective NC for multifunctional water treatment applications compared to ABMOT.

## CHAPTER 5: CONCLUSIONS

### 5.1 Conclusion

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This study successfully synthesized ABMOT and AGCNT NCs via a POME-induced, microwave-assisted green synthesis approach, enabling rapid, energy-efficient formation of well-crystallized ternary heterojunctions using POME as a renewable reducing agent. This method reduced reaction time to minutes, avoided hazardous chemicals, and demonstrated a sustainable route for producing advanced photocatalysts.

Excellent photocatalytic performance was achieved through matrix-controlled silver speciation, as confirmed by XRD, FTIR, and SEM–EDX analyses. In the ABMOT system, Ag was predominantly present in the metallic state ( $\text{Ag}^0$ ), likely due to favorable interfacial electron transfer and charge trapping at the Ag– $\text{Bi}_2\text{MoO}_6/\text{TiO}_2$  interfaces, which helped stabilize  $\text{Ag}^0$  during synthesis and photocatalytic operation. In contrast, in the AGCNT system, Ag was mainly stabilized as  $\text{Ag}_2\text{O}$ , influenced by the oxidative surface environment of g- $\text{C}_3\text{N}_4$  and its interactions with oxygen-containing species, leading to the formation of  $\text{Ag}_2\text{O}$  and p–n heterojunctions with  $\text{TiO}_2$ . These heterostructures enhanced charge separation and light absorption, resulting in higher degradation efficiency at lower to moderate OTC concentrations. These complementary trends highlight that the composition of the composite matrix and interfacial interactions play a key role in Ag speciation and photocatalytic performance under different application conditions.

The incorporation of NCs into PAN membranes via the phase-inversion method further improved their functional performance. FTIR confirmed the introduction of hydrophilic functional groups, while SEM imaging revealed increased surface roughness, and EDX mapping showed a uniform distribution of the NCs across the membrane surface. These structural changes led to substantial improvements in pure water flux and rejection compared with bare PAN. AGCNT@PAN (2%) achieved the highest PWF (170.94

$\text{L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ ) and rejection (90.34%), while ABMOT@PAN (2%) maintained a high PWF ( $163.48 \text{ L}\cdot\text{m}^{-2}\cdot\text{h}^{-1}$ ) with slightly lower rejection (85.69%), confirming that NC integration significantly enhanced both permeability and selectivity.

When comparing powdered and immobilized samples for OTC removal, powdered ABMOT and AGCNT showed higher intrinsic degradation rates because they were directly suspended in the reaction medium, ensuring maximum light exposure and catalyst–pollutant interaction. However, immobilized NCs in PAN membranes still achieved effective degradation while offering operational advantages, including ease of recovery, reduced post-treatment separation steps, and potential for continuous-flow applications. Although photocatalytic activity was slightly reduced in the immobilized form due to a lower exposed surface area, the membranes demonstrated stable performance and practical applicability, especially for scalable water treatment systems.

In summary, the research demonstrated the synergistic integration of green synthesis, matrix-optimized photocatalytic performance, and membrane technology to produce multifunctional materials capable of effectively removing OTC in both slurry and immobilized configurations. These findings provide a strong platform for advancing sustainable photocatalytic membrane systems tailored for real-world water purification challenges.

## 5.2 Recommendation

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Future work on this study should focus on further optimization of the POME-induced, microwave-assisted synthesis parameters, including Ag precursor concentration, microwave power, irradiation time, and POME extract composition, to fine-tune particle size, morphology, and silver speciation. Given the strong influence of silver phase composition on photocatalytic activity, controlled synthesis strategies such as sequential reduction–oxidation steps or selective post-synthesis thermal treatments could be explored to tailor the dominant Ag species for targeted applications. The incorporation process into PAN membranes could also be refined by optimizing nanocomposite loading levels, dispersion methods, and casting parameters to achieve an improved balance between permeability and rejection.

While the current study has demonstrated excellent OTC degradation performance under controlled laboratory conditions, future work should extend this investigation to real-world water matrices, such as greywater, municipal wastewater, and industrial effluents, to assess robustness against complex contaminant mixtures. Pilot-scale demonstrations using both slurry and immobilized membrane configurations should be conducted outdoors under sunlight to evaluate performance under variable environmental conditions, along with assessments of silver leaching and potential ecotoxicity. Mechanistic studies, including charge-carrier dynamics and pathways for reactive oxygen species generation, should be conducted using advanced spectroscopic techniques to better understand the photocatalytic processes at play. Furthermore, extended operational trials should examine how membrane surface morphology, hydrophilicity, and fouling resistance evolve over time. Comparative analyses between powdered and immobilized systems should also be expanded to incorporate catalyst recovery efficiency, operational costs, and energy requirements. Finally, techno-economic and life-cycle assessments are recommended to evaluate the scalability, cost-effectiveness, and sustainability of deploying these materials in full-scale water treatment systems.

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# APPENDICES

## Appendix 1: Raw data analysis of OTC degradation by parameters

### Effect of dosage

#### Catalyst: Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>

##### 0.03 g

Time (Minutes)	Abs 1	Abs 2	Abs 3	Average	Concentration	C/C <sub>0</sub>	Removal (%)
0	0.1928	0.1927	0.1927	0.1927	10.4751	1.0000	0.0000
30	0.0955	0.0948	0.0946	0.0949	5.1601	0.4926	50.7393
60	0.0957	0.0956	0.0957	0.0957	5.1988	0.4963	50.3698
90	0.0484	0.0484	0.0484	0.0484	2.6308	0.2511	74.8852
120	0.0367	0.0367	0.0367	0.0367	1.9970	0.1906	80.9356
150	0.0341	0.0341	0.0341	0.0341	1.8536	0.1770	82.3045
180	0.0224	0.0226	0.0225	0.0225	1.2238	0.1168	88.3170

##### 0.06 g

Time (Minutes)	Abs 1	Abs 2	Abs 3	Average	Concentration	C/C <sub>0</sub>	Removal (%)
0	0.1930	0.1950	0.1980	0.1953	10.6159	1.0000	0.0000
30	0.0910	0.0920	0.0910	0.0913	4.9638	0.4676	53.2423
60	0.0900	0.0900	0.0910	0.0903	4.9094	0.4625	53.7543
90	0.0250	0.0290	0.0280	0.0273	1.4855	0.1399	86.0068
120	0.0190	0.0190	0.0180	0.0187	1.0145	0.0956	90.4437
150	0.0110	0.0110	0.0120	0.0113	0.6159	0.0580	94.1980
180	0.0050	0.0057	0.0056	0.0054	0.2953	0.0278	97.2184

**0.12 g**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>0</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.1960	0.1980	0.1960	0.1967	10.6884	1.0000	0.0000
<b>30</b>	0.0650	0.0650	0.0660	0.0653	3.5507	0.3322	66.7797
<b>60</b>	0.0660	0.0650	0.0660	0.0657	3.5688	0.3339	66.6102
<b>90</b>	0.0230	0.0230	0.0220	0.0227	1.2319	0.1153	88.4746
<b>120</b>	0.0200	0.0220	0.0220	0.0213	1.1594	0.1085	89.1525
<b>150</b>	0.0190	0.0170	0.0200	0.0187	1.0145	0.0949	90.5085
<b>180</b>	0.0160	0.0160	0.0160	0.0160	0.8696	0.0814	91.8644

**Catalyst: Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>**

**0.03 g**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>0</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.2090	0.2100	0.2150	0.2113	11.4855	1.0000	0.0000
<b>30</b>	0.1910	0.1850	0.1900	0.1887	10.2536	0.8927	10.7256
<b>60</b>	0.1900	0.1910	0.1900	0.1903	10.3442	0.9006	9.9369
<b>90</b>	0.1690	0.1740	0.1690	0.1707	9.2754	0.8076	19.2429
<b>120</b>	0.1470	0.1450	0.1490	0.1470	7.9891	0.6956	30.4416
<b>150</b>	0.1250	0.1200	0.1230	0.1227	6.6667	0.5804	41.9558
<b>180</b>	0.1080	0.1090	0.1080	0.1083	5.8877	0.5126	48.7382

**0.06 g**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>o</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.2130	0.2110	0.2130	0.2123	11.5399	1.0000	0.0000
<b>30</b>	0.1620	0.1610	0.1630	0.1620	8.8043	0.7630	23.7049
<b>60</b>	0.1630	0.1620	0.1620	0.1623	8.8225	0.7645	23.5479
<b>90</b>	0.0820	0.0880	0.0890	0.0863	4.6920	0.4066	59.3407
<b>120</b>	0.0620	0.0660	0.0610	0.0630	3.4239	0.2967	70.3297
<b>150</b>	0.0500	0.0550	0.0490	0.0513	2.7899	0.2418	75.8242
<b>180</b>	0.0400	0.0400	0.0390	0.0397	2.1558	0.1868	81.3187

**0.12 g**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>o</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.2110	0.2100	0.2090	0.2100	11.4130	1.0000	0.0000
<b>30</b>	0.1540	0.1590	0.1580	0.1570	8.5326	0.7476	25.2381
<b>60</b>	0.1560	0.1560	0.1580	0.1567	8.5145	0.7460	25.3968
<b>90</b>	0.1040	0.1050	0.1040	0.1043	5.6703	0.4968	50.3175
<b>120</b>	0.0700	0.0710	0.0700	0.0703	3.8225	0.3349	66.5079
<b>150</b>	0.0590	0.0590	0.0580	0.0587	3.1884	0.2794	72.0635
<b>180</b>	0.0480	0.0490	0.0500	0.0490	2.6630	0.2333	76.6667

## Effect of pollutant initial concentration

Catalyst: Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>

10 ppm

Time (Minutes)	Abs 1	Abs 2	Abs 3	Average Concentration	C/C <sub>0</sub>	Removal (%)	
0	0.1930	0.1950	0.1980	0.1953	10.6159	1.0000	0.0000
30	0.0910	0.0920	0.0910	0.0913	4.9638	0.4676	53.2423
60	0.0900	0.0900	0.0910	0.0903	4.9094	0.4625	53.7543
90	0.0250	0.0290	0.0280	0.0273	1.4855	0.1399	86.0068
120	0.0190	0.0190	0.0180	0.0187	1.0145	0.0956	90.4437
150	0.0110	0.0110	0.0120	0.0113	0.6159	0.0580	94.1980
180	0.0050	0.0057	0.0056	0.0054	0.2953	0.0278	97.2184

15 ppm

Time (Minutes)	Abs 1	Abs 2	Abs 3	Average Concentration	C/C <sub>0</sub>	Removal (%)	
0	0.2820	0.2830	0.2810	0.2820	15.3261	1.0000	0.0000
30	0.1570	0.1590	0.1578	0.1579	8.5833	0.5600	43.9953
60	0.1500	0.1510	0.1560	0.1523	8.2790	0.5402	45.9811
90	0.0510	0.0515	0.0516	0.0514	2.7917	0.1822	81.7849
120	0.0390	0.0391	0.0380	0.0387	2.1033	0.1372	86.2766
150	0.0240	0.0243	0.0239	0.0241	1.3080	0.0853	91.4657
180	0.0180	0.0183	0.0181	0.0181	0.9855	0.0643	93.5697

**20 ppm**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>0</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.3600	0.3580	0.3560	0.3580	19.4565	1.0000	0.0000
<b>30</b>	0.2280	0.2240	0.2285	0.2268	12.3279	0.6336	36.6387
<b>60</b>	0.2300	0.2310	0.2305	0.2305	12.5264	0.6438	35.6182
<b>90</b>	0.1090	0.1080	0.1050	0.1073	5.8333	0.2998	70.0186
<b>120</b>	0.0770	0.0760	0.0760	0.0763	4.1486	0.2132	78.6778
<b>150</b>	0.0580	0.0590	0.0584	0.0585	3.1775	0.1633	83.6685
<b>180</b>	0.0500	0.0501	0.0509	0.0503	2.7355	0.1406	85.9404

**Catalyst: Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>****10 ppm**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>0</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.2130	0.2110	0.2130	0.2123	11.5399	1.0000	0.0000
<b>30</b>	0.1620	0.1610	0.1630	0.1620	8.8043	0.7630	23.7049
<b>60</b>	0.1630	0.1620	0.1620	0.1623	8.8225	0.7645	23.5479
<b>90</b>	0.0820	0.0880	0.0890	0.0863	4.6920	0.4066	59.3407
<b>120</b>	0.0620	0.0660	0.0610	0.0630	3.4239	0.2967	70.3297
<b>150</b>	0.0500	0.0550	0.0490	0.0513	2.7899	0.2418	75.8242
<b>180</b>	0.0400	0.0400	0.0390	0.0397	2.1558	0.1868	81.3187

**15 ppm**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>o</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.2840	0.2880	0.2850	0.2857	15.5254	1.0000	0.0000
<b>30</b>	0.2310	0.2330	0.2310	0.2317	12.5906	0.8110	18.9032
<b>60</b>	0.2340	0.2330	0.2330	0.2333	12.6812	0.8168	18.3197
<b>90</b>	0.1550	0.1530	0.1550	0.1543	8.3877	0.5403	45.9743
<b>120</b>	0.1070	0.1090	0.1060	0.1073	5.8333	0.3757	62.4271
<b>150</b>	0.0760	0.0760	0.0780	0.0767	4.1667	0.2684	73.1622
<b>180</b>	0.0570	0.0560	0.0550	0.0560	3.0435	0.1960	80.3967

**20 ppm**

<b>Time (Minutes)</b>	<b>Abs 1</b>	<b>Abs 2</b>	<b>Abs 3</b>	<b>Average</b>	<b>Concentration</b>	<b>C/C<sub>o</sub></b>	<b>Removal (%)</b>
<b>0</b>	0.3790	0.3780	0.3790	0.3787	20.5797	1.0000	0.0000
<b>30</b>	0.3160	0.3160	0.3150	0.3157	17.1558	0.8336	16.6373
<b>60</b>	0.3130	0.3150	0.3140	0.3140	17.0652	0.8292	17.0775
<b>90</b>	0.2900	0.2910	0.2930	0.2913	15.8333	0.7694	23.0634
<b>120</b>	0.2630	0.2660	0.2630	0.2640	14.3478	0.6972	30.2817
<b>150</b>	0.2390	0.2380	0.2410	0.2393	13.0072	0.6320	36.7958
<b>180</b>	0.2180	0.2180	0.2170	0.2177	11.8297	0.5748	42.5176

Appendix 2: Raw data analysis of antibacterial activity against *E. Coli* and *S. Aureus*

**Control Experiment**

***E. Coli***

Sample	ZOI 1 (mm)	ZOI 2 (mm)	ZOI 3 (mm)	Average (mm)	STD
Blank	0.0000	0.0000	0.0000	0.0000	0.0000
TiO <sub>2</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
Bi <sub>2</sub> MoO <sub>6</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
BMOT	0.0000	0.0000	0.0000	0.0000	0.0000
ABMOT (1%)	6.2000	5.2000	7.3000	6.2333	1.0504
ABMOT (4%)	9.3000	8.4000	7.7500	8.4833	0.7784
ABMOT (8%)	10.9500	9.3500	10.5000	10.2667	0.8251
ABMOT (10%)	14.1000	13.4500	13.5000	13.6833	0.3617

Sample	ZOI 1 (mm)	ZOI 2 (mm)	ZOI 3 (mm)	Average (mm)	STD
Blank	0.0000	0.0000	0.0000	0.0000	0.0000
TiO <sub>2</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
g-C <sub>3</sub> N <sub>4</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
GCNT	0.0000	0.0000	0.0000	0.0000	0.0000
AGCNT (1%)	6.6500	6.8000	7.5500	7.0000	0.4822
AGCNT (4%)	7.1000	6.6500	7.3000	7.0167	0.3329
AGCNT (8%)	8.2500	6.7000	7.1000	7.3500	0.8047
AGCNT (10%)	8.8000	8.8000	7.7000	8.4333	0.6351

***S. Aureus***

<b>Sample</b>	<b>ZOI 1 (mm)</b>	<b>ZOI 2 (mm)</b>	<b>ZOI 3 (mm)</b>	<b>Average (mm)</b>	<b>STD</b>
Blank	0.0000	0.0000	0.0000	0.0000	0.0000
TiO <sub>2</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
Bi <sub>2</sub> MoO <sub>6</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
BMOT	0.0000	0.0000	0.0000	0.0000	0.0000
ABMOT (1%)	0.0000	0.0000	0.0000	0.0000	0.0000
ABMOT (4%)	8.1000	7.1000	6.8500	7.3500	0.6614
ABMOT (8%)	6.3500	7.2000	7.8000	7.1167	0.7286
ABMOT (10%)	10.3000	11.4000	9.2500	10.3167	1.0751

<b>Sample</b>	<b>ZOI 1 (mm)</b>	<b>ZOI 2 (mm)</b>	<b>ZOI 3 (mm)</b>	<b>Average (mm)</b>	<b>STD</b>
Blank	0.0000	0.0000	0.0000	0.0000	0.0000
TiO <sub>2</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
g-C <sub>3</sub> N <sub>4</sub>	0.0000	0.0000	0.0000	0.0000	0.0000
GCNT	0.0000	0.0000	0.0000	0.0000	0.0000
AGCNT (1%)	0.0000	0.0000	0.0000	0.0000	0.0000
AGCNT (4%)	7.5000	7.0000	8.3000	7.6000	0.6557
AGCNT (8%)	8.1000	8.2000	9.0000	8.4333	0.4933
AGCNT (10%)	9.5000	9.5000	8.8000	9.2667	0.4041

## Effect of catalyst concentration

Catalyst: Ag/Bi<sub>2</sub>MoO<sub>6</sub>/TiO<sub>2</sub>

*E. Coli*

Concentration (g/L)	ZOI 1 (mm)	ZOI 2 (mm)	ZOI 3 (mm)	Average (mm)	STD
10	12.1000	13.4000	13.5000	13.0000	0.7810
8	10.3000	10.3000	8.7500	9.7833	0.8949
4	9.4000	9.7000	11.8500	10.3167	1.3364
2	10.4000	10.7000	9.8000	10.3000	0.4583
1	8.1500	8.4500	10.6500	9.0833	1.3650

*S. Aureus*

Concentration (g/L)	ZOI 1 (mm)	ZOI 2 (mm)	ZOI 3 (mm)	Average (mm)	STD
10	13.5500	13.8500	10.0000	12.2583	2.1415
8	11.5500	11.8000	10.5000	9.7000	0.6898
4	9.7000	9.4000	9.5500	8.4833	0.1500
2	7.0000	7.9000	8.4500	7.5000	0.7320
1	6.7000	7.4000	6.4000	6.7750	0.5132

Catalyst: Ag/g-C<sub>3</sub>N<sub>4</sub>/TiO<sub>2</sub>

*E. Coli*

Concentration (g/L)	ZOI 1 (mm)	ZOI 2 (mm)	ZOI 3 (mm)	Average (mm)	STD
10	11.1000	11.4000	9.8000	10.7667	0.8505
8	9.6500	13.3000	12.3000	11.7500	1.8861
4	8.4500	11.5000	9.4500	9.8000	1.5548
2	8.7500	8.4500	9.0000	8.7333	0.2754
1	8.0000	9.8000	8.4000	8.7333	0.9452

*S. Aureus*

<b>Concentration (g/L)</b>	<b>ZOI 1 (mm)</b>	<b>ZOI 2 (mm)</b>	<b>ZOI 3 (mm)</b>	<b>Average (mm)</b>	<b>STD</b>
10	11.7500	11.9000	12.5000	12.0500	0.3969
8	8.4500	8.8000	7.1000	8.1167	0.8977
4	7.6500	7.0000	7.6000	7.4167	0.3617
2	6.9000	7.4500	7.3000	7.2167	0.2843
1	7.2500	6.4000	6.5000	6.7167	0.4646